## Semiconductor-based Photocatalytic Hydrogen Generation

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#### 1. Introduction

Because of its high energy capacity and environmental friendliness, hydrogen has been identified as a potential energy carrier in many low greenhouse gas (GHG) energy scenarios.<sup>1,2</sup> In a proposed hydrogen energy system,<sup>3</sup> hydrogencontaining compounds such as fossil fuels, biomass, or even water are potential sources of hydrogen.<sup>4–11</sup> When hydrogen is derived from hydrocarbons such as fossil fuels or biomass, CO<sub>2</sub> capture and sequestration are requirements in a low GHG scenario. 12-14 On the other hand, hydrogen produced from water does not present the challenge of unwanted emissions at the point of conversion, but it does require that energy be supplied from an external resource. 15 If this energy can be obtained from a renewable energy source such as solar energy, hydrogen can then be considered a green energy alternative capable of powering everything from laptops to submarines. Figure 1 shows a diagram of photocatalytic hydrogen generation in the hydrogen energy system. Such an approach to energy production is one that exhibits due concern for environmental issues and that is becoming increasingly relevant in our world. However, the technology to produce hydrogen in a cost-effective, low-GHG manner has not yet been developed.

Since the discovery of hydrogen evolution through the photoelectrochemical splitting of water on n-type TiO<sub>2</sub>

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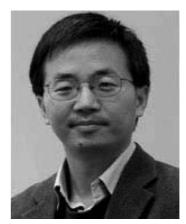


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electrodes,18 the technology of semiconductor-based photocatalytic water splitting to produce hydrogen using solar energy has been considered as one of the most important approaches to solving the world energy crisis. 19 Hence, the



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development of the necessary semiconductor photocatalysts has undergone considerable research. Over the past 40 years, many photocatalysts reportedly exhibited high photocatalytic activities for splitting water into a stoichiometric mixture of H<sub>2</sub> and O<sub>2</sub> (2:1 by molar ratio) in the ultraviolet (UV) light region. These include La doped NaTaO<sub>3</sub>,  $^{20}$  Sr<sub>2</sub>M<sub>2</sub>O<sub>7</sub> (M = Nb, Ta),  $^{21}$  La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub>,  $^{22}$  K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub>,  $^{23}$  and  $\beta$ -Ge<sub>3</sub>N<sub>4</sub>,  $^{24}$  among others. Of particular note is the NiO/NaTaO3:La photocatalyst, which shows the highest activity with a quantum yield amounting to 56% at 270 nm.20 However, these oxide photocatalysts are only active under UV irradiation. With respect to the solar spectrum, only a small fraction (ca. 4%) of the incoming solar energy lies in the ultraviolet region, whereas the visible light in the solar spectrum is far more abundant (ca. 46%). It is essential, therefore, as an alternative to UV-active photocatalysts to develop visible-light-driven photocatalysts that are stable and highly efficient for the practical, large-scale production of hydrogen using solar energy. Over the recent years, continuing breakthroughs have been made in the development of novel visible-light-driven photocatalysts, leading to the enhancement of photocatalytic activity for water splitting and inspiring great enthusiasm. A large number of semiconductor materials have been developed as photocatalysts for water splitting to hydrogen under visible-light irradiation.

A significant process has been achieved on semiconductorbased photocatalytic hydrogen generation through water

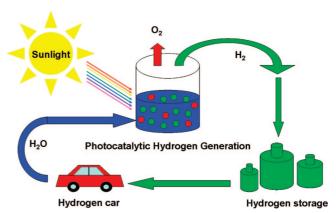


Figure 1. Schematic diagram of photocatalytic hydrogen generation in the hydrogen energy system.

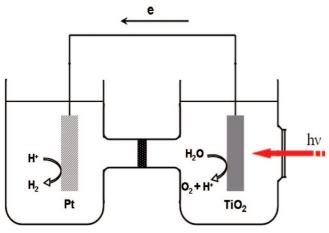
splitting over the past several decades, <sup>25–31</sup> and many excellent reviews have been published. <sup>32–62</sup> In this review, we aim to put together the research effort having been made so far, with a view of providing a good reference and inspiring new ideas for tackling this important challenge. Starting with a brief introduction to semiconductor-based photocatalysts for hydrogen generation from water splitting, we overview the development of high-efficiency, visible-light-driven photocatalysts. A number of synthetic and modification techniques for adjusting the band structure to harvest visible light and improve the charge separation in photocatalysis are discussed. Photocatalytic systems for water splitting are also reviewed and classified into two main kinds: sacrificial reagent-containing water-splitting systems and overall water-splitting systems.

## 2. Basic Principles of Photocatalytic Hydrogen Generation

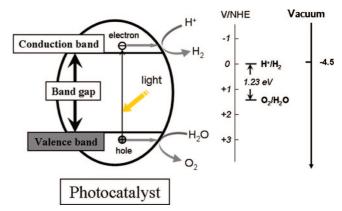
## 2.1. Fundamental Mechanism of Photocatalytic Hydrogen Generation

In Fujishima and Honda's pioneering work, the electrochemical cell they constructed for the decomposition of water into hydrogen and oxygen is shown in Figure 2.  $^{18}$  When the surface of the  $\rm TiO_2$  electrode was irradiated by UV light, as a result of a water oxidation reaction, oxygen evolution occurred at the  $\rm TiO_2$  electrode. Concomitant reduction led to hydrogen evolution at the platinum black electrode. This concept, which emerged from the use of photoelectrochemical cells with semiconductor electrodes, was later applied by Bard to the design of a photocatalytic system using semiconductor particles or powders as photocatalysts.  $^{63-65}$ 

A photocatalyst absorbs UV and/or visible (Vis) light irradiation from sunlight or an illuminated light source. The electrons in the valence band of the photocatalyst are excited to the conduction band, while the holes are left in the valence band. This, therefore, creates the negative-electron (e<sup>-</sup>) and positive-hole (h<sup>+</sup>) pairs. This stage is referred to the semiconductor's "photo-excited" state, and the energy difference between the valence band and the conduction band is known as the "band gap". This must correspond to the wavelength of the light for it to be effectively absorbed by the photocatalyst. After photoexcitation, the excited electrons and holes separate and migrate to the surface of photocatalyst. Here, in the photocatalytic water-splitting reaction, they act



**Figure 2.** Schematic representation of a photoelectrochemical cell (PEC). Reprinted with permission from ref 18. Copyright 1972 Nature Publishing Group.



**Figure 3.** Fundamental principle of semiconductor-based photocatalytic water splitting for hydrogen generation.

as reducing agent and oxidizing agent to produce  $H_2$  and  $O_2$ , respectively. A schematic representation of the principle of the photocatalytic system for water is depicted in Figure 3.

Water splitting into  $H_2$  and  $O_2$  is an uphill reaction. It needs the standard Gibbs free energy change  $\Delta G^0$  of 237 kJ/mol or 1.23 eV, as shown in eq 1.

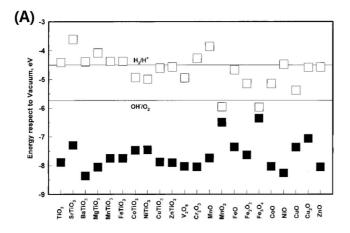
$$H_2O \rightarrow \frac{1}{2}O_2 + H_2; \quad \Delta G = +237 \text{ kJ/mol}$$
 (1)

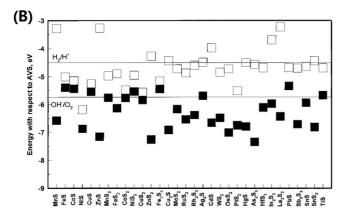
Therefore, the band gap energy ( $E_g$ ) of the photocatalyst should be >1.23 eV (<1000 nm) to achieve water splitting. However, to use visible light, it should be <3.0 eV (>400 nm).

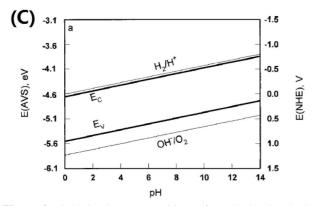
To facilitate both the reduction and oxidation of H<sub>2</sub>O by photoexcited electrons and holes, the match of the band gap and the potentials of the conduction and valence bands are important. Both the reduction and oxidation potentials of water should lie within the band gap of the photocatalyst. The bottom level of the conduction band has to be more negative than the reduction potential of  $H^+/H_2$  (0 V vs normal hydrogen electrode (NHE)), whereas the top level of the valence band has to be more positive than the oxidation potential of O<sub>2</sub>/H<sub>2</sub>O (1.23 V). Parts A and B of Figure 4 show the conduction band edge and valence band edge of some oxide- and sulfide-based semiconductor materials.<sup>66</sup> We can see that there are many semiconductor systems whose electronic structures match well with the redox potential of water into hydrogen and oxygen molecules. The band structure requirement is a thermodynamic requirement for water splitting. Other factors, such as overpotentials, charge separation, mobility, and lifetime of photogenerated electrons and holes, affect the photocatalytic generation of hydrogen from water splitting as well. For example, the band edges of the semiconductor photocatalyst usually vary with the change of pH, as shown in Figure 4C.<sup>66</sup> The phase stability of the semiconductor photocatalyst changes in different pH environments as well.

## 2.2. Main Processes of Photocatalytic Hydrogen Generation

The processes in the photocatalytic generation of hydrogen are illustrated in Figure 5. They include light absorption of the semiconductor photocatalyst, generation of excited charges (electrons and holes), recombination of the excited charges, separation of excited charges, migration of the charges, trap of excited charges, and transfer of excited



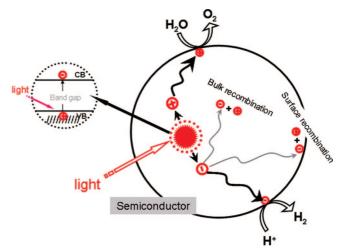




**Figure 4.** Calculated energy positions of conduction band edges and valence band edges at pH 0 for selected metal oxide (A) and metal sulfide (B) semiconductors. The bottom of open squares represent conduction band edges, and the top of solid squares represent valence band edges. The solid lines indicate water stability limits. (C) pH dependence of the conduction band edge and valence band edge in an aqueous electrolyte solution. Reprinted with permission from ref 66. Copyright 2000 The Mineralogical Society of America.

charges to water or other molecules. All of these processes affect the final generation of hydrogen from the semi-conductor photocatalyst system.

The total amount of hydrogen generated is mainly determined by the amount of excited electrons in the water/photocatalyst interface in reducing water. Apparently, any other processes that consume excited electrons should be avoided in order to maximize the efficiency of the hydrogen generation of the photocatalyst system. Any process that generates excited electrons should be considered to act in a possible way to improve the efficiency. Thus, if we look at



**Figure 5.** Processes in photocatalytic water splitting. Reprinted with permission from ref 67. Copyright 1995 American Chemical Society.

the charge-generation process, the semiconductor photocatalyst should first have a low band gap to absorb as much light as possible, and reflection or scattering of light by the photocatalyst should be minimized. Second, using the absorbed photons, the semiconductor photocatalyst should have a high efficieny in generating excited charges, instead of generating phonons or heat.

After excited charges are created, charge recombination and separation/migration processes are two important competitive processes inside the semiconductor photocatalyst that largely affect the efficiency of the photocatalytic reaction for water splitting.<sup>67</sup> Charge recombination reduces the excited charges by emitting light or generating phonons. It includes both surface and bulk recombination and is classified as a deactivation process, and it is ineffective for water splitting. The separation of excited electrons and holes sometimes may need to overcome an energy barrier, which is the sometimes binding energy of the excited electron—hole pairs, excitons. Charge separation and migration, on the other hand, is an activation process. This is as a result of the charges being on the surface of the photocatalyst ready for the desired chemical reaction. It is beneficial for hydrogen generation through water splitting. Efficient charge separation and fast charge transport, avoiding any bulk/surface charge recombination avoided, are fundamentally important for photocatalytic hydrogen generation through water splitting. Any approach beneficial to the charge separation and transport should be taken into account such as design of internal-built electric field and use of high photoconductive semiconductor materials.

The reaction of photogenerated  $H_2$  and  $O_2$  to form  $H_2O$  on the photocatalyst surface is normally called "surface backreaction (SBR)". It will inevitably have a negative effect on any enhancement of the photocatalytic activity, because it reduces the amount of  $H_2$  emitted from the photocatalyst. There are two main approaches to suppress SBR: one involves the addition of sacrificial reagents into the photocatalytic reaction environment and the second creates a separation of the photoactive sites on the surface of the photocatalysts. In general, the electron donor and acceptor sacrificial reagents that are added work as an external driving force for the surface chemical reaction and depress the  $H_2O$  formation from  $H_2$  and  $O_2$ . The separation of the photoactive sites necessary for hydrogen and oxygen evolution, and

which is always accompanied by the surface separation of the photogenerated electrons and holes, has been shown to be greatly affected by the surface properties of the photocatalysts. As well as the surface reaction sites themselves, the surface states and morphology also play an important role.

Taking into consideration the basic mechanism and processes of photocatalytic water splitting, there are two keys to developing a suitable high-efficiency semiconductor for the visible-light-driven photocatalytic splitting of water into H<sub>2</sub> and/or O<sub>2</sub>: (1) A photocatalyst should have a sufficiently narrow band gap (1.23 eV  $< E_g < 3.0$  eV) to both harvest visible light and possess the correct band structure. (2) Photoinduced charges in the photocatalyst should be separated efficiently in order to avoid bulk/surface electron/hole recombination. In addition, they must migrate to the photocatalyst surface for hydrogen and/or oxygen evolution at the respective photocatalytic active sites. 58 In summary, it is generally accepted that the correct band structure for efficient visible-light harvesting and effective separation between the photoexcited electrons and the holes is essential to improve the photocatalytic properties of the semiconductor. In addition, with the development of efficient visiblelight-driven photocatalysts, it is important that economical, highly efficient photocatalytic systems for light-to-hydrogen energy conversion, in which aqueous solutions containing sacrificial reagents can be used to depress the backward reaction of hydrogen and oxygen to water on the surface of photocatalysts, can be constructed.

## 2.3. Evaluation of Photocatalytic Water Splitting

There are two apparent indicators that should be paid attention in evaluating the hydrogen generation through photocatalytic water splitting. One is photocatalytic activity, and the other one is photocatalytic stability.

#### 2.3.1. Photocatalytic Activity

The efficiency of photocatalytic hydrogen generation from water splitting can be measured directly on the amount of hydrogen gas evoluted or indirectly on the electrons transferred from semiconductor to water within a certain time period under light irradiation. Different photocatalytic setup configurations, such as inner irradiation type and top irradiation type, and light sources, such as Xe lamp and Hg lamp, are commonly used by different research groups and scientists, which may give different rates of gas evolution when exactly the same photocatalyst is used. This makes it difficult to make direct comparison across the results from different research groups and photocatalytic hydrogen generation systems. Nevertherless, it seems helpful to get approximate correlations between various results if we normalize the rates of gas evolution to the amount of photocatalyst employed within a unit of time. Here, we use the rate of gas (O2 and H2) evolution with units such as  $\mu$ mol·h<sup>-1</sup> and  $\mu$ mol·h<sup>-1</sup>·g<sup>-1</sup><sub>catalyst</sub> to make the mensurable comparison between different photocatalysts under similar experimental conditions.

The (apparent) quantum yield, as an extension from the overall quantum yield in a homogeneous photochemical system, becomes important and acceptable to evaluate the photocatalytic activity for water splitting. The overall quantum yield and apparent quantum yield are defined by eqs 2 and 3, respectively.<sup>68</sup>

Overall quantum yield (%) =  $\frac{\text{Number of reacted electrons}}{\text{Number of absorbed photons}} \times 100\% \quad (2)$ 

 $(Apparent) \ Quantum \ yield \ (QY, \%) = \\ \frac{Number \ of \ reacted \ electrons}{Number \ of \ incident \ photons} \times 100\% = \\ \frac{2 \times Number \ of \ evolved \ H_2 \ molecules}{Number \ of \ incident \ photons} \times \\ 100\% \ (for \ H_2 \ evolution) = \\ \frac{4 \times Number \ of \ evolved \ O_2 \ molecules}{Number \ of \ incident \ photons} \times \\ 100\% \ (for \ O_2 \ evolution)$ 

The apparent quantum yield is estimated to be smaller than the total quantum yield because the number of absorbed photons is usually smaller than that of incident light. In addition to the quantum yield, the solar energy conversion efficiency that is usually used for evaluation of solar cells is also sometimes reported in the literature. It is defined as

Solar energy conversion (%) =  $\frac{\text{Output energy of hydrogen evolved}}{\text{Energy of incident solar light}} \times 100 \quad (4)$ 

## 2.3.2. Photocatalytic Stability

As a good photocatalyst, it should have a good stability for  $H_2$  and/or  $O_2$  production, besides a high photocatalytic activity or quantum yield. To test the photocatalytic stability, a long-time experiment or a repeated experiment is always necessary. Photocorrosion is considered to be the main reason causing the poor stability of photocatalysts, especially the metal sulfide photocatalysts. CdS has frequently been reported to be unstable for photocatalytic  $H_2$  evolution.  $S^{2-}$  in CdS rather than water is self-oxidized by photoinduced holes in the valence band of CdS. The photocorrosion reaction occurs as in eq 5,  $^{55}$ 

$$CdS + 2h^{+} \rightarrow Cd^{2+} + S \tag{5}$$

### 3. UV-Active Photocatalysts for Water Splitting

A wide range of semiconducting materials have been developed as photocatalysts for use under UV irradiation. These are shown in Table 1. On the basis of their electronic configuration properties, these UV-active photocatalysts can be typically classified into four groups: (1)  $d^0$  metal ( $Ti^{4+},\ Zr^{4+},\ Nb^{5+},\ Ta^{5+},\ W^{6+},\ and\ Mo^{6+})$  oxide photocatalysts, (2)  $d^{10}$  metal ( $In^{3+},\ Ga^{3+},\ Ge^{4+},\ Sn^{4+},\ and\ Sb^{5+})$  oxide photocatalysts, (3)  $f^0$  metal ( $Ce^{4+}$ ) oxide photocatalysts, and (4) a small group of nonoxide photocatalysts.

## 3.1. d<sup>0</sup> Metal Oxide Photocatalyts

#### 3.1.1. Ti-, Zr-Based Oxides

 $TiO_2$  is the first reported photocatalyst for water splitting under UV irradiation. <sup>69</sup>  $TiO_2$  can produce hydrogen and/or oxygen from water vapor, pure water, and aqueous solutions containing electron donor. <sup>70–77</sup> It was found that, under UV irradiation, colloidal  $TiO_2$  combined with ultrafine Pt and  $RuO_2$  particles generated  $H_2$  with a high quantum yield of  $30 \pm 10\%$  and  $O_2$  in stoichiometric proportions from water. <sup>70</sup> The reaction solution had a pH of 1.5, which was adjusted

Table 1. UV-Light-Active Photocatalysts for Water Splitting to Hydrogen and/or Oxygen

activity  $(\mu \text{mol} \cdot \text{h}^{-1} \cdot \text{g}^{-1})$ 

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Fau (,0 0,2 0,2 0,2 0,2 0,2 0,7 0,2			200 W Uz	Dimen 61ton	CHOH	D+1-5667			0
r r r r O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> /ZrO <sub>2</sub>		•	8H W-00	ryrex IIIter	CH <sub>3</sub> OH	FU~366./			90
FEu 5.0 5.0 1.0.2 1.0.2 1.0.2 1.0.2 1.0.2 1.0.2 1.0.2 1.0.2 1.0.2		0.1	300-W Hg	Pyrex filter	CH3OH	$Pt/\sim 7500$			82
Fu (O O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> Co <sub>2</sub> /ZrO <sub>2</sub>		0.1	300-W Ho	Pyrex filter	CHOH	Pt/~7800			82
r FEu 50 00, 1100, de de 7270,			200 W H2	Dimen 61ton	CHOH	D#/~ 7690			100
Fau (O) (O) (O) (O) (O) (O) (O) (O) (O) (O)			00-w Hg	Fyrex Illier	CH <sub>3</sub> OH	FU~/680			70
r FBu O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> /ZrO <sub>2</sub>	calcination 0	Ξ.	300-W Hg	Pyrex filter	CH3OH	Pt/~4800			82
Eu Fu 5,0 5,2 110,2 160,2 02/ZrO2,		0.1	300-W Hg	Pyrex filter	CH <sub>2</sub> OH	Pt/~6600			82
Fu (O O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> /ZrO <sub>2</sub>		_	O W H	Dyray filter	ПОП	D+/~,0300			63
1. 5.0 5.0 0.2 11.02, 6.02/ZrO <sub>2</sub>			811 W-00	r yrea mier	CHISCH	7, 1990			79
Eu ,O ,O ,O ,O ,O ,O ,O ,O ,O ,O ,O ,O ,O		_	300-w Hg	Pyrex niter	CH3OH	$VV \sim 10200$			78
n Fau O₂ O₂ O₂/ZrO₂		0.1	300-W Hg	Pyrex filter	CH <sub>3</sub> OH	$Pt/\sim 10800$			82
Fu Fu 50 50 60 70 70 70 70 70 70 70		0.1	300-W Ho	Pyrex filter	CHOH	$P_t/\sim 13200$			82
1 Fau O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> /ZrO <sub>2</sub>		_	200 W US	Dimon filton	пОп	D+/- 1000			6
n FEu 50 00, 1100, de de 7270,			811 W-00	r yrea mier	CHISCH	1 U - 4800			79
1. Eu 5.0 5.2 11.0 11.0 10.2 0.2/ZrO <sub>2</sub>		0.1	300-W Hg	Pyrex filter	CH <sub>3</sub> OH	Pt/~8400			78
r Fau O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> O <sub>2</sub> /ZrO <sub>2</sub>		0.1	300-W Hg	Pyrex filter	CHOH	$Pt/\sim 10200$			82
r FBu C, O O <sub>2</sub> O <sub>2</sub> GiO <sub>2</sub> de O <sub>2</sub> /ZrO <sub>2</sub>	Calcination		300-W Ha	Dyrey filter	CHOH	Dt/~8400			82
Eu ,O ,O ,O ,O ,O ,O ,C ,C ,O ,C ,C ,O			200 W 115	Piroz filter	CIISOII	D#/- 9700			70 8
Eu ,0 52 00, 7170, 60,/ZrO,			300-w rig	ryrex IIIter	CH3OH	FU~8/00			70
Fau 6,0 2,2 0,2 11,02 de 0,2/ZrO <sub>2</sub>		0.1	00-W Hg	Pyrex filter	CH <sub>3</sub> OH	Pt/~8400			82
Eu ,O ,O <sub>2</sub> TiO <sub>2</sub> de O <sub>2</sub> /ZrO <sub>2</sub>		0.1	00-W Hg	Pyrex filter	CH <sub>2</sub> OH	$Pt/\sim 7200$			82
,0 52,0 10,0 110,0 02/ZrO2		0.2	288-W Fd	guartz filter	CHOH	Pd/~92		40.4	81
 02 03 1102 02/ZrO2	pethod		400 W Ha	anartz filter	CH.OH	18 500			85-87
0.2 0.2 II:0.2 de 0.2/ZrO.2		200	200 W E	quartz filtər	CHOIL	Dd/2.02			63
O., IiO., de O./ZrO.			7 M-00	quantz miter	Ch3On	FU/~03			03
TiO <u>2</u> de O <sub>2</sub> /ZrO <sub>2</sub>	pc	0.	solar light	quartz filter	CH <sub>3</sub> OH	000 / 9			88-90
de ${ m O}_2/{ m ZrO}_2$			150-W Hg		HC00H	560			91
$O_2/ZrO_2$		0.3	400-W Hg	quartz filter	pure water	Pt/73	Pt/36.7		92, 93
7) 1 7	need self-assembly process	,	500-W Ho	Pyrex filter	CH <sub>2</sub> OH	Pt/~24			84
Call C			500-W Ho	silica olass	HOEN	Pt/76	Pt/18		120
			400 W Hg	Driver filter	NoOH	0E 0/ 0EN	NIO (2.22		100-114
			81 W-00-	ryica iiitei	NaOn	MON' ~ /0	INIOx/:~32		109-114
			400-w Hg	quartz niter	Na <sub>2</sub> CO <sub>3</sub>	CoO <sub>x</sub> /~2800			911
			450-W Hg	Pyrex filter	pure water	$Rh_xCr_{2-x}O_3/\sim 14\ 160$			117
SrTiO <sub>3</sub> :Na solid-state reaction		0.3	450-W Hg	Pyrex filter	pure water	$Rh_xCr_{2-x}O_3/\sim 22\ 220$		0	117
Sr <sub>3</sub> Ti <sub>2</sub> O <sub>7</sub> polymerized c	polymerized complex method	4	400-W Hg	quartz filter	pure water	NiO,/144	NiO,/72		118
	polymerized complex method	7	400-W Hσ	duartz filter	nire water	N:0.7170		4.5 (360 nm)	119
	nolymerized complex method		450-W Ha	quartz filter	KOH	NiO /2186	NiO /1131	(in)	23 135 130
	r compres memor	1,	SI W-O	quantz miter	NOH	MO72160	MIO <sub>x</sub> /1131	0000	135, 133, 139
	eaction	7	450-w Hg	quartz niter	KDOH	NiOx/869	NiO <sub>x</sub> /430	5 (330 nm)	135
<sub>.5</sub> O <sub>10</sub>	eaction	4	450-W Hg	quartz filter	RbOH	$NiO_x/725$	$NiO_x/358$		135
RbLa <sub>2</sub> Ti <sub>2</sub> NbO <sub>10</sub> solid-state reaction	eaction 1	4	450-W Hg	quartz filter	RbOH	%OiN	NiO <sub>x</sub> /30		135
	eaction	7	450-W Hg	ouartz filter	pure water	NiO./700	NiO./340		135
, O.,	eaction	7	450-W Ho	martz filter	pure water	NiO /540	NiO. 765		135
	- I - I - I - I - I - I - I - I - I - I	7	450-W Ha	quartz filter	pure water	N:O /115	05/ OiN		135
	Caction		450 W US	quartz filter	pure muci	NiO /127	00%		301
	eaction	1 ,	81 W-05	quartz miter	pure water	MO/13/			56.
	eaction	4	450-W Hg	quartz filter	pure water	NiO <sub>x</sub> /442			125
			450-W Hg	quartz filter	pure water	NiO <sub>x</sub> /386			125
2		0.5	400-W Hg	quartz filter	pure water	NiO/714	NiO/358		133
	polymerized complex method	4	50-W Hg	quartz filter	pure water	$NiO_x/960$	$NiO_x/478$	27	22, 122—127, 131, 134
La,Ti,O,:Sr solid-state reaction	eaction 1	4	450-W Hg	quartz filter	pure water	NiO,/1510			125
			)			e			

Table 1. Continued									
						activity $(\mu \text{mol} \cdot \text{h}^{-1} \cdot \text{g}^{-1})$	$\operatorname{ol} \cdot \mathrm{h}^{-1} \cdot \mathrm{g}^{-1}$		
photocatalyst	synthetic method	mass (g)	light source	incident light	aqueous reaction solution	$co$ -catal./ $H_2$	co-catal./O <sub>2</sub>	QY (%)	reference
La <sub>2</sub> Ti <sub>2</sub> O <sub>7</sub> :Ca	solid-state reaction		450-W Hg	quartz filter	pure water	NiO <sub>x</sub> /850		4	125
La <sub>2</sub> Ti <sub>2</sub> O <sub>7</sub> :Ba	solid-state reaction		450-W Hg	quartz filter	NaOH	NiO <sub>x</sub> /~5000		20	125
M2 112O7	solid ctate reaction		450-w Hg	quartz niter quartz filter	pure water	NiO <sub>x</sub> /150			127
Nd, Ti,O,	solid-state reaction		450-W Hg	quartz filter	pure water	057, 257 NiO,/99			127
$NdLaTi_2O_7$	solid-state reaction	1	450-W Hg	quartz filter	pure water	NiO <sub>x</sub> /131			127
$\text{KaLaZr}_{0.3}\text{Ti}_{0.7}\text{O}_4$	two-step ion-exchange reaction	0.3	450-W Hg	quartz filter	pure water	NiO <sub>x</sub> /766.7	$NiO_x/386.7$	12.5	143
$\widetilde{\text{La}_{4}}\text{CaTi}_{5}\text{O}_{17}$	solid-state reaction	- 0	450-W Hg	quartz filter	pure water	NiO,/499		20 (<320 nm)	134
CaLa <sub>4</sub> T <sub>14</sub> O <sub>15</sub>	polymerized complex method	0.5	400-W Hg	quartz filter	pure water	NiO <sub>x</sub> /1186	NiO <sub>x</sub> /552		133
SrLa <sub>4</sub> I <sub>14</sub> O <sub>15</sub>	polymerized complex method	0.5 5.0	400-w Hg	quartz filter	pure water	NiO,/2342	NiO <sub>x</sub> /1092	0000	133
BaLa <sub>4</sub> T <sub>14</sub> O <sub>15</sub>	polymerized complex method	0.5	400-W Hg	quartz filter	pure water	NiO <sub>x</sub> /4600	NiO <sub>x</sub> /2308	15 (Z/0 nm)	133
Ba <sub>3</sub> LaNb <sub>3</sub> O <sub>12</sub>	polymerized complex method	0.5	400-W Hg	quartz filter	pure water	NiO <sub>x</sub> /2370	NiO <sub>x</sub> /11/6		133
Sr <sub>5</sub> Nb <sub>4</sub> O <sub>15</sub>	polymerized complex method	0.5 5.0	400-w Hg	quartz filter	pure water	NiO,/4400	NiO <sub>x</sub> /2200		133
KIIINBOs No Tii O	polymerized complex method	0.5 \$0.0	450-W Hg	quartz niter	pure water	NiO <sub>x</sub> /~60	NIO,/~20		142
Na21 6013	solid state meeting	0.23	400-w Ae	quartz filter	pure water	KuO <sub>2</sub> /29.2 D.::O /167	NuO2/14		101 - 104 $104 - 107$ 012 014
Da11409	solid-state reaction	0.2	400-w Ae	quartz filter	pure water	KUO2/10/	KUO2/ / 8		104 - 107, 913, 914
Gd2112O7	polymerized complex method	C.O S.O	400-w Ae	quartz iilter quartz filter	pure water	NiO <sub>x</sub> /800	NiO <sub>x</sub> /390	6 (313 nm)	131
12.112.07 Bi Ti O.	polymented complex memor		450 W Hg	qualitz ilitei Dygav filtar	CH-OH/A GNO.	D+/0 6	1410x/ - 1000	0 (313 11111)	176, 130, 131
BaBi Ti O.	solid-state reaction		450-W Hg	Dyrev filter	CH-OH/AgNO	Dr/8.7	2.		145
Bi-TiNbO	solid-state reaction		450-W Ho	Pyrex filter	CH <sub>2</sub> OH/A <sub>2</sub> NO <sub>3</sub>	Pt/33	31.		145
Na,Ti,O,	solid-state reaction	0.5	500-W Xe	They made	CH <sub>2</sub> OH	Pt/38	,		96
K,Ti,O	solid-state reaction	0.5	500-W Xe		CH,OH	Pt/69.4			96
H <sup>+</sup> -K <sub>2</sub> Ti,O <sub>5</sub>	solid-state reaction, H <sup>+</sup> -exchange	0.5	500-W Xe		CH3OH	Pt/220		10 (330–360 nm)	96
K,Ti,Oo	solid-state reaction	0.5	500-W Xe		CH <sub>2</sub> OH	Pt/9.6			96
$K_2Ti_6O_{13}$	solid-state reaction	0.5	500-W Xe		CH <sub>3</sub> OH	Pt/121.4			96
$\mathrm{Cs_2Ti_2O_5}$	solid-state reaction	1	400-W Xe	quartz filter	$CH_3OH/AgNO_3$	200	155		100
$\mathrm{H^{+}-Cs_{2}Ti_{2}O_{5}}$	solid-state reaction, H <sup>+</sup> -exchange	1	400-W Xe	quartz filter	$CH_3OH$	Pt/2512			100
$\mathrm{Cs_2Ti_5O_{11}}$	solid-state reaction		400-W Xe	quartz filter	$CH_3OH$	06			100
$Cs_2Ti_6O_{13}$	solid-state reaction		400-W Xe	quartz filter	CH <sub>3</sub> OH	38			100
H <sup>+</sup> -CsTiNbO <sub>5</sub>	solid-state reaction, H <sup>+</sup> -exchange		450-W Hg	Pyrex filter	CH <sub>3</sub> OH	Pt/320			44.
H - CS 112NBO7	solid-state reaction, H -exchange	<b>-</b> -	450-W Hg	Pyrex filter	CH <sub>3</sub> OH	PU/8/			144
NB13P0115O16 SiO - V Ti O	polymerized complex method	100	450 W Hg	quartz niter Dygay filtar	Cer /Cer	33.21 D+6.2600	1.32		140
SiO <sub>2</sub> -K <sub>2</sub> II <sub>4</sub> O <sub>9</sub> SiO <sub>2</sub> -K <sub>2</sub> Ti <sub>2</sub> -Mn <sub>0</sub> ,O <sub>0</sub>	solid-state reaction stepwise ion-exchange	0.2	450-W Hg	r yrex filter	CH3OH	~1600			86
Tio 2 Tro 2 PO4	sol—gel method	0.5	300-W Xe	>290 nm	Na <sub>2</sub> CO <sub>3</sub>	Pt/17.7			155
$Zr_2(PO_4)PVX_3$	hydrothermal method	0.025	200-W Hg-Xe	<320 nm	EDTA	Pt/267.8		4 (<320 nm)	156
$CaTi_{0.93}Zr_{0.07}O_3$	polymerized complex method	0.1	500-W Hg	quartz filter	pure water	Pt/1400	!	1.91 (365 nm)	121
$ZrO_2$	Soekawa Chemicals		400-W Hg	quartz filter	NaHCO <sub>3</sub>	309	167		78, 147–150
Zr-MCM-41 B°ZrO	nydrotnermal memod Dachini tyna process	0.1	400-w Hg	quartz iilter quartz filter	pure water	\$27.5 \$20.5	131	7.7	151
BaZrozSno.O.	Pechini-type process	0.2	400-W Hs	quartz filter	pure water	069	185		153
$La_2Zr_2O_7$	solid-state reaction	!	500-W Xe	quartz filter	pure water	0.91"			154
$Nb_2O_5$	evaporation-induced self-assembly method	0.1	400-W Hg	quartz filter	CH <sub>3</sub> OH	Pt/12315			158
Nb/In mixed oxide	evaporation-induced self-assembly method	0.2	$400-W H^e$	quartz filter	$CH_3OH$	159.3			159
K4Nb <sub>6</sub> O <sub>17</sub>	solid-state reaction	0.5	450-W Hg	quartz filter	pure water	NiO <sub>x</sub> /3674	NiO,/1700	5.3 (330 nm)	157, 160 - 171
Kb4Nb6O17	solid-state reaction		400-W Hg	quartz filter	pure water	NiO <sub>x</sub> /936	NiO <sub>x</sub> /451	10 (330 nm)	157, 171
K2Kb2lNb <sub>6</sub> O <sub>17</sub>	solid-state reaction	<b>-</b> -	450-W Hg	quartz niter	pure water	NiO <sub>x</sub> //8	NiO <sub>2</sub> /39	5.5	1/1
NaMbO	solid-state feacuoli polymerized complex method	10	400-W Hg	quartz filter quartz filter	pure water	NIO <sub>x</sub> /409	NIO <sub>2</sub> /198		127
KNPO.	potynicated complex intentor hydrothermal method	0.7	400-W Ho	quartz filter	puic water methanol	Pt/5170	NuO2/100		173
Ca <sub>2</sub> Nb <sub>2</sub> O <sub>7</sub>	solid-state reaction		450-W Hg	quartz filter	pure water	NiO,/101		7 (<288 nm)	134
$Sr_2Nb_2O_7$	hydrothermal method	0.2	400-W Hg	quartz filter	pure water	RuO <sub>2</sub> /475	$RuO_2/220$	32 (300 nm)	21, 177, 222, 223
SrNb <sub>2</sub> O <sub>6</sub>	hydrothermal method	0.2	400-W Hg	quartz filter	pure water	RuO <sub>2</sub> /435	$RuO_2/205$	19 (300 nm)	177
BiNbO <sub>4</sub>	solid-state reaction	п,	400-W Hg	quartz filter	pure water	∞ ₹			200
B11a <sub>0.8</sub> INb <sub>0.2</sub> O <sub>4</sub>	solid-state reaction	<b>-</b> -	400-w Hg	quartz filter	pure water	41 N:O //OO	0217		200
Sr <sub>2</sub> I a <sub>0.5</sub> IND <sub>1.5</sub> O <sub>7</sub>	solid-state reaction	1	400-w rig	quartz niter	pure water	INIO <sub>x</sub> / 200	NIO <sub>x</sub> /120		577

Table 1. Continued									
						activity $(\mu \text{mol} \cdot \text{h}^{-1} \cdot \text{g}^{-1})$	$\operatorname{sl} \cdot h^{-1} \cdot \operatorname{g}^{-1}$		
photocatalyst	synthetic method	mass (g)	light source	incident light	aqueous reaction solution	co-catal./H <sub>2</sub>	co-catal./O <sub>2</sub>	QY (%)	reference
BasNb4O15	polymerized complex method	0.5	400-W Hg	quartz filter	pure water	NiO./8042	NiO./3922	17 (270 nm)	133, 960
$ZnNb_2O_6$	solid-state reaction	1	450-W Hg	-	pure water	NiO <sub>x</sub> /54	$NiO_x/21$	,	194
NaCa2Nb3O10	solid-state reaction	0.3	450-W Hg	quartz filter	pure water	RuO <sub>2</sub> /393.3	RuO <sub>2</sub> /186.7		187
TBA <sub>x</sub> [H <sub>1-x</sub> Ca <sub>2</sub> Nb <sub>3</sub> O <sub>10</sub> ] nanosheet	solid-state reaction, ion-exchange	0.1	750-W Hg	290—600 nm	pure water	Pt/784	0000	7.48	188
CS2NB4O11 1 : NFO	solid-state reaction	5.0	400-W Hg	quartz filter	CSOH	NiO <sub>x</sub> /3400	NiO,/1600		1/5
La31NbO <sub>7</sub>	polymenzed complex memod	0.5 5.0	400-w Hg	quartz mier	pure water	MIO <sub>x</sub> / /0	52 6		151, 192
Eal/3/NO3 RhNhWO	solid-state reaction	0.3	1000-W Ho	quartz filter	RhOH	NiO./38	NiO./14.3		176
CSNbWO	solid-state reaction	0.3	1000-W Hg	quartz filter	CsOH	NiO./33.7	NiO./15.7		176
BaZn, aNb. o.	solid-state reaction	0.5	400-W Hg	quartz filter	pure water	NiO./291.2	NiO./145.6		178, 179
In-BiNbO	solid-state reaction		400-W Hg	quartz filter	pure water	54.3	26.6		196
Ga,BiNbO,	solid-state reaction		400-W Hg	quartz filter	pure water	72.6	35.7		196
$\mathrm{Bi}_2\mathrm{ar{Y}NbO}_7$	solid-state reaction	1	400-W Hg	quartz filter	pure water	21			199
$\mathrm{Bi}_2\mathrm{CeNbO}_7$	solid-state reaction	1	400-W Hg	quartz filter	pure water	13			199
Bi <sub>2</sub> GdNbO <sub>7</sub>	solid-state reaction	_	400-W Hg	quartz filter	pure water	10			199
$\mathrm{Bi}_2\mathrm{SmNbO}_7$	solid-state reaction	_	400-W Hg	quartz filter	pure water	2			199
$\mathrm{Bi}_2\mathrm{NdNbO}_7$	solid-state reaction	_	400-W Hg	quartz filter	pure water	4			199
$\mathrm{Bi}_{2}\mathrm{PrNbO}_{7}$	solid-state reaction	_	400-W Hg	quartz filter	pure water	2.1			199
Bi <sub>2</sub> LaNbO <sub>7</sub>	solid-state reaction		400-W Hg	quartz filter	pure water	1.8 1.8	1		199
Bi <sub>2</sub> AINbO <sub>7</sub>	solid-state reaction	_ ;	400-W Hg	quartz filter	$CH_3OH/Ce(SO_4)_2$	Pt//10	25		197
Bi <sub>2</sub> AINbO <sub>7</sub> :La	solid-state reaction	0.1	350-W Hg	quartz filter	CH <sub>3</sub> OH	NiO,/1414	Ç		198
Bi2GaNbO7 B: T-NH-O	solid-state reaction		400-W Hg	quartz niter	CH <sub>3</sub> OH/Ce(SO <sub>4</sub> ) <sub>2</sub>	PU/	10		197
Di2IIINDO7	Solid-state reaction	100	250 W Hg	quartz miter	CH3OH/Ce(SO4)2	580	105		197
Cabi <sub>2</sub> 1N <sub>2</sub> O <sub>9</sub>	solid state reaction	2.0	250 W Hg	quartz iiiter	CH3OH/AgNO3	3660	103		193
SIDI2INU2O9 Bob: NIP O	solid state reaction	7.0	350 W Hg	quartz filter	CH OH A AND	3000	305		193
H++KI aNb-O-	solid-state reaction H <sup>+</sup> -exchange		450-W Hg	quartz filter	CH,OH/A oNO,	Pt/3800	46		38
$H^+$ —RhI aNh $O_r$	solid-state reaction H <sup>+</sup> -exchange	· -	450-W Ho	quartz filter	CH, OH/A oNO	Pt/2600	2 6		38.0
H+-Cst aNb-0	solid-state reaction H <sup>+</sup> -exchange		450-W Ho	quartz filter	CH,OH/A NO,	Pt/2200	1 cc		38.0
$H^+$ – $KCa_2Nb_3O_{10}$	solid-state reaction. H <sup>+</sup> -exchange		450-W Hg	quartz filter	CH <sub>3</sub> OH/A <sub>2</sub> NO <sub>3</sub>	Pt/19 000	, oc		38
H+-RbCa2Nb2O10	solid-state reaction, H <sup>+</sup> -exchange	-	450-W Hg	quartz filter	CH <sub>3</sub> OH/AgNO <sub>3</sub>	Pt/17000	16		38
H+-CsCa <sub>2</sub> Nb <sub>3</sub> O <sub>10</sub>	solid-state reaction, H <sup>+</sup> -exchange	-	450-W Hg	quartz filter	CH3OH/AgNO3	Pt/8300	10		38
$\mathrm{H^{+}-KSr_{2}Nb_{3}O_{10}}$	solid-state reaction, H <sup>+</sup> -exchange	1	450-W Hg	quartz filter	CH <sub>3</sub> OH/AgNO <sub>3</sub>	Pt/43 000	30		38
$\mathrm{H^{+}-KCa_{2}NaNb_{4}O_{13}}$	solid-state reaction, H <sup>+</sup> -exchange	-	450-W Hg	quartz filter	CH <sub>3</sub> OH/AgNO <sub>3</sub>	Pt/18 000	39		38
$H_{2.33}Sr_{0.67}Nb_5O_{14.335}$	solid-state reaction, stepwise ion-exchange	_	100-W Hg	Pyrex filter	$CH_3OH$	Pt/6834.4			181
$H_2Ca_4Nb_6O_{20}$	solid-state reaction		100-W Hg	Pyrex filter	CH <sub>3</sub> OH	928			
$SiO_2 - HCa_2Nb_3O_{10}$	solid-state reaction, stepwise ion-exchange	<del>-</del> -	450-W Hg	Pyrex filter	CH3OH	Pt/10800			189, 190
N3+-C32N3-010	solid-state reaction, stepwise ion-exchange	0.1	500-W Xe	Pyrex iiller Pyrex filter	CH <sub>3</sub> OH	Pt/6200			186, 187
HCa2Nb2OtoT.a	solid-state reaction		100-W Ho	Pyrex filter	CH3OH	Pt/7544			182
H <sub>2</sub> LaNb <sub>2</sub> O <sub>7</sub> :In	solid-state reaction	· —	100-W Hg	>290 nm	CH <sub>3</sub> OH	Pt/5268		1.54 (>290 nm)	183
$H_2^{\perp}LaNb_2^{\perp}O_7:Mo$	solid-state reaction	П	100-W Hg	>290 nm	CH <sub>3</sub> OH	Pt/3586			184
$Ta_2O_5$	solid-state reaction	-	400-W Hg	quartz filter	pure water	$NiO_x/1154$	$NiO_x/529$		78, 157, 201, 202
Mg-Ta oxide	sol-gel method	0.3	450-W Hg		pure water	NiO/340	NiO/170		204
$Ta_2O_5$ :Ni	ligand-assisted templating method	0.35	450-W Hg		pure water	194.3	9.801		205
$K_2$ Pr $Ta_5$ O <sub>15</sub>	solid-state reaction		400-W Hg	quartz filter	pure water	NiO/1550	NiO/830		40, 235
K <sub>2</sub> SmTa <sub>5</sub> O <sub>15</sub>	solid-state reaction	_ ,	400-W Hg	quartz filter	pure water	NiO/623	NiO/272		235
K <sub>2</sub> DyTa <sub>5</sub> O <sub>15</sub>	solid-state reaction	<b>-</b>	400-W Hg	quartz filter	pure water	NiO/358	NiO/1/2		235
K <sub>2</sub> ImTa <sub>5</sub> O <sub>15</sub>	solid-state reaction	<b>-</b>	400-W Hg	quartz filter	pure water	NiO/1/3	N10//0		235
K3 I 43 S12 O 13	solid ctota magation	1 5 0	400-W Hg	quartz niter	pure water	MIO/308	MIO/188	( 5 ( 75 / nm)	738
N3143D2O12	solid-state reaction	 -	450-W Hg	qualiz mier	pure water	4780	2420	0.5 (254 11111)	201 206
NaTaO,	bydrothermal method	0.3	450-W Hg	quartz filter	pure water	NiO/29 333	NiO/14 667		201, 200
KTaO	solid-state reaction	0.5	400-W Ho	quartz filter	pure water	An/116	Au/50		157, 167, 201, 206
$AgTaO_3$	solid-state reaction	0.15	300-W Xe	Pyrex filter	pure water	NiO/138	NiO/63.3		242
KTaO <sub>3</sub> :Zr	solid-state reaction	0.1	500-W Xe	Pyrex filter	pure water	$NiO_x/935$	NiO <sub>x</sub> /421		212, 213
KTaO <sub>3</sub> :Ga	solid-state reaction	0.1	500-W Xe	Pyrex filter	pure water	$NiO_x/677$	NiO <sub>x</sub> /223		213

Table 1. Continued						activity (umol·b <sup>-1</sup> •g	01.4-1.0-1)		
					adueous reaction	Car non	(a )		
photocatalyst	synthetic method	mass (g)	light source	incident light	solution	co-catal./H2	co-catal./O <sub>2</sub>	QY (%)	reference
KTaO <sub>3</sub> :Hf	solid-state reaction	0.1	500-W Xe	Pyrex filter	pure water	NiO,/985	NiO,/398	1	213
NaTaO3:La NaTaO;Sr	solid-state reaction	1 0.5	400-w Hg 400-w Ho	quartz niter quartz filter	pure water	NiO/19 800 NiO/19 000	NiO/9660 NiO/9400	26 (2/0 nm)	20, 214
NaTaO <sub>3</sub> :Ba	solid-state reaction	0.5	400-W Hg	quartz filter	pure water	NiO/18 600	NiO/9380		215
NaTaO <sub>3</sub> :Pr	solid-state reaction		400-W Hg	quartz filter	pure water	NiO <sub>3</sub> /5290	NiO,/2580		208, 214
NaTaO3:Nu NaTaO3:Sm	solid-state reaction		400-w ng 400-W Hg	quartz filter	pure water	NiO,/5290	NiO,/2630		208, 214 208, 214
Na <sub>2</sub> Ta <sub>2</sub> O <sub>6</sub>	hydrothermal method	0.5	450-W Hg	quartz filter	NaOH	NiO/782	NiO/390		
$K_2Ta_2O_6$	hydrothermal method	0.5	450-W Hg	quartz filter	NaOH	NiO/974	NiO/452		218
CaTa <sub>2</sub> O <sub>2</sub>	solid-state reaction		400-w Hg 400-W Hs	quartz filter	pure water NaOH	4 NiO/72	NiO/32		208, 220
SrTa <sub>2</sub> O <sub>6</sub>	solid-state reaction		400-W Hg	quartz filter	pure water	096/O!N	NiO/490	7 (270 nm)	208, 220
$BaTa_2O_6$	solid-state reaction		400-W Hg	quartz filter	pure water	NiO/629	NiO/303	,	201, 208, 220
$ZnTa_2O_6$	solid-state reaction		400-W Hg	quartz filter	pure water	NiO/15	NiO/6		201
Rb, Ta, O, 7	solid-state reaction		400-w Hg 400-W Hg	quartz filter	pure water	NiO./92	4 NiO./46		157
$Ca_2Ta_2O_7$	hydrothermal method	0.5	450-W Hg	quartz filter	NaOH	NiO/340	NiO/166		218
$Sr_2Ta_2O_7$	polymerized complex method	0.7	400-W Hg	quartz filter	pure water	NiO/5024	NiO/2476	24 (270 nm)	21,221-223
$K_2SrTa_2O_7$	solid-state reaction	0.5	400-W Hg	quartz filter	pure water	748	384		226
Rb-SrT2-O-	solid-state reaction	0.5	400-w Hg 400-W Hg	quantz miter	pure water	176	22.8		226
RbNdTa <sub>2</sub> O <sub>7</sub>	solid-state reaction	0.2	400-W Hg	quartz filter	pure water	NiO <sub>x</sub> /235	NiO <sub>3</sub> /126.5		231-233
RbLaTa <sub>2</sub> O <sub>7</sub>	solid-state reaction	0.2	400-W Hg	quartz filter	pure water	NiO <sub>x</sub> /6	NiO <sub>x</sub> /3		231—233
RbSmTa <sub>2</sub> O <sub>7</sub>	solid-state reaction	0.2	400-W Hg	quartz filter	pure water	NiO <sub>x</sub> /53	NiO <sub>x</sub> /28.5		231-233
$H_2La_{2/3}1a_2O_7$ $K_2I_{35,5}Ta_2O_7$	solid-state reaction, H '-exchange	C.O S.O	400-w Hg 400-w Hg	quartz niter	pure water	NiO/1880 NiO/292	NiO/918		177
K2Sr15Ta2O10	solid-state reaction	0.5	400-W Hg	Pyrex filter	pure water	RuO,/200	RuO./78.8	2 (252.5 nm)	228
$\mathrm{LiCa}_2\mathrm{Ta}_3\mathrm{O}_{10}$	solid-state reaction, hydrothermal treatment	0.2	400-W Hg	quartz filter	pure water	NiO <sub>x</sub> /3540	NiO <sub>x</sub> /1665		229
$NaCa_2Ta_3O_{10}$	solid-state reaction	0.5	400-W Hg	quartz filter	pure water	NiO,/1465 NiO /885	NiO,/810		229
RbCa2Ta3O10	solid-state reaction	0.2	400-W Hg	quantz filter	pure water	006/0IN	NiO./455		229
$CsCa_2Ta_3O_{10}$	solid-state reaction	0.2	400-W Hg	quartz filter	pure water	NiO,/445	NiO,,/265		229
$KBa_2Ta_3O_{10}$	solid-state reaction	0	450-W Hg	quartz filter	pure water	NiO,/150	1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	8 (<350 nm)	134
Sr.Ta.O.	polymerized complex method nolymerized complex method	0.5	400-w Hg 400-W Ho	quartz niter quartz filter	pure water	NiO/64	NiO/1444		221
Ba <sub>5</sub> Ta <sub>4</sub> O <sub>15</sub>	polymerized complex method	. –	450-W Hg	quartz filter	pure water	NiO/7110	NiO/3621		224
H <sub>1.8</sub> Sr <sub>0.81</sub> Bi <sub>0.19</sub> Ta <sub>2</sub> O <sub>7</sub>	solid-state reaction, acid treatment	0.1	350-W Hg	quartz filter	pure water	2460	1110		230
Mg−Ta oxide	sol—gel method	0.3	450-W Hg	quartz filter	pure water	NiO <sub>x</sub> /340 NiO /578	NiO,/170 NiO,/257 5		204 236
PrTaO <sub>4</sub>	solid-state reaction	0.2	400-W Hg	quartz filter	pure water	NiO,/23.5	NiO,/23/		236
$La_3TaO_7$	polymerized complex method	0.5	400-W Hg	quartz filter	pure water	NiO,/328	NiO,/160		131, 192
La <sub>1/3</sub> TaO <sub>3</sub>	solid-state reaction	0.5	400-W Hg	quartz filter	pure water	NiO//0	NiO/15.8		177
CsTaWO	solid-state reaction	0.3	1000-W Hg	quantz filter	CsOH	NiO./65.7	NiO./29.3		176
$Bi_2LaTaO_7$	solid-state reaction	-	400-W Hg	quartz filter	pure water	41.8	20.5		239
$Bi_2YTaO_7$	solid-state reaction		400-W Hg	quartz filter	pure water	33.6	16		239
Sn-Ta <sub>2</sub> O <sub>7</sub>	solid-state reaction	0.3	300-W Xe	quanz miter Pvrex filter	CH3OH	Pt/7			240 243
BaNi <sub>1/3</sub> Ta <sub>2/3</sub> O <sub>3</sub>	solid-state reaction	0.5	400-W Hg	quartz filter	CH <sub>3</sub> OH	Pt/84.4			178, 225
BaZn <sub>1/3</sub> Ta <sub>2/3</sub> O <sub>3</sub>	solid-state reaction	0.5	400-W Hg	quartz filter	CH <sub>3</sub> OH	Pt/389.6	300		178
Cabi <sub>2</sub> 1a <sub>2</sub> O <sub>9</sub> SrBi <sub>3</sub> Ta <sub>2</sub> O <sub>9</sub>	solid-state reaction	0.2	350-W Hg	quartz filter	CH3OH/AgNO3	11 300	009		241 241
BaBi <sub>2</sub> Ta <sub>2</sub> O <sub>9</sub>	solid-state reaction	0.2	350-W Hg	quartz filter	CH3OH/AgNO3	550	250		241
PbWO <sub>4</sub>	solid-state reaction	0.25	200-W Hg-Xe	quartz filter	pure water	$RuO_2/96$	$RuO_2/48$		244, 245
Na2 w 4O13 Bi2W2O	solid-state reaction		450-w Hg	Pyrex filter	CH <sub>2</sub> OH/AgNO <sub>3</sub>	PV18	218		145
PbMoO <sub>4</sub>	Morton Thiokol Inc.	$\hat{0.3}$	200-W Hg	Pyrex filter	CH3OH/AgNO3	Pt/98.3	161.7		246
(NaBi) <sub>0.5</sub> MoO <sub>4</sub>	solid-state reaction	0.3	300-W Xe	Pyrex filter	CH <sub>3</sub> OH/AgNO <sub>3</sub>	Pt/2	193.3		248
$(Agg_1)_{0.5}MoO_4  (NaBi)_{0.5}WO_4$	sond-state reaction solid-state reaction	0.3	300-W Xe	Pyrex filter	CH3OH/AgNO3	FV0 PV23.3	53.7 4.3		248 248
(AgBi) <sub>0.5</sub> WO <sub>4</sub>	solid-state reaction	0.3	300-W Xe	Pyrex filter	CH3OH/AgNO3	Pt/3.3	19.3		248
B <sub>12</sub> MoO <sub>6</sub>	solid-state reaction air annealing	- × O	450-W Hg 550-W Xe	Pyrex filter	CH3OH/AgNO3	Pt/0.01	2.1		145 269
CeO <sub>2</sub> :Sr	solid-state reaction	1.1	450-W Hg	quartz filter	pure water	$\text{RuO}_2/{\sim}100$	$RuO_2/\sim 50$		270

Table 1. Continued

						activity (µm	activity $(\mu \text{mol} \cdot \text{h}^{-1} \cdot \text{g}^{-1})$		
photocatalyst	synthetic method	mass (g)	light source	incident light	aqueous reaction solution	co-catal./H2	co-catal./02	OY (%)	reference
Ce-MCM-41	impregnation method	0.05	400-W Ho	guartz filter	nure water	70 800	15 250		271
BaCeO	polymerized complex method	0.2	400-W Hg	in the second	pure water	RuO <sub>2</sub> /295	RuO <sub>2</sub> /130		272
$^{\circ}$ $^{\circ}$ $^{\circ}$	electrodeposition mehtod		500-W Hg	quartz filter	ethanol	$20^{a}$	1	38.7	250
$NaInO_2$	solid-state reaction	0.25	200-W Hg-Xe	quartz filter	pure water	$RuO_2/\sim 3.6$	$RuO_2/\sim 1.2$		258, 260
$CaIn_2O_4$	precipitation, calcination	0.25	200-W Hg-Xe	quartz filter	pure water	$RuO_2/\sim 84$	$\text{RuO}_2/\sim 40$		259-262
$SrIn_2O_4$	precipitation, calcination	0.25	200-W Hg-Xe	quartz filter	pure water	$RuO_2/\sim 28$	$RuO_2/\sim 12$		259-262
$\mathrm{Sr}_{0.93}\mathrm{Ba}_{0.07}\mathrm{InO}_4$	precipitation, calcination	0.25		quartz filter	pure water	$RuO_2/\sim 28$	$RuO_2/\sim 12$		260
$LaInO_3$	solid-state reaction	0.25	200-W Hg-Xe	quartz filter	pure water	$RuO_2/\sim 4.8$	$RuO_2/\sim 2$		259
$Y_x In_{2-x}O_3$	solid-state reaction	0.25		quartz filter	pure water	$RuO_2/\sim 32$	$RuO_2/\sim 14$		257
$NaSbO_3$	solid-state reaction	0.25		quartz filter	pure water	$RuO_2/\sim 26$	$RuO_2/\sim 8$		261, 263
$CaSb_2O_6$	solid-state reaction	0.25		quartz filter	pure water	$RuO_2/\sim 5.6$	$RuO_2/\sim 0.8$		263
$Ca_2Sb_2O_7$	solid-state reaction	0.25		quartz filter	pure water	$RuO_2/\sim 12$	$RuO_2/\sim 4$		263
$Sr_2Sb_2O_7$	solid-state reaction	0.25		quartz filter	pure water	$RuO_2/\sim 30.4$	$RuO_2/\sim 12$		263
$\mathrm{Sr}_2\mathrm{SnO}_4$	solid-state reaction	0.25	200-W Hg-Xe	quartz filter	pure water	$RuO_2/\sim 16$	$RuO_2/\sim6.8$		261
$SrSnO_3$	hydrothermal, calcination	0.2	400-W Hg	quartz filter	CH <sub>3</sub> OH/AgNO <sub>3</sub>	Pt/8200	2500		268
$ZnGa_2O_4$	solid-state reaction	0.25	200-W Hg-Xe	quartz filter	pure water	$RuO_2/\sim 48$	$RuO_2/\sim 16$		264
$\mathrm{Zn_2GeO_4}$	solid-state reaction	0.25	200-W Hg-Xe	quartz filter	pure water	$RuO_2/\sim 84$	$RuO_2/\sim 40$		265
$LiInGeO_4$	solid-state reaction	0.25	200-W Hg-Xe	quartz filter	pure water	$RuO_2/\sim 104$	$RuO_2/\sim 52$		596
$Ga_{1.14}In_{0.86}O_3$	solid-state reaction		450-W Hg	>300 nm	CH <sub>3</sub> OH/AgNO <sub>3</sub>	$Pt/\sim 30^a$	$\sim$ 25 $^a$		256
$In_2O_3$	Kanto Chemicals		450-W Hg	>300 nm	CH <sub>3</sub> OH/AgNO <sub>3</sub>	$Pt/0^a$	$\sim\!\!09$		256
$\beta$ -Ga <sub>2</sub> O <sub>3</sub>	rare metallic		300-W Xe	quartz filter	CH <sub>3</sub> OH/AgNO <sub>3</sub>	$Pt/\sim 50^a$	$\sim$ 7 $^a$		256
$Ga_2O_3$	high-purity chemical	_	450-W Hg	quartz filter	pure water	Ni/338	Ni/171		253, 254
$Ga_2O_3$ :Zn	impregnation, calcination	_	450-W Hg	quartz filter	pure water	Ni/4100	Ni/2200	$\sim 20$	254
$Ga_2O_3:Ca$	impregnation, calcination	_	450-W Hg	quartz filter	pure water	Ni/950	Ni/550		254
$Ga_2O_3$ :Sr	impregnation, calcination	_	450-W Hg	quartz filter	pure water	Ni/640	Ni/282		254
$Ga_2O_3$ :Ba	impregnation, calcination	_	450-W Hg	quartz filter	pure water	Ni/765	Ni/348		254
$Ga_2O_3$ : $Ta$	impregnation, calcination		450-W Hg	quartz filter	pure water	Ni/534	Ni/256		254
$\mathrm{Bi_2GaVO}_{\gamma}$	solid-state reaction	_	400-W Hg	quartz filter	pure water	46.8	22.9		251
$\mathrm{Bi_2YVO_8}$	solid-state reaction		400-W Hg	quartz filter	pure water	40.7	20.7	0.628 (>390 nm)	252
$Zn-Lu_2O_3/Ga_2O_3$	solid-state reaction	0.5	400-W Hg	quartz filter	pure water	$NiO_x/100.4$	NiO <sub>x</sub> /53.4	6.81 (320 nm)	255
$Ge_3N_4$	NH <sub>3</sub> nitridation	0.5	450-W Xe	>200 nm	$\mathrm{H}_2\mathrm{SO}_4$	$RuO_2/\sim 3600$	$RuO_2/\sim 1600$	9 (300 nm)	24, 277 - 279
GaN	NH <sub>3</sub> nitridation	0.3		>300 nm	$\mathrm{H}_2\mathrm{SO}_4$	$Rh_{2-x}Cr_{x}O_{3}/63.3$	$Rh_{2-x}Cr_xO_3/31.7$	0.7 (300 - 340  nm)	275
GaN:Mg	NH <sub>3</sub> nitridation	8.0	450-W Hg	Pyrex filter	pure water	$RuO_2/\sim750$	$RuO_2/\sim 362.5$		276, 759
GaN:Zn	NH <sub>3</sub> nitridation	0.8	450-W Hg	Pyrex filter	pure water	$RuO_2/\sim 250$	$RuO_2/\sim 125$		276, 759
GaN:Be	NH <sub>3</sub> nitridation	8.0		Pyrex filter	pure water	$RuO_2/\sim 587.5$	$RuO_2/\sim 287.5$		759
ZuZ	precipitation	0.4 6.0	200-W Hg	quartz filter	$Na_2SO_3$	Pt/34 037.5		90 (313 nm)	273, 364
InP	Kojundo Chemical Laboratory	0.2	250-W Hg	quartz filter	Na <sub>2</sub> S	PV100			274
AgBr/SiO <sub>2</sub>	Schumann emulsion	0.5	100-W Hg	quartz filter	methanol	650	17.1		280
H4SI W 12 O40/SIO2	impregnation memod	-	430-w rig	quartz mier	pure water	97.0	10.4		749

<sup>a</sup> The unit of H<sub>2</sub>/O<sub>2</sub> evolution rate is  $\mu$ mol·h<sup>-1</sup>, as the weight of photocatalyst has not been given in the reference. <sup>b</sup> Mercury lamp. <sup>c</sup> Xenon lamp. <sup>d</sup> Fluorescent lamp. <sup>e</sup> Metal halide lamp.

with HCl.<sup>70</sup> The addition of either NaOH or Na<sub>2</sub>CO<sub>3</sub> was found to be effective for water splitting using the Pt/TiO<sub>2</sub> photocatalyst. 71,75,78 When TiO<sub>2</sub> was doped with metal ions, the photocatalytic activity for water splitting was effectively enhanced. Chae et al. reported that, whereas Ga doped TiO<sub>2</sub> powder could split pure water stoichiometrically under UV irradiation, pure TiO<sub>2</sub> did not show any activity.<sup>79</sup> The Ni<sup>2+</sup> doping enhanced the photoactivity of the TiO<sub>2</sub> for hydrogen production from an aqueous methanol solution.80 Sn/Eu codoped TiO<sub>2</sub> exhibited a high activity for hydrogen generation with a quantum efficiency of  $\sim 40\%$  with Pd as the cocatalyst under the irradiation from a fluorescent lamp.<sup>81</sup> Zalas and La studied the effect of lanthanide doping on the photocatalytic activity of TiO<sub>2</sub>. 82 The best performance for hydrogen production from an aqueous methanol solution was obtained for the TiO<sub>2</sub> containing 0.5 mol % of Gd oxide as the dopant. The UV-driven photocatalytic activity of TiO<sub>2</sub> was also improved by combining it with a second oxide semiconductor. All of the mixed oxides with heterophasestructures, SnO<sub>2</sub>/TiO<sub>2</sub>, <sup>83</sup> ZrO<sub>2</sub>/TiO<sub>2</sub>, <sup>84</sup> Cu<sub>x</sub>O/TiO<sub>2</sub>, <sup>85-87</sup> Ag<sub>x</sub>O/  $TiO_2$ , <sup>88-90</sup> and  $MTiO_3/TiO_2$  (M = Ca, Sr, Ba), <sup>91</sup> displayed higher rates of photocatalytic hydrogen evolution from aqueous solutions containing electron donors other than TiO<sub>2</sub> alone. With Pt as a cocatalyst, the Ti/B binary oxide stoichiometrically decomposes pure water under UV irradiation. 92,93 When TiO<sub>2</sub> nanoclusters were dispersed in the mesoporous structures of MCM-41 and MCM-48, the formed Ti-MCM-4194 and Ti-MCM-4895 showed much higher photocatalytic activity for hydrogen evolution under UV irradiation than bulk TiO<sub>2</sub>.

Many white titanates are known to work as efficient photocatalysts for water splitting under UV irradiation. Shibata et al. reported that the layered titanates, Na<sub>2</sub>Ti<sub>3</sub>O<sub>7</sub>, K<sub>2</sub>Ti<sub>2</sub>O<sub>5</sub>, and K<sub>2</sub>Ti<sub>4</sub>O<sub>9</sub>, were active in photocatalytic H<sub>2</sub> evolution from aqueous methanol solutions even without the presence of Pt cocatalyst. 96 These layered titanates, consisting of titanium oxide layers and interlayers, can be modified using ion-exchange reactions. 96,97 Of the materials studied, the H<sup>+</sup>-exchanged K<sub>2</sub>Ti<sub>2</sub>O<sub>5</sub> exhibited a high activity with a quantum yield of up to ca. 10%. After being pillared with SiO<sub>2</sub> in the interlayers, K<sub>2</sub>Ti<sub>4</sub>O<sub>9</sub> showed an enhanced photocatalytic acitivity for H<sub>2</sub> evolution from CH<sub>3</sub>OH/H<sub>2</sub>O mixtures. This is in agreement with an increase in the surface area. 98 The Na<sub>2</sub>Ti<sub>2</sub>O<sub>5</sub> titanate nanotube material with a nickel complex intercalated into the interlayers also showed a high photocatalytic activity for H<sub>2</sub> evolution from water/methanol solutions under UV irradiation.<sup>99</sup> Kudo and Kondo found that a range of Cesium compounds,  $Cs_2Ti_nO_{2n+1}$  (n = 2, 5,6), with layered structures showed photocatalytic activities for H<sub>2</sub> and O<sub>2</sub> evolution from aqueous solutions. <sup>100</sup> The Cs<sub>2</sub>Ti<sub>2</sub>O<sub>5</sub> with a five-coordinate structure consisting of TiO<sub>5</sub> units was more active than Cs<sub>2</sub>Ti<sub>5</sub>O<sub>11</sub> and Cs<sub>2</sub>Ti<sub>6</sub>O<sub>13</sub> with six-coordinate structures consisting of TiO<sub>6</sub> units. The unsaturated coordination state of the five-coordinate structure worked as the active sites of catalytic reactions and contributed to the photoactivity. The photoactivity of Cs<sub>2</sub>Ti<sub>2</sub>O<sub>5</sub> was also greatly enhanced by the H<sup>+</sup>-exchange reaction. Inoue and co-workers investigated a series of alkali-metal titanates with a chemical formula of  $M_2Ti_nO_{2n+1}$  (M = Na, K, Rb and n = 2, 3, 4, 6) as photochemical water-splitting catalysts. 101-104 The alkaline metal atoms, M, in M<sub>2</sub>Ti<sub>6</sub>O<sub>13</sub> showed a great effect on the photocatalytic activity for water splitting. Interestingly,  $RuO_2/M_2Ti_6O_{13}$  (M = Na, K, and Rb) with rectangular tunnel structures showed higher photocatalytic activity than RuO<sub>2</sub>/Cs<sub>2</sub>Ti<sub>6</sub>O<sub>13</sub> with a layered structure. The activity increased in the order of Na > K > Rb > Cs. Kohno et al. reported that, in a photocatalytic system of ruthenium oxide-deposited barium titanates (BaTi<sub>4</sub>O<sub>9</sub>, Ba<sub>2</sub>Ti<sub>9</sub>O<sub>20</sub>, Ba<sub>4</sub>Ti<sub>13</sub>O<sub>30</sub>, and Ba<sub>6</sub>Ti<sub>17</sub>O<sub>40</sub>), only RuO<sub>2</sub>/BaTi<sub>4</sub>O<sub>9</sub> was active in water decomposition. 105 The pentagonal prism tunnel structure of RuO2/BaTi4O9 gave rise to a higher photocatalytic activity than RuO<sub>2</sub>/K<sub>2</sub>Ti<sub>4</sub>O<sub>9</sub> with a zigzag layer structure. It is believed that the tunnel structure was responsible for the high dispersion of the  $RuO_2\, particles. ^{104,106,107}$ 

The method of catalyst preparation also appears to play a role in the final activity. For example, BaTiO<sub>3</sub> synthesized with a polymerized complex method exhibited enhanced photocatalytic activity when compared to the materials prepared by traditional solid-state reaction method. 108 This was due to the larger surface area. Domen and co-workers reported that a NiO-loaded SrTiO<sub>3</sub> powder was capable of decomposing pure water as well as water vapor into H2 and O<sub>2</sub> under UV irradiation. <sup>109-114</sup> The activity of the photocatalyst was increased considerably by a pretreatment in H<sub>2</sub> and using a concentrated NaOH solution for the photocatalytic reaction.<sup>111</sup> The photocatalytic activity of SrTiO<sub>3</sub> was also greatly improved by using a modified preparation method<sup>115</sup> or a suitable concentration of metal cations doping (such as La<sup>3+</sup>, <sup>116</sup> Ga<sup>3+</sup>, <sup>117</sup> and Na<sup>+ 117</sup>). Some derivatives, such as Sr<sub>3</sub>Ti<sub>2</sub>O<sub>7</sub> and Sr<sub>4</sub>Ti<sub>3</sub>O<sub>10</sub>, were also found to be active in water decomposition when loaded with NiO as the cocatalysts. 118,119 Mizoguchi et al. reported that platinized CaTiO<sub>3</sub> powder, with band gap of 3.5 eV estimated from optical absorption edge, exhibited a high photocatalytic activity under UV irradiation. 120 By doping with Zr4+ to form a CaTi<sub>1-x</sub>Zr<sub>x</sub>O<sub>3</sub> solid solution, the activity was further increased. Quantum yields of up to 1.91% and 13.3% for H<sub>2</sub> evolution from pure water and aqueous ethanol solution, respectively, were reported. 121

Lee and co-workers investigated a series of perovskites whose layers were integrated of the intergrowth with the same elements (La and Ti) but had different layer thicknesses. 122-124 La<sub>2</sub>TiO<sub>5</sub>, La<sub>2</sub>Ti<sub>3</sub>O<sub>9</sub>, and La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub>, which had layered structures made up of slabs of 1, 3, and 4 units, respectively, showed much higher photocatalytic activities under UV irradiation than bulk LaTiO<sub>3</sub>. Alkaline earth component-doping (Ba, Sr, and Ca) was shown to improve the photoactivities of La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub>. In particular, the NiOmodified Ba-doped La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub> proved extremely active for overall water splitting with a quantum yield close to 50% on condition that alkaline hydroxide was introduced into the reaction system as an external additive. 125 The activity of La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub> was highly enhanced by synthesizing the catalyst using the polymerized approach instead of the solid-state reaction method.<sup>22,126</sup> In contrast, La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub>, Ln<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub> (Ln = Pr, Nd) with a layered structure was also active for water splitting but exhibited lower activities, with the activity decreasing in the order  $La_2Ti_2O_7 \gg PrLaTi_2O_7 > Pr_2Ti_2O_7 >$  $NdLaTi_2O_7 > Nd_2Ti_2O_7.^{127}$ 

The use of titanate  $R_2Ti_2O_7$  (R = Y, Eu-Lu) with pure cubic-pyrochlore structure as water-splitting photocatalysts was first reported in 2004 by Abe's group. 128 Among them, NiO<sub>x</sub>-loaded Y<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub> demonstrated the most efficient evolution of H<sub>2</sub> and O<sub>2</sub> in a stoichiometric ratio from pure water under UV irradiation. Y<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub> photocatalysts, with better crystallinity and higher activity, were obtained by the addition of excess Y in the polymerized complex synthetic procedure. 129,130 They also found that the high photocatalytic activity observed for overall water splitting over  $R_2Ti_2O_7$  (R=Y,Gd) was related to the increased mobility of the electrons and holes caused by the corner-shared octahedral  $TiO_6$  network in these materials. Uno et al. also studied the photocatalytic activities for hydrogen evolution using  $Ln_2Ti_2O_7$  (Ln=La, Pr, Nd, Sm, Gd, Dy, Ho, Er, and Yb). However, only  $La_2Ti_2O_7$  and  $Sm_2Ti_2O_7$  showed detectable photocatalytic hydrogen evolution, and even that was a small rate.

Miseki et al. studied the photocatalytic properties of  $ALa_4Ti_4O_{15}$  (A = Ca, Sr, and Ba) with a (111) plane-type layered perovskite structure. 133 Of these, NiO<sub>x</sub>-modifed BaLa<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub> showed the highest activity for water splitting, with a quantum yield of 15% at 270 nm. The highly donordoped (110) layered perovskite La<sub>4</sub>CaTi<sub>5</sub>O<sub>17</sub> was found to be an efficient photocatalyst for overall water splitting with quantum yield as high as 20% under UV irradiation. 134 The spontaneously hydrated layered perovskites with a general formula of  $A_{2-x}La_2Ti_{3-x}Nb_xO_{10}$  (A = interlayer cations such as K, Rb, Cs; x = 0-1) were found to be efficient photocatalysts for water decomposition under UV irradiation. 135-137 NiO/Rb<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub> produced H<sub>2</sub> and O<sub>2</sub> from a RbOH aqueous solution with quantum yield up to 5%. Suitable modification techniques such as cocatalyst loading, using Ni or Au, 138,139 metal-ion doping (Zn<sup>2+</sup> doping), 140 and alternative synthetic methods<sup>23,141</sup> all led to the enhanced activity of K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub>. The polymerized complex synthesis method was also utilized to prepare the high-purity stoichiometric KTiNbO<sub>5</sub> photocatalyst, which was demonstrated with increased activity after NiO loading. 142 For KLaTiO<sub>4</sub>, Zr<sup>4+</sup> doping had a positive effect on the photocatalytic activity in a water-splitting reaction, giving rise to a quantum yield as high as 12.5%. 143

Sekine et al. were the first to examine the photocatalytic reactions on the ion-exchangable layered titanoniobate compounds, CsNbTi<sub>2</sub>O<sub>7</sub> and CsNbTiO<sub>5</sub>. <sup>144</sup> They found that the H<sup>+</sup>-exchanged forms of those compunds work as efficient photocatalysts for H<sub>2</sub> or O<sub>2</sub> evolution from an aqueous solution containing methanol or silver nitrate under UV irradiation. Members of the aurivillius-type layered perovskites  $(Bi_2O_2)^{2+}(A_{n-1}B_nO_{3n+1})^{2-}$  (A = Ba, Bi, etc.; B = Ti, Nb, etc.), Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub>, BaBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub>, and Bi<sub>3</sub>TiNbO<sub>9</sub> evolved H<sub>2</sub> and O<sub>2</sub> from aqueous methanol and AgNO<sub>3</sub> solutions, respectively. 145 He and Yang investigated the photocatalytic activity for hydrogen production over another layered perovskite KBi<sub>3</sub>PbTi<sub>5</sub>O<sub>16</sub> under UV irradiation. 146 It was found that the photocatalyst prepared by the polymerized complex method showed much higher activity than that prepared by the solid-state reaction method. In addition, the rate of hydrogen evolution was greatly improved and affected by the addition of  $Ce(SO_4)_2$  in an aqueous suspension.

Sayama and Arakawa were the first to find that the photocatalytic decomposition of pure water proceeded over ZrO<sub>2</sub> powder without any loaded metals as cocatalysts under UV irradiation. The activity was affected significantly by the pressure of the reaction system, the nature of the additive, and the pH of the solution. Specifically, the addition of Na<sub>2</sub>CO<sub>3</sub> or NaHCO<sub>3</sub> led to a remarkable increase in the activity and stability of the gas-evolution rate. However, the activity decreased when metals such as Pt, Au, Cu, and RuO<sub>2</sub> were loaded onto the surface of the ZrO<sub>2</sub>. It was presumed that the large electronic barrier height of the semiconductor—metal junction prevented the migration of electrons from ZrO<sub>2</sub> to the metal. This would lead to the loaded metals,

possibly blocking the reaction sites on the ZrO<sub>2</sub>. Reddy et al. revealed that the photocatalytic activity of ZrO<sub>2</sub> prepared by the precipitation method was highly dependent on the hydrolyzing agent used. 149 The highest photocatalytic activity was obtained for the ZrO<sub>2</sub> with the highest surface area when KOH was used as the hydrolyzing agent. Zou et al. reported that a clean and direct metal-support interface of a NiOloaded ZrO<sub>2</sub> photocatalyst could be obtained using a plasma method. 150 In photocatalytic reactions, this interface proved more efficient for the charge separation and transfer, which in turn led to the higher photocatalytic activity for water splitting using NiO/ZrO<sub>2</sub> by the plasma treatment than that prepared with the traditional thermal treatment. Compared to the conventional bulk ZrO<sub>2</sub>, the photocatalytic watersplitting activity was greatly enhanced by the high dispersion of ZrO<sub>2</sub> into the amorphous wall of MCM-41.<sup>151</sup>

BaZrO<sub>3</sub> with a cubic perovskite structure produced hydrogen efficiently with a quantum yield up to 3.7% from pure water without the assistance of any cocatalysts under UV irradiation. 152 The high photoactivity of BaZrO<sub>3</sub> was attributed to the highly negative potential of the photoinduced electrons, the 180° Zr-O-Zr bond angle, and the large dispersion of the conduction band edge composed of Zr 4d orbitals. When the Zr element was partially substituted by Sn, the photocatalytic activity for water splitting was obviously improved. 153 Uno et al. investigated the photocatalytic activities for hydrogen evolution of a series of lanthanide zirconium oxides,  $Ln_2Zr_2O_7$  (Ln = La, Ce, Nd and Sm).<sup>154</sup> Under the illumination of a 500-W Xenon lamp, hydrogen gas was clearly evolved in a distilled water suspension of La<sub>2</sub>Zr<sub>2</sub>O<sub>7</sub>, Sm<sub>2</sub>Zr<sub>2</sub>O<sub>7</sub>, and Nd<sub>2</sub>Zr<sub>2</sub>O<sub>7</sub>. On the other hand, Ce<sub>2</sub>Zr<sub>2</sub>O<sub>7</sub> showed no photocatalytic activity because of its lower conduction band level.

After modification with Pt as a cocatalyst, mesoporous zirconium—titanium phosphates demonstrated considerable activity in photocatalytic water decomposition.  $^{155}$  The  $\rm H_2$  production rate was gradually increased on the addition of Zr, and the maximum  $\rm H_2$  evolution was observed for the Zr\_{0.5}Ti\_{0.5}PO\_4 material. Furthermore, the use of sodium carbonate as a pH adjuster was essential and significant for hydrogen generation and also provided stability for the photocatalytic reaction system. A zirconium phosphate/ phosphonate compound with quantum yield of 4% was developed to produce hydrogen photochemically from water. It was based only on the ultraviolet portion of the spectrum in the presence of a sacrificial reductant (EDTA).  $^{156}$ 

#### 3.1.2. Nb-, Ta-Based Oxides

Pure  $Nb_2O_5$ , with a band gap of ca. 3.4 eV determined from the threshold of UV-vis absorption, is not active for pure water splitting under UV irradiation. <sup>157</sup> After modification with Pt as a cocatalyst, however, it can efficiently produce  $H_2$  from aqueous solutions containing methanol as an electron donor. <sup>158</sup> Mesoporous  $Nb_2O_5$ , synthesized via an evaporation-induced self-assembly method, demonstrated a photocatalytic activity 20 times higher for hydrogen evolution than a bulk  $Nb_2O_5$  without any porosity. <sup>158</sup> Intercalation of  $In_2O_3$  into the mesoporous structure further increased the photoactivity of mesoporous  $Nb_2O_5$  by 2.7 times. <sup>159</sup>

Besides  $Nb_2O_5$ , a large number of niobates can produce  $H_2$  and  $O_2$  via water splitting upon UV irradiation. In 1986, Domen and co-workers developed  $K_4Nb_6O_{17}$  as the first example of a niobate photocatalyst that showed high and stable activity for  $H_2$  evolution from aqueous methanol

solution without any assistance from other materials such as the noble metals. 160,161 This niobate is composed of layers of niobium oxide sheets, in which potassium ions are located in two different kinds of interlayers. One type of interlayer contains water molecules and potassium ions, and the other contains only potassium ions. The potassium ions between the niobium oxide layers can be exchanged with many other cations including transition metal ions. The activities of catalysts exchanged with H<sup>+</sup>, Cr<sup>3+</sup>, and Fe<sup>3+</sup> ions were higher than the original K<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub>. Of particular note is the H<sup>+</sup>exchanged K<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub>, which showed the highest activity for H<sub>2</sub> evolution from an aqueous methanol solution. Its quantum yield was up to ca. 50% at 330 nm. 160,162 After modification with NiO, 163–166 Au, 167 Pt, 168,169 and Cs 170 as cocatalysts, K<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub> was quite efficient for simultaneous hydrogen and oxygen evolution from pure water. Upon the addition of alkaline hydroxide (KOH, NaOH) into the aqueous impregnation solution, the activity of the NiO/K<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub> photocatalyst for overall water splitting was enhanced, with a quantum efficiency of 5.3% (330 nm) under optimum conditions. 166 Rb<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub>, which has the same layered structure, exhibited a high activity for photocatalytic water splitting to form H<sub>2</sub> and O<sub>2</sub> under band gap irradiation. The quantum efficiency at 330 nm was ca. 10% in the initial stage of the reaction over the NiO(0.1 wt %)–Rb<sub>4</sub>Nb<sub>6</sub>O $_{17}$  photocatalyst.  $^{171}$  When doped with Ta<sub>2</sub>O<sub>5</sub>, layered compounds of the type  $A_4Ta_xNb_{6-x}O_{17}$  (A = K or Rb, x = 2, 3, and 4) were able to decompose water stoichiometrically after the pretreatment of H<sub>2</sub> reduction and O<sub>2</sub> reoxidation at high temperatures. This was possible even in the absence of any loaded metals as cocatalysts. 157 Some other alkaline-metal niobates such as  $ANbO_3$  (A = Li, Na, K)<sup>172-174</sup> and  $Cs_2Nb_4O_{11}^{175}$  also catalyzed H2 and/or O2 evolution from water under UV irradiation, but only after modification with Pt, RuO<sub>2</sub>, or NiO. Ikeda et al. found that tungsten-containing alkaline niobates with a defect pyrochlore structure, ANbWO<sub>6</sub> (A = Rb, Cs), and loaded with nickel oxide, showed photocatalytic activity for overall water splitting under UV irradiation. 176 The conduction bands of the materials were thought to be composed of the W5d orbital hybridized with the Nb5d orbital.

Photocatalytic water splitting over the alkaline-earth niobates have been studied by various researchers. The related strontium niobates  $SrNb_2O_6$ ,  $^{177}$   $Sr_2Nb_2O_7$ ,  $^{21,134,177}$  and Sr<sub>5</sub>Nb<sub>4</sub>O<sub>15</sub><sup>133</sup> exhibited efficient photocatalytic activities for hydrogen and oxygen production from pure water under UV irradiation. In particular, Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub> with a highly donor-doped (110) layered perovskite structure gave quantum yields as high as 23%. The activity of Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub> was further enhanced to give a quantum yield of 32%, by using a hydrothermal synthetic process that produced a 1D nanostructure with larger Brunauer-Emmett-Teller (BET) surface areas.<sup>177</sup> In comparison, the quantum yield of Ca<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub> was 7%.<sup>134</sup> Ba<sub>5</sub>Nb<sub>4</sub>O<sub>15</sub> with a layered perovskite structure was studied by Kudo and co-workers. 133,134 It gave a 17% quantum yield at 270 nm for water splitting when loaded with NiO cocatalysts. Partial substitution of Nb<sup>5+</sup> with Zn<sup>2+</sup> gave the resulting  $BaZn_{1/3}Nb_{2/3}O_3$  with a distorted perovskite structure, which showed favorable photocatalytic activity under UV irradiation. 178,179

Domen and co-workers first reported a novel Dion-Jacobsen series of ion-exchangeable niobates,  $A(M_{n-1}Nb_n-1)$  $O_{3n+1}$ ) (A = Na, K, Rb, Cs; M = La, Ca, Sr, etc.), with layered perovskite structures that showed unique photocatalytic activities. This was especially true for the H<sup>+</sup>-exchanged forms, for H<sub>2</sub> evolution from aqueous alcohol solutions as well as O2 evolution from an aqueous silver nitrate solution. 38,180 For example, KSr<sub>2</sub>Nb<sub>3</sub>O<sub>10</sub> produced hydrogen at a rate of 0.11 mmol/h/g. After cation exchange with protons, the rate of HSr<sub>2</sub>Nb<sub>3</sub>O<sub>10</sub> increased to 43 mmol/h/g.<sup>38</sup> The related layered niobate K<sub>2,33</sub>Sr<sub>0,67</sub>Nb<sub>5</sub>O<sub>14,335</sub> was also reported by Wu and co-workers to show much higher photoactivity for hydrogen evolution after a proton-exchange reaction.<sup>181</sup> They further reported that other modifications for proton-exchanged  $H(M_{n-1}Nb_nO_{3n+1})$  (M = La, Ca, Sr, etc.), such as metal-ion doping (La<sup>3+</sup>, <sup>182</sup> In<sup>3+183</sup>, and Mo<sup>6+184</sup> doped into HCa<sub>2</sub>Nb<sub>3</sub>O<sub>10</sub> and H<sub>2</sub>LaNb<sub>2</sub>O<sub>7</sub>, respectively), efficiently improved the photocatalytic hydrogen evolution from aqueous methanol solutions. When H<sub>2</sub>LaNb<sub>2</sub>O<sub>7</sub> was synthesized using a polymerized complex method, it showed higher activity for water splitting than the same material prepared by a solid-state reaction. 185

Ebina and co-workers synthesized a restacked aggregate of exfoliated nanosheets of [Ca<sub>2</sub>Nb<sub>3</sub>O<sub>10</sub>] by flocculation with NaOH and KOH aqueous solutions. Under UV irradiation, the restacked aggregates, with a 10-fold enhancement of the surface area, showed higher activities for photocatalytic hydrogen evolution than the KCa<sub>2</sub>Nb<sub>3</sub>O<sub>10</sub> starting compound; 186 overall photocatalytic splitting of water was achieved when RuO<sub>x</sub> was intercalated between the layers during the exfoliation-restacking route. 187 After exfoliation using tetrabutylammonium hydroxide, the restacked  $TBA_x[H_{1-x}Ca_2Nb_3O_{10}]$  sheets loaded with Pt cocatalysts produced hydrogen from pure water with a quantum efficiency of 7.5%. However, no oxygen was observed. Transient absorption measurements of the nanosheets revealed charge separation on a nanosecond time scale. 188 SiO<sub>2</sub>pillared HCa<sub>2</sub>Nb<sub>3</sub>O<sub>10</sub>, prepared from the layered perovskite KCa<sub>2</sub>Nb<sub>3</sub>O<sub>10</sub> via alkylammonium ion-intercalated HCa<sub>2</sub>-Nb<sub>3</sub>O<sub>10</sub>, showed much higher photocatalytic activity of H<sub>2</sub> evolution from aqueous solutions of long-chain alcohols. 189,190 This effect was attributed to the expanded interlayer space facilitating the intercalation of such alcohols that could then serve as efficient electron donors. In 2008 a new member of the Dion-Jacobsen perovskites, H<sub>2</sub>Ca<sub>4</sub>Nb<sub>6</sub>O<sub>20</sub>, was reported to be active for H<sub>2</sub> evolution in the presence of methanol as a sacrificial agent under UV irradiation. 191 When Nb was partially substituted by Ta, the resulting H<sub>2</sub>Ca<sub>4</sub>Ta<sub>2</sub>Nb<sub>4</sub>O<sub>20</sub> showed the best photocatalytic activity. The photocatalytic H<sub>2</sub> evolution rate was 8.5 mmol/h/g. Abe and co-workers investigated the effect of crystal structure on water splitting using  $R_3NbO_7$  (R = Y, Yb, Gd, La).  $^{131,192}$  Only La<sub>3</sub>NbO<sub>7</sub>, which has an orthorhombic weberite structure and which formed a NbO<sub>6</sub> octahedral network that increased the mobility of both electrons and holes, was active for the photocatalytic water-splitting reaction. On the other hand, La<sub>1/3</sub>NbO<sub>3</sub>, crystallizing in an A-site deficient perovskitetype structure instead, catalyzed H<sub>2</sub> evolution with a Pt cocatalyst from CH<sub>3</sub>OH/H<sub>2</sub>O solution. In addition, O<sub>2</sub> evolution from AgNO3 aqueous solution under UV irradiation was observed. 193

Kudo et al. reported a new ZnNb<sub>2</sub>O<sub>6</sub> photocatalyst consisting of d<sup>10</sup> and d<sup>0</sup> metal ions. <sup>194</sup> Under UV irradiation, the activity of the native ZnNb<sub>2</sub>O<sub>6</sub> was negligible, whereas NiO-loaded ZnNb<sub>2</sub>O<sub>6</sub> showed the high activity after a H<sub>2</sub>reduction and O2-oxidation pretreatment. Chen and coworkers prepared a new series of layered perovskite photocatalysts,  $ABi_2Nb_2O_9$  (A = Ca, Sr, Ba), by the conventional

solid-state reaction method. 195 Under UV irradiation, these photocatalysts showed photocatalytic activity for both H<sub>2</sub> and O<sub>2</sub> evolution from aqueous solutions containing sacrificial reagents (methanol and Ag<sup>+</sup>). The activities decreased in the order of SrBi<sub>2</sub>Nb<sub>2</sub>O<sub>9</sub> > BaBi<sub>2</sub>Nb<sub>2</sub>O<sub>9</sub> > CaBi<sub>2</sub>Nb<sub>2</sub>O<sub>9</sub>. Zou and co-workers found that the compounds  $M_2BiNbO_7$  (M =  $In^{3+}$ , Ga<sup>3+</sup>) with pyrochlore structures were sensitive to UV irradiation and had the ability to split water stoichiometrically to produce H<sub>2</sub> and O<sub>2</sub>. <sup>196</sup> On the other hand, Bi<sub>2</sub>MNbO<sub>7</sub> (M =  $A1^{3+}$ ,  $Ga^{3+}$ ,  $In^{3+}$ ) with the same pyrochlore structure only evolved H<sub>2</sub> or O<sub>2</sub> in the presence of CH<sub>3</sub>OH or Ce(SO<sub>4</sub>)<sub>2</sub> as sacrificial agents. 197 The photocatalytic hydrogen production over NiO/Bi<sub>2</sub>AlNbO<sub>7</sub> was efficiently improved with La<sup>3+</sup> doping. <sup>198</sup> In comparison,  $Bi_2MNbO_7$  (M =  $Y^{3+}$ ,  $Ce^{3+}$ ,  $Gd^{3+}$ , Sm<sup>3+</sup>, Nd<sup>3+</sup>, Pr<sup>3+</sup>, and La<sup>3+</sup>) evolved H<sub>2</sub> at only a small rate from pure water under UV irradiation. 199 The increase of ion radius of M<sup>3+</sup> in Bi<sub>2</sub>MNbO<sub>7</sub> led to the decrease in the photocatalytic activity. The narrower band gap formed by the smaller ion radius of M<sup>3+</sup> suggested easier excitation for an electron from the valence band to the conduction band in the oxide semiconductor. BiNbO<sub>4</sub> with a triclinic structure produced only a small amount of hydrogen from pure water under UV irradiation. When doped with Ta, the resulting BiTa<sub>0.8</sub>Nb<sub>0.2</sub>O<sub>4</sub> had an orthorhombic structure and exhibited much higher activity due to the modified band levels and band gaps.200

Ta-based oxides are known to be active photocatalysts for water splitting under UV irradiation. Under band gap (4.0 eV) irradiation, Ta<sub>2</sub>O<sub>5</sub> alone can only produce a very small amount of H<sub>2</sub> and no O<sub>2</sub> from pure water. 157,182 After modification with NiO and RuO2 as the cocatalysts, it displayed great activity for the overall water decomposition. 201 The addition of Na<sub>2</sub>CO<sub>3</sub> is also effective for improving the photocatalytic activity of Ta<sub>2</sub>O<sub>5</sub>, as observed for the TiO<sub>2</sub> photocatalytic system.<sup>78</sup> Mesoporous Ta<sub>2</sub>O<sub>5</sub> was found by Domen and co-workers to be an active catalyst for photocatalytic water decomposition after NiO loading.<sup>202</sup> Although the walls of the mesoporous Ta<sub>2</sub>O<sub>5</sub> were amorphous, the photocatalytic activity was higher than that of crystallized Ta<sub>2</sub>O<sub>5</sub>. This was because the small wall thickness of mesoporous Ta<sub>2</sub>O<sub>5</sub> favored the migration of excited electron to the surface. When mixed with TiO<sub>2</sub>, the mesoporous Ta-Ti mixed oxides formed showed relatively high activities for photocatalytic hydrogen generation from methanol/water mixtures under the irradiation of  $\lambda > 300$  nm. However, a higher concentration of TiO<sub>2</sub> in the mixed oxides led to the destruction of the mesoporous structure.<sup>203</sup> In contrast, the introduction of MgO into the mesoporous structure of Ta<sub>2</sub>O<sub>5</sub> improved the thermal stability and also gave rise to stable photocatalytic activity for overall water decomposition over Mg-Ta mixed oxides.<sup>204</sup> Ni-mixed mesoporous Ta oxide possessed an incomplete mesoporous structure but exhibited higher photocatalytic activity for overall water decomposition than nonmodified mesoporous Ta oxide under UV irradiation, while Cu-mixed mesoporous Ta oxide evolved H2 only at a low rate.205

In 1998, Kato and Kudo reported that alkali tantalates ATaO<sub>3</sub> (A = Li, Na, and K) showed high activities for photocatalytic water splitting into  $H_2$  and  $O_2$  under UV irradiation. The excess of alkali in the synthetic process of the solid-state reaction improved the photocatalytic activities of naked tantalates. The order of the activities was  $KTaO_3 \ll NaTaO_3 < LiTaO_3$ . Nickel oxide and nanosized gold particles were shown to function as

efficient cocatalysts for photocatalytic water splitting. Among them, NiO/NaTaO<sub>3</sub> was the most photocatalytically active and produced  $H_2$  and  $O_2$  from pure water with a quantum yield of 28% at 270 nm.<sup>207</sup> Compared to the solid-state reaction method, both the hydrothermal method<sup>209,210</sup> and the sol-gel method<sup>211</sup> produced ATaO<sub>3</sub> (A = K, Na) with good crystallinity as well as high surface areas. These exhibited higher photocatalytic activity in water splitting than the solidstate produced material. Ishihara and co-workers found for the first time that controlling the charge density in KTaO<sub>3</sub> by doping small amounts of acceptors such as tri- or tetravalent cations was effective for improving the photolysis activity of H<sub>2</sub>O.<sup>212,213</sup> In particular, NiO/KTaO<sub>3</sub> doped with 8 mol % Zr<sup>4+</sup> exhibited a higher photocatalytic activity than the well-known photocatalyst Pt/TiO<sub>2</sub>. The increased activity was brought about by an increase in the lifetime of the photoexcited charge, which in return was caused by a decrease in the charge density.

Kudo and Kato investigated the effects of doping lanthanide (La, Pr, Nd, Sm, Gd, Tb, and Dy)<sup>214</sup> and alkalineearth metal ions (Ca, Sr, and Ba)<sup>215</sup> into NaTaO<sub>3</sub> photocatalysts for efficient water splitting. Lanthanum was the most effective dopant. The apparent quantum yield at 270 nm amounted to 56%, 20 which is the highest quantum yield ever reported for catalysts in pure water splitting. The positive effects on the photocatalytic properties were mainly due to the decrease in the particle size and the ordered surface nanostructure. The many characteristic steps created by the doping affected the electron-hole recombination kinetics as revealed by a time-resolved infrared absorption study of the NaTaO<sub>3</sub>-based photocatalysts. <sup>216,217</sup> The related photocatalyts  $NiO/A_2Ta_2O_6$  (A = K, Na) also worked efficiently for overall water splitting under UV irradiation.<sup>218,219</sup> For the alkalineearth tantalates  $ATa_2O_6$  (A = Ca, Sr, Ba), the order of photocatalytic activities was  $SrTa_2O_6 > BaTa_2O_6 > CaTa_2O_6$ . This corresponded to the band gaps and the energy of the photogenerated-electron/hole pairs transferring in the crystal (emission energy). <sup>201,208,220</sup> When the NiO cocatalysts were loaded, NiO/SrTa<sub>2</sub>O<sub>6</sub> showed the highest activity for overall water splitting. The quantum yield was 7% at 270 nm. 220 In the related series of strontium tantalates  $Sr_mTa_nO_{(m+5n/2)}$ , the photocatalytic activities for water splitting into H<sub>2</sub> and O<sub>2</sub> decreased in the following order: Sr<sub>2</sub>Ta<sub>2</sub>O<sub>7</sub> > Sr<sub>5</sub>Ta<sub>4</sub>O<sub>15</sub> >  $SrTa_2O_6 > Sr_4Ta_2O_9$ . The quantum yield of NiO (0.15 wt %)/Sr<sub>2</sub>Ta<sub>2</sub>O<sub>7</sub> prepared by the polymerized complex method was estimated to be  $\sim$ 24% at 270 nm. <sup>21,222</sup> Substitution of Ta for Nb reduced the band gap of Sr<sub>2</sub>Ta<sub>2</sub>O<sub>7</sub> from 4.5 to 3.9 eV. 222,223 Under UV irradiation, all the  $Sr_2(Ta_{1-x}Nb_x)_2O_7$  solid solutions loaded with NiO cocatalysts decomposed water into H<sub>2</sub> and O<sub>2</sub> stoichiometrically. But the photocatalytic activity decreased dramatically even when the amount of Nb was small. Ca<sub>2</sub>Ta<sub>2</sub>O<sub>7</sub> with a pyrochlore structure produced H<sub>2</sub> and O<sub>2</sub> in a stoichiometric ratio under UV irradiation. The activity was higher than NiO-loaded niobate pyrochlore, NiO/ Ca<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub>.<sup>218</sup> Otsuka et al. claimed that, in comparison to BaTa<sub>2</sub>O<sub>6</sub>, Ba<sub>5</sub>Ta<sub>4</sub>O<sub>15</sub> prepared under a Ta-rich atmosphere showed a higher photocatalytic activity in the decomposition of H<sub>2</sub>O into H<sub>2</sub> and O<sub>2</sub> under UV irradiation.<sup>224</sup> When Ta was partially replaced by Ni or Zn, only H<sub>2</sub> from water was evolved from the resulting  $BaM_{1/3}Ta_{2/3}O_3$  (M = Ni, Zn). Methanol was the electron donor, and Pt was the cocatalyst. 178,225

Shimizu and co-workers first developed  $A_2A'Ta_2O_7$  (A = H, K, and Rb; A' = Sr and  $La_{2/3}$ ) with a hydrated layered perovskite structure. It proved highly efficient for overall

water splitting, even without cocatalysts loading.<sup>226,227</sup> These catalysts showed higher activities than the anhydrous perovskites (Li<sub>2</sub>SrTa<sub>2</sub>O<sub>7</sub>, La<sub>1/3</sub>TaO<sub>3</sub>, and KTaO<sub>3</sub>). This was attributed to their hydrated layered structure where the photogenerated electrons and holes can be effectively transferred to the interlayer water. Moreover, as a result of the intercalation of small NiO clusters into the layers of H<sub>2</sub>La<sub>2/3</sub>Ta<sub>2</sub>O<sub>7</sub> via an ion-exchange reaction, their cocatalyst action remarkably increased the overall activity. This was achieved by shortening the migration distance of the photogenerated charges to the reactive sites. On the other hand, the relatively large NiO particles at the external surface of H<sub>2</sub>SrTa<sub>2</sub>O<sub>7</sub> did not improve the activity. Interestingly, hydration under aqueous conditions changed the crystal structure of K<sub>2</sub>Sr<sub>1.5</sub>Ta<sub>3</sub>O<sub>10</sub> from orthorhombic to a tetragonal symmetry. This was due to water intercalation into the interlayer space.<sup>228</sup> With RuO<sub>x</sub> as the cocatalyst, the hydrous K<sub>2</sub>Sr<sub>1.5</sub>Ta<sub>3</sub>O<sub>10</sub> photocatalyst was active in the pure watersplitting process: the quantum yield at 252.5 nm was  $\sim$ 2%. For the layered perovskites  $ACa_2Ta_3O_{10}$  (A = Cs, Rb, K, Na, and Li), interlayer hydration was only observed for A = Na and Li.<sup>229</sup> Hydration of the Li phase doubled the rate of photocatalytic gas evolution when loaded with the cocatalyst NiO; this was the highest in the ACa<sub>2</sub>Ta<sub>3</sub>O<sub>10</sub> series. Analogously, a quantum yield of 8% was achieved for the Ba-based (100) layered perovskite KBa<sub>2</sub>Ta<sub>3</sub>O<sub>10</sub>, after modification with NiO. 134

Li et al. prepared another hydrated layered perovskite, namely, the tantalate H<sub>1.81</sub>Sr<sub>0.81</sub>Bi<sub>0.19</sub>Ta<sub>2</sub>O<sub>7</sub>, from Bi<sub>2</sub>SrTa<sub>2</sub>O<sub>9</sub> using an ion-exchange reaction in hydrochloric acid solution.<sup>230</sup> Under UV irradiation, the H<sub>1.81</sub>Sr<sub>0.81</sub>Bi<sub>0.19</sub>Ta<sub>2</sub>O<sub>7</sub> photocatalyst showed favorable photocatalytic activity in splitting pure water into H<sub>2</sub> and O<sub>2</sub> even without the assistance of a cocatalyst. Layered lanthanide tantalates and their ion-exchanged phases ( $MLnTa_2O_7$ , M = Cs, Rb, Na, and H; Ln = La, Pr, Nd, and Sm) were prepared by Machida and co-workers to evaluate their photocatalytic activity for water splitting under UV irradiation. <sup>231–234</sup> The photocatalytic activity was sensitive to not only Ln but also the interlayer cation, M. The highest activity was obtained for M = Rbwith the following sequence of Ln: Rb > Nd > Sm > La > Pr. The effects of lanthanide ions on the photocatalytic activities of LnTaO<sub>4</sub> (Ln = La, Ce, Pr, Nd, and Sm) with monoclinic structures and  $K_2LnTa_5O_{15}$  (Ln = La, Pr, Nd,Sm, Gd, Tb, Dy, and Tm) with tungsten bronze structures were also investigated by Machida's group. 235,236 The photocatalytic activities for water splitting under UV irradiation strongly depended on the particular lanthanide ion. For the LnTaO<sub>4</sub> series, LaTaO<sub>4</sub> showed the highest rate for the stoichiometric evolution of H<sub>2</sub> and O<sub>2</sub> from pure water; while among the K<sub>2</sub>LnTa<sub>5</sub>O<sub>15</sub> series, K<sub>2</sub>PrTa<sub>5</sub>O<sub>15</sub> and K<sub>2</sub>SmTa<sub>5</sub>O<sub>15</sub> showed relatively high activities for water splitting ( $h\nu$  > 3.9-4.1 eV).

The photocatalytic activity of  $R_3TaO_7$  (R = Y, Yb, Gd, La) for water splitting was studied by Arakawa and co-workers. <sup>131,192</sup> They examined the effect of R<sup>3+</sup> ionic radius on the crystal structure. 131,192 The crystal structures of R<sub>3</sub>TaO<sub>7</sub> changed with increasing ionic radius of the R<sup>3+</sup> ion from a fluorite-type cubic structure to a pyrochloretype cubic structure and finally to a weberite-type orthorhombic structure. In the case of the La<sub>3</sub>TaO<sub>7</sub>, the photocatalytic activity was greatly increased by the phase transition from cubic to orthorhombic. K<sub>3</sub>Ta<sub>3</sub>Si<sub>2</sub>O<sub>13</sub> and K<sub>3</sub>Ta<sub>3</sub>B<sub>2</sub>O<sub>12</sub>, which have similar crystal structures, consisting of pillars formed by a corner sharing of three linear TaO<sub>6</sub> chains, were active for water splitting without any cocatalyst. 237,238 The TaO<sub>6</sub> pillars in K<sub>3</sub>Ta<sub>3</sub>Si<sub>2</sub>O<sub>13</sub> and K<sub>3</sub>Ta<sub>3</sub>B<sub>2</sub>O<sub>12</sub> are linked by Si<sub>2</sub>O<sub>7</sub> ditetrahedral units and BO3 triangle units, respectively. Thus, the bond angle of O-Ta-O in  $K_3Ta_3B_2O_{12}$  (171.5°) is slightly smaller than that in K<sub>3</sub>Ta<sub>3</sub>Si<sub>2</sub>O<sub>13</sub> (173.1°). The distortion due to the smaller bond angle of K<sub>3</sub>Ta<sub>3</sub>B<sub>2</sub>O<sub>12</sub> than that found in K<sub>3</sub>Ta<sub>3</sub>Si<sub>2</sub>O<sub>13</sub> resulted in a wider band gap and higher photocatalytic activity. NiO cocatalyst loading increased the activity of K<sub>3</sub>Ta<sub>3</sub>Si<sub>2</sub>O<sub>13</sub> drastically but proved ineffective for K<sub>3</sub>Ta<sub>3</sub>B<sub>2</sub>O<sub>12</sub>. In a manner similar to niobates, the tantalates  $ATaWO_6$  (A = Rb, Cs) crystallized into a defect pyrochlore structure with a conduction band composed of the W5d orbital hybridized with the Ta4d orbitals. 176 Under UV irradiation, they evolved H<sub>2</sub> and O<sub>2</sub> in a stoichiometric ratio from an aqueous AOH (A = Rb, Cs) solution. It was loaded with NiO as cocatalyst. Even though the pyrochlore-like  $Bi_2MTaO_7$  (M = Y and La) showed a strong optical absorption in the visible region, as photocatalyts these oxides could only produce H2 and O2 from pure water under UV irradiation.<sup>239</sup>

Chen and co-workers investigated the photocatalytic water splitting of La<sub>2</sub>AlTaO<sub>7</sub> with a view to studying the effect of aluminum on the electronic structure.<sup>240</sup> It was found that instead of the Ta5d state it was the Al3s3p states that acted as the lower conduction band. Under UV irradiation, La<sub>2</sub>AlTaO<sub>7</sub> showed photocatalytic activity in splitting pure water into H<sub>2</sub> and O<sub>2</sub> even in the absence of a cocatalyst. In comparison, using  $ABi_2Ta_2O_9$  (A = Ca, Sr, Ba), also developed by Chen's group, H<sub>2</sub> or O<sub>2</sub> evolved only from the aqueous solution containing either methanol or AgNO<sub>3</sub> as the sacrificial reagent. Simultaneous production of H<sub>2</sub> and O<sub>2</sub> from pure water was not observed. <sup>241</sup> The photocatalytic activities decreased in the order of SrBi<sub>2</sub>Ta<sub>2</sub>O<sub>9</sub> > CaBi<sub>2</sub>Ta<sub>2</sub>O<sub>9</sub> > BaBi<sub>2</sub>Ta<sub>2</sub>O<sub>9</sub>. A large range of transition-metal tantalates has been investigated by different researchers with water decomposition as the aim. Under UV irradiation, NiTa<sub>2</sub>O<sub>6</sub> produced both H<sub>2</sub> and O<sub>2</sub> from pure water without a cocatalyst. On the other hand, with MnTa<sub>2</sub>O<sub>6</sub>, CoTa<sub>2</sub>O<sub>6</sub>, CrTaO<sub>4</sub>, PbTa<sub>2</sub>O<sub>6</sub>, FeTaO<sub>4</sub>, and BiTaO<sub>4</sub>, only traces of H<sub>2</sub> evolved. 200,201 When NiO was loaded as the cocatalyst, AgTaO<sub>3</sub> and ZnTa<sub>2</sub>O<sub>6</sub> were active in water splitting, producing both  $H_2$  and  $O_2$ .<sup>201,242</sup> In contrast, in an aqueous methanol solution, Sn<sub>2</sub>Ta<sub>2</sub>O<sub>7</sub> with Pt as the cocatalyst only produced H<sub>2</sub> while SnTa<sub>2</sub>O<sub>6</sub> was totally inactive.<sup>243</sup>

### 3.1.3. W-, Mo-Based Oxides

The number of heterogeneous photocatalysts based on either tungstates or molybdates for H<sub>2</sub> or O<sub>2</sub> evolution is quite small. Some were found to be active for water splitting only under UV irradiation even though they showed optical absorption in the visible region. Inoue and co-workers found that PbWO<sub>4</sub> incorporating a WO<sub>4</sub> tetrahedron showed high and stable photocatalytic activity for the overall splitting of water. A stoichiometric quantity of H<sub>2</sub> and O<sub>2</sub> was produced under UV irradiation when RuO2 was loaded onto the metal oxide. 244,245 The photocatalytic performance was attributed to large dispersions in both the valence and conduction bands. This generated very mobile photoexcited holes and electrons. In contrast, a small dispersion in the conduction band was observed for the photocatalytically inactive CaWO<sub>4</sub>, which has a similar crystal structure. PbMoO<sub>4</sub> catalyzed hydrogen evolution from aqueous methanol solution. It also was capable of oxygen evolution from aqueous silver nitrate

solution under UV irradiation, and its oxygen evolution activity was comparable to that observed on TiO<sub>2</sub>. <sup>246</sup>

Kudo and co-workers have extensively investigated the photocatalytic activities of tungstates and molybdates.<sup>247,248</sup> Under UV irradiation,  $Na_2W_4O_{13}^{247}$  and  $Bi_2W_2O_9$ ,  $^{145}$  with layered structures, were active for photocatalytic hydrogen (Pt as cocatalyst) and oxygen evolution in the presence of suitable sacrificial reagents. However, Bi<sub>2</sub>MoO<sub>6</sub> with a similar structure evolved only oxygen from AgNO<sub>3</sub> aqueous solution at a low rate. 145 They also found that some scheelitetype molybdates and tungstates functioned as photocatalysts for both H2 or O2 evolution in the presence of sacrificial reagents.248 The band gaps (BGs) of scheelite compounds narrowed when they were composed of Ag<sup>+</sup> and Bi<sup>3+</sup> ions.  $(NaBi)_{0.5}MoO_4$  (BG = 3.1 eV),  $(AgBi)_{0.5}WO_4$  (BG = 3.2 eV), and  $(AgBi)_{0.5}MoO_4$  (BG = 3.0 eV) all showed photocatalytic activity for O2 evolution from an aqueous solution containing an electron acceptor. On the other hand,  $(NaBi)_{0.5}WO_4$  (BG = 3.5 eV) produced H<sub>2</sub> from an aqueous solution containing an electron donor. In comparison,  $(AgLn)_{0.5}MoO_4$  (Ln = La, Ce, Eu, Yb) barely produced H<sub>2</sub> or O<sub>2</sub> from aqueous solutions under UV irradiation.

Nguyen et al. developed a novel silicotungstic acid  $(SWA)-SiO_2$  photocatalyst by impregnating of SWA on a silica support. Under UV irradiation, an approximately stoichiometric production ratio of  $H_2$  and  $O_2$  was observed on these  $SWA-SiO_2$  photocatalyts. The role of the photoactive sites in  $SiO_2$  as a donor source for hydrogen formation and that of SWA as an inhibitor for the recombination of photoexcited electrons and holes were appreciable and responsible for the superior photocatalytic performance of the  $SWA-SiO_2$  system.

## 3.1.4. Other d<sup>0</sup> Metal Oxides

Some other miscellaneous  $d^0$  metal oxides that can catalyze water splitting to  $H_2$  and/or  $O_2$  under UV irradiation are described below. Wang et al. observed that a new crystal structure for nanostructured  $VO_2$ , with a body centered-cubic structure (bcc) and a large optical band gap of  $\sim\!2.7$  eV, surprisingly showed excellent photocatalytic activity in hydrogen production from a solution of water and ethanol under UV irradiation.  $^{250}$  The bcc  $VO_2$  phase exhibited a high quantum efficiency of  $\sim\!38.7\%$  when synthesized as nanorods. Luan and co-workers first prepared  $Bi_2GaVO_7$  and  $Bi_2YVO_8$  with tetragonal structures by solid-state reactions. These two compounds initiated both  $H_2$  and  $O_2$  evolution from pure water only under UV irradiation. This is in spite of the fact that both of them showed strong optical absorption in the visible region  $(\lambda > 420 \text{ nm}).^{251,252}$ 

## 3.2. d<sup>10</sup> Metal Oxide Photocatalyts

Various typical metal oxides with d<sup>10</sup> (In<sup>3+</sup>, Ga<sup>3+</sup>, Ge<sup>4+</sup>, Sn<sup>4+</sup>, Sb<sup>5+</sup>) configurations have all been shown to be effective photochemical water-splitting catalysts under UV irradiation. Of these, Ni-loaded Ga<sub>2</sub>O<sub>3</sub> was one of the promising photocatalysts for overall water splitting.<sup>253</sup> Its photocatalytic activity could be effectively improved by the addition of Ca, Cr, Zn, Sr, Ba, and Ta ions.<sup>254</sup> In particular, Zn ion doping remarkably improved the photocatalytic activity, with an apparent quantum yield for Ni/Zn-Ga<sub>2</sub>O<sub>3</sub> of ~20%. By combining with Lu<sub>2</sub>O<sub>3</sub>, the resulting Zn-doped Lu<sub>2</sub>O<sub>3</sub>/Ga<sub>2</sub>O<sub>3</sub> proved to be a novel composite photocatalyst for stoichiometric water splitting under UV irradiation. When

the system was loaded with NiO as the cocatalyst, the quantum yield at 320 nm was estimated to be 6.81%. <sup>255</sup> For solid solutions consisting of  $Ga_2O_3$  and  $In_2O_3$ ,  $Ga_{1.14}In_{0.86}O_3$  showed the highest photocatalytic activity for  $H_2$  evolution from aqueous methanol solutions and for  $O_2$  evolution from aqueous silver nitrate solutions. <sup>256</sup> In comparison, the solid solutions of  $Y_2O_3$  and  $In_2O_3$ ,  $Y_{1.3}In_{0.7}O_3$ , showed the highest photocatalytic activity for the overall water splitting when combined with  $RuO_2$  as a promoter. <sup>257</sup>

Inoue and co-workers investigated the photocatalytic properties for water decomposition of alkali metal, alkaline earth metal, and lanthanum indates with an octahedrally coordinated In<sup>3+</sup> d<sup>10</sup> configuration ion.<sup>258–262</sup> The photocatalytic activity for water decomposition under UV irradiation was considerably large for RuO<sub>2</sub>-dispersed CaIn<sub>2</sub>O<sub>4</sub>, SrIn<sub>2</sub>O<sub>4</sub>, and Sr<sub>0.93</sub>Ba<sub>0.07</sub>In<sub>2</sub>O<sub>4</sub> but very poor for RuO<sub>2</sub>-dispersed AInO<sub>2</sub> (A = Li, Na) and  $LnInO_3$  (Ln = La, Nd). The geometric structures of the InO<sub>6</sub> octahedral units for these indates were compared. It was shown that the photocatalytically active indates possessed distorted InO<sub>6</sub> octahedra with dipole moments. The internal fields that arose because of the dipole moment promoted the charge separation in the very initial process of photoexcitation. In addition, the broad sp conduction bands with large dispersions permitted the photoexcited electrons to move to the dispersed promoter RuO<sub>2</sub> particles. A group of p-block metal oxides was reported to have stable activity in decomposing water to H<sub>2</sub> and O<sub>2</sub> under UV irradiation when combined with RuO2 or Pt as cocatalyt.261-268 They consist of metal ions with d<sup>10</sup> configurations and have distorted octahedral and/or tetrahedral structures. For example, the distorted SbO<sub>6</sub> octahedra in  $M_2Sb_2O_7$  (M = Ca, Sr), 263 CaSb<sub>2</sub>O<sub>6</sub>, 263 and NaSbO<sub>3</sub>, 263 the distorted GeO<sub>4</sub> tetrahedra in Zn<sub>2</sub>GeO<sub>4</sub>,<sup>265</sup> and the distorted InO<sub>6</sub> octahedra and GeO<sub>4</sub> tetrahedra in LiInGeO<sub>4</sub><sup>266</sup> were dominantly responsible for photocatalytic activity for water decomposition. Some other metal oxides with d<sup>10</sup> configuration such as ZnGaO<sub>4</sub>,<sup>264</sup> Sr<sub>2</sub>SnO<sub>4</sub>,<sup>261</sup> and SrSnO<sub>3</sub><sup>267,268</sup> were also reported to show photocatalytic activity for water splitting.

## 3.3. f<sup>0</sup> Metal Oxide Photocatalysts

The f-block metal oxides usually combined with other metal oxides as photocatalysts. Pure CeO<sub>2</sub> powder was reported to show a consistent activity toward O<sub>2</sub> production in aqueous solutions containing Fe<sup>3+</sup> and Ce<sup>4+</sup> as electron acceptors. 269 Sr2+-doped CeO2 was an active photocatalyst for overall water splitting when RuO2 was loaded as a promoter.<sup>270</sup> Ce(III) oxide supported zeolites showed higher photocatalytic activity for pure water splitting.<sup>271</sup> Nonstoichiometric H2 and O2 evolution was observed. Photoirradiation of  $Ce^{3+}$  species generated electrons ( $Ce^{3+} + h\nu \rightarrow$ Ce<sup>4+</sup> + e<sup>-</sup>) that were captured effectively by a water molecule for the production of hydrogen. Yuan et al. reported that BaCeO<sub>3</sub> produced H<sub>2</sub> and O<sub>2</sub> from aqueous solutions containing CH<sub>3</sub>OH and AgNO<sub>3</sub> sacrificial reagents, respectively. It also showed some activity under UV irradiation for overall water splitting with the aid of RuO<sub>2</sub> loading.<sup>272</sup>

## 3.4. Nonoxide Photocatalysts

More than 25 years ago, efficient hydrogen production was achieved by irradiating suspensions of ZnS in  $SO_3^{2-}$  solutions under UV irradiation. The quantum yield was determined to be 90% at 313 nm. This was even without any noble metal as cocatalyst.<sup>273</sup> InP, the sole phosphide photocatalyst

developed so far, can also produce H<sub>2</sub> from water containing inorganic sacrificial reducing agents.<sup>274</sup> Aqueous sulfite solutions are particularly efficient.

For the past few years, there have been a few reports on nonoxide photocatalyts capable of decomposing pure water into H<sub>2</sub> and O<sub>2</sub> under UV irradiation. Maeda et al. found that the photocatalytic water-splitting activity of GaN is strongly dependent on the crystallinity of the material and the cocatalyst employed.<sup>275</sup> Modification of well-crystallized GaN with Rh<sub>2-v</sub>Cr<sub>v</sub>O<sub>3</sub> nanoparticles as a cocatalyst for H<sub>2</sub> evolution resulted in the stable stoichiometric decomposition of H<sub>2</sub>O into H<sub>2</sub> and O<sub>2</sub> under UV irradiation. RuO<sub>2</sub> modification, on the other hand, did not bring about appreciable H<sub>2</sub> and O<sub>2</sub> evolution. However, Zn<sup>2+</sup>, Mg<sup>2+</sup>, and Be<sup>2+</sup> doping of GaN converted it into a remarkably active and stable photocatalyst.<sup>276</sup> Again, the presence of RuO<sub>2</sub> as a cocatalyst was required.<sup>276</sup>

 $\beta$ -Ge<sub>3</sub>N<sub>4</sub> was another effective nitride photocatalyst to show efficient activity for splitting water into hydrogen and oxygen when combined with RuO2 nanoparticles reported by Domen's group.<sup>24,277,278</sup> The photocatalytic activity of RuO<sub>2</sub>-loaded  $\beta$ -Ge<sub>3</sub>N<sub>4</sub> was strongly dependent on the reaction conditions employed. The highest activity was obtained when the reaction was carried out in 1 M H<sub>2</sub>SO<sub>4</sub> aqueous solution. Moreover, treatment of as-prepared  $\beta$ -Ge<sub>3</sub>N<sub>4</sub> powder under high-pressure ammonia effectively increased the photocatalytic activity by up to 4 times. This was attributed to a decrease in the density of anion defects in the bulk and surface.<sup>279</sup> The AgBr/SiO<sub>2</sub> catalyst prepared from a Schumann emulsion showed a stable and high photocatalytic activity for H<sub>2</sub> generation from CH<sub>3</sub>OH/H<sub>2</sub>O solution under UV irradiation. The high activity of this AgBr/SiO<sub>2</sub> catalyst related to photogenerated Ag species, which could act as the site for H<sub>2</sub> formation.<sup>280</sup>

## 4. Approaches to Modifying the Electronic Band Structure for Visible-Light Harvesting

With a view to developing photocatalytic applications using visible-light irradiation, beginning with the photocatalytic splitting of water into H<sub>2</sub> and O<sub>2</sub> using TiO<sub>2</sub> under UV irradiation, <sup>18</sup> prophase studies were also carried out on some narrow band gap semiconductors such as CdS281,282 and WO<sub>3</sub>.<sup>283–285</sup> However, the fact that serious photocorrosion of CdS was observed in the photocatalytic reaction<sup>286–288</sup> and that the relatively positive conduction band of WO<sub>3</sub> proved dissatisfactory for hydrogen production<sup>284</sup> created the major impediments for the efficient performance of these two photocatalysts for use in visible-light-driven water splitting. Some studies were carried out to improve the photocatalytic stability of CdS, <sup>288–292</sup> whereas others used WO<sub>3</sub> as the photoelectrode in the photoelectrochemical cell to satisfy the energy requirement for water splitting with an applied potential.<sup>293-299</sup> To overcome these obstacles, many efforts has been made to develop new visible-light-driven photocatalysts with high water-splitting activities. These are shown in Table 2. On the basis of numerous experimental results from the past 20 years, several common approaches have been adopted in order to make photocatalysts visible-light active for water splitting into hydrogen and/or oxygen: (1) metal or/and nonmetal ions doping for band gap narrowing; (2) developing solid solutions to control the band structure; (3) dye sensitization to make UV-light-active photocatalysts harvest visible light; (4) developing novel single-phase Visactive photocatalysts through band gap engineering.

## 4.1. Metal and Nonmetal Doping

### 4.1.1. Metal Ion Doping

One of the most effective ways to develop visible-lightdriven photocatalysts is to create impurity levels in the forbidden band through metal ion doping. This makes the wide band gap photocatalysts active in the visible-light region, and this approach has been known for a long time. Over the past decades, there have been numerous reports on the modification of wide band gap photocatalysts using metal ion doping to make them visible-light active. These include doped TiO<sub>2</sub>, 300-303 doped SrTiO<sub>3</sub>, 303,304 doped La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub>,305 and doped ZnS, 306,307 among others. Figure 6 depicts the sketch of visible-light-driven photocatalysts by metal ion doping to create active photocatalysts with wide band gaps. In the forbidden band, either a donor level above the original valence band or an acceptor level below the original conduction band is created to make the photocatalysts respond to visible light.

As early as 1982, Borgarello et al. found that Cr5+-doped TiO<sub>2</sub> could produce hydrogen and oxygen via sustained water cleavage under visible-light (400-550 nm) irradiation.<sup>308</sup> Until now, many different metal ions have been doped into TiO<sub>2</sub> to improve the visible-light absorption and photocatalytic activities. These include ions such as V, Ni, Cr, Mo, Fe, Sn, Mn, and so on. 300-314 The electronic structures of TiO<sub>2</sub> compounds doped with the 3d transition metals (V, Cr, Mn, Fe, Co, and Ni) were analyzed by Umebayashi et al. using ab initio band calculations (Figure 7A).<sup>315</sup> They found that the 3d metal doping created an occupied level either in the band gap or in valence band (VB) due to the  $t_{2\,g}$  state of the dopant. The charge-transfer transition between this t<sub>2 g</sub> level and the conduction band (CB) (or VB) of TiO<sub>2</sub> contributed to the photoexcitation under visible light. Using  $DV-X_{\alpha}$  calculations, Nishikawa et al. demonstrated the possibility of shifing the absorption edge of titania to the sunlight region in the case of V-, Cr-, Mn-, Fe-, Co-, Ni-, or Rh-doping and then discussed the relationship between the ionic radius and the change of the band gap (Figure 7B)<sup>316</sup> and found that, of the cations addressed, Ni3+ and V5+ reduced the band gap most effectively. Cao et al. reported that Sn<sup>4+</sup>-doped TiO<sub>2</sub> nanoparticle films prepared by the chemical vapor deposition (CVD) method displayed a higher photocatalytic activity than pure TiO2 under both UV and visible light.<sup>317</sup> The visible-light absorption can be assigned to an electronic transition from the valence band to the doping energy level of the Sn<sup>4+</sup> ions. This was located 0.4 eV below the conduction band and acted as an electron acceptor level. In contrast, Klosek and Raftery demonstrated that the visible absorption in V<sup>4+</sup>-doped TiO<sub>2</sub> was indicative of the photoexcitation from the V 3d electron donor level to the TiO<sub>2</sub> conduction band.<sup>318</sup> This brings in a more efficient visible-light-driven activity for ethanol photooxidation over V<sup>4+</sup>-doped TiO<sub>2</sub> than than of pure TiO<sub>2</sub>. The visible-lightdriven photocatalytic activity of Fe<sup>3+</sup>-doped TiO<sub>2</sub> for water splitting was also attributed to the photoexcitation of doping the donor level (i.e., Fe 3d orbitals) to the TiO<sub>2</sub> conduction band.319-321

Anpo and co-workers prepared various metal-ion-doped TiO<sub>2</sub> materials using advanced ion implantation.<sup>322–327</sup> They found that the absorption band of metal-ion-implanted TiO<sub>2</sub> (the metals used were V, Cr, Mn, Fe, and Ni) shifted smoothly toward the visible-light region. The extent of the red shift depended on the amount and type of metal ions

Semiconductor photocatalysts for water splitting under visible-light irradiation

			•			activity $(\mu \text{mol} \cdot \text{h}^{-1} \cdot \text{g}^{-1})$	$ol \cdot h^{-1} \cdot g^{-1}$		
photocatalyst	synthetic method	mass (g)	light source	incident light	aqueous reaction solution	cocatal./H <sub>2</sub>	cocatal./O <sub>2</sub>	QY (%)	reference
TiO <sub>2</sub> :Cr	thermal hydrolysis	0.025	450-W Xe	>415 nm	pure water, $HCI(pH = 3)$	Pt-RuO <sub>2</sub> /268	Pt-RuO <sub>2</sub> /52		308
TiO <sub>2</sub> :Fe	hydrothermal method	0.5	300-W Xe	>400 nm	pure water	3.6			319
TiO <sub>2</sub> :Sb/Ni	solid-state reaction	0.5	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	0	25.6		303
TiO <sub>2</sub> :Sb/Cr	solid-state reaction	0.5	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/0.12	63		303
TiO <sub>2</sub> :Sb/Rh	solid-state reaction	0.3	300-W Xe	>440 nm	AgNO <sub>3</sub>		37	(2000)	332
De ionized TiO	Solid-State reaction	0.1	300 W Ye	7440 IIIII	recl <sub>3</sub>	91	~21.5 7.3	0.14 (433 mm)	337
reionized 1102 Ir-ionized TiO,	nydrothermal method	5.0	300-W Xe	×400 nm	pure water	2.54	1.27		338
Co-ionized TiO,	hydrothermal method	0.5	300-W Xe	×400 nm	pure water	1.34	0.67		338
TiO <sub>2</sub> :Ti <sup>3+</sup>	radiofrequency magnetron		500-W Xe	>420 nm	methanol/AgNO3	Pt/0.09	0.5		340
	sputtering deposition				)				
InTaO <sub>4</sub> :Ni	solid-state reaction	0.5	300-W Xe	>420 nm	pure water	NiO <sub>y</sub> /33.2	NiO <sub>y</sub> /16.6	0.66 (402 nm)	353
$BiYWO_6$	solid-state reaction	0.3	500-W Xe	>420 nm	pure water	RuO <sub>2</sub> /13.7	$RuO_2/6.3$	0.17 (420 nm)	559
$\mathrm{Bi_{0.5}Dy_{0.5}VO_4}$	solid-state reaction	0.2	300-W Xe	>420 nm	pure water	Pt-Cr <sub>2</sub> O <sub>3</sub> /18.2	$Pt-Cr_2O_3/9.6$		560
$LiCr(WO_4)_2$	polymer complex method	0.5/0.3	300-W Xe	=420 nm/>420 nm	methanol/AgNO <sub>3</sub>	Pt/2.4	4.17		099
NiNb <sub>2</sub> O <sub>6</sub>	solid-state reaction	0.5	300-w Xe	>420 nm	pure water	~0.30			199
	solid-state reaction	C.O	300-W Xe	>420 nm	pure water	~0.4 V:O.4			100
INVO4	solid-state reaction	0.5 5.0	300-w Xe	>420 nm	pure water	C/OIN			908
IIIN BO4	Solid-state reaction	C.O	300-W Ae	2420 nm	pure water	NIO,/3.3			668, 691
CoCo. NB. O.	solid state reaction	C.O.O.	300 W Ye	2420 mm	pure water	NiO,/4.0			680 681
STO NEO	solid-state reaction	0.5	300-W Xe	>420 nm	pure water	NiO / 1 72			680,681
BaCo. Nh. O.	solid-state reaction	5.0	300-W Xe	>420 nm	pare water	NiO.72.74			680, 681
SrTiO <sub>3</sub> :Sb/Cr	solid-state reaction	0.5	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/156	8.		303
SrTiO <sub>3</sub> :Ru	solid-state reaction	0.3	300-W Xe	×440 nm	methanol/AsNO;	Pt/5.7	2		344
InTaO <sub>4</sub> :Ni	solid-state reaction	0.5	300-W Xe	>420 nm	pure water	33.2	16.6	0.66 (402 nm)	353-355
BiTaO4:Cu	solid-state reaction	0.1	350-W Xe	>400 nm	methanol	$RuO_2/878$			359
NaNbO <sub>3</sub> :Ir/Sr	solid-state reaction	0.3	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/36.7	16		361
NaTaO <sub>3</sub> :Ir/La	solid-state reaction	0.3	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/36.7	0		361
$\text{Bi}_{x}\text{In}_{1-x}\text{TaO}_4$	solid-state reaction	0.5	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/2.14	24.02		552
AgNbO <sub>3</sub>	solid-state reaction	0.3	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/1.7	123.3		242
BaCr <sub>2</sub> O <sub>4</sub>	solid-state reaction	0.5	300-W Xe	>540 nm/>420 nm	methanol/Ce(SO <sub>4</sub> ) <sub>2</sub>	$Pt/\sim 10$	7.5	0000	658
PbB14114O <sub>15</sub>	solid-state reaction	0.3	450-W Xe	>400 nm	methanol/AgNO <sub>3</sub>	Pt/37.3	1443.3	0.27/21 (>400 nm)	707
$PbB_{12}Nb_2O_9$	solid-state reaction	0.3	450-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/25.3	1/33.3	0.95/29 (>420 nm)	707, 708
Span Span Span Span Span Span Span Span	solid state reaction	0.3	300 W Ve	2400 mm	methanol/AgNO.	FU45.3 D+/60	0.743.3	0.33/20 (>400 IIIII)	743 700
Sin N2 Co	solid-state reaction	0.3	300-W Xe	>420 nm	methanol/AgNO3	Dr/48	IrO <sub>2</sub> /2003		710
$In_2O_3(ZnO)_0$	solid-state reaction	1	300-W Xe	>420 nm	methanol/AgNO3	Pt/1.1	1.3		724
$H(Rb)Pb_2Nb_3O_{10}$	solid-state reaction,	1	500-W Xe	>420 nm	methanol/AgNO3	Pt/24	1.1	5/(420-500 nm)	656
	ion-exchange					:			
${ m Sn}^{2+}/{ m K_4Nb_6O_{17}}$	solid-state reaction, ion-exchange	0.2	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/115	20		711
Sn='/KINBOS Sn2+/CcTi-NBO-	solid-state reaction, ion-exchange	0.7	300-w Ae	>420 nm	methanol/AgNO3	Pt/2/U			711
Sn /Cs 1121007 Sn2+/K, Ti .O.	solid-state reaction ion-exchange	2.0	300-W Xe	7420 nm	methanol/AgNO3	FU20 Pt/115			711
Sn <sup>2+</sup> /K,Ti,O <sub>5</sub>	solid-state reaction, ion-exchange	0.2	300-W Xe	×420 nm	methanol/AgNO3	Pt/25	0 0		711
$Sn^{2+}/C_{5}$ TikO <sub>13</sub>	solid-state reaction, ion-exchange	0.2	300-W Xe	>420 nm	methanol/AgNO3	Pt/35	0		711
K <sub>0.5</sub> La <sub>0.5</sub> Ca <sub>0.75</sub> Pb <sub>0.75</sub>	solid-state reaction	0.3	450-W Xe	>400 nm	methanol/AgNO <sub>3</sub>	Pt/traces	413.3	-/6(>400 nm)	707
Nb <sub>3</sub> O <sub>10</sub>	1000		450 W V.	400	ON- Wilson die	7,7	023	( 400	100
No.5La <sub>0.25</sub> B <sub>10.25</sub> Ca <sub>0.75</sub> P <sub>0.75</sub> Nb <sub>2</sub> O <sub>10</sub>	solid-state reaction	0.3	450-w Ae	>400 nm	metnanol/AgiNO <sub>3</sub>	Pt/traces	000	-/8(>400 nm)	/0/
Ca <sub>2</sub> NiWO <sub>6</sub>	solid-state reaction	0.5	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/0	0.7		664
BaCrO₄	solid-state reaction	0.5	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/1.34	0		657
$SrCrO_4$	solid-state reaction	0.5	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/0.18	0		657
SrTiO <sub>3</sub> :Rh	solid-state reaction	0.3	300-W Xe	>440 nm	methanol	Pt/390		5.2 (420 nm)	344
SrI 1O3:Ir	solid-state reaction	0.3	300-w Xe	>440 nm	methanol	Pt/28.7			344

		reference	344	346	347	348	178, 225	305, 349	305, 349	350	350	358	360	362	351	140	252	220	700	002, 000	667	251	451	714	714	715	716	284, 929	285	554			675	675	929	0/0	677	685	289	289	363	145	00/	VV.	717 684	336	829	629	683	481	986	717
		QY (%)																		(00000000000000000000000000000000000000	(11111 014) 6:00								19 (420 nm)	10 (420 nm)	0.3 (420 IIIII) 16 4 (420 4 nm)	9 (450 nm)			0.23 (420 nm)	0.04 (420 nm)			1.9 (420 nm)	1.3 (420 nm)	6 (420 nm)					0.14 (435 nm)	1.8 (420 nm)	0.76 (420 nm)	,			80 (480 nm)
	activity $(\mu \text{mol} \cdot \text{h}^{-1} \cdot \text{g}^{-1})$	cocatal./O <sub>2</sub>	6													2000	$VV \sim 220$			1073 3	10/3.3						3.52	$RuO_2/67$	490	201.4	324 324	478	$\sim 12.6$	~5.4	7.2	105 9	30	7.2	165	120	143	3.0	110	7.01	26.7	7.8.7	13	2.3	,			1272
	activity (µ	cocatal./H <sub>2</sub>		Pt/28.3	$NiO_x/22.7$	Pt/7	Pt/0.74	Pt/30	Pt/20	$\sim 0.27$	~0.06	144	58.1	Pt/4.4	KuO <sub>2</sub> /42.2	KuO <sub>2</sub> /55.5	FU~30	C//OIN	PU/0.0 D±/16.4	FU10.4	D+/8 68	0.00	0.0	NiO./337.5	NiO./104	NiO,/4.6	Pt/3.75																	D+/1 8	FV1.0				NiO/1565	Pt/7	17.03	
Semiconductor photocatalysts for water splitting under visible-light irradiation		aqueous reaction solution	AgNO <sub>3</sub>	methanol	methanol	methanol	methanol	methanol	methanol	Na <sub>2</sub> S/Na <sub>2</sub> SO <sub>3</sub>	Na <sub>2</sub> S/Na <sub>2</sub> SO <sub>3</sub>	methanol	methanol	methanol	KI	KI	pure water	memanol	methanol	AGNO	AgivO <sub>3</sub>	memanor	purc water	Na <sub>3</sub> SO <sub>3</sub>	Na <sub>3</sub> SO <sub>3</sub>	pure water	methanol/AgNO <sub>3</sub>	$AgNO_3$	Fe <sup>3+</sup>	AgNO <sub>3</sub>	AgNO3	AgNO	AgNO <sub>3</sub>	$AgNO_3$	AgNO <sub>3</sub>	AgNO <sub>3</sub>	AgNO3	AgNO <sub>3</sub>	$AgNO_3$	$AgNO_3$	$AgNO_3$	AgNO <sub>3</sub>	AginO <sub>3</sub>	AginO <sub>3</sub>	memanol A o'NO;	FeCl <sub>2</sub> + H <sub>2</sub> SO <sub>4</sub> (pH = 2.4)	Fe <sup>3+</sup>	$Fe^{3+}$	$CO_2/H_2O$	methanol	pure water	$AgNO_3$
for water splitting		incident light	>440 nm	>420 nm	>400 nm	>420 nm	>420 nm	>420 nm	>420 nm	visible light	visible light	400-800 nm	>400 nm	>420 nm			400	7400 nm	>420 nm	2420 IIIII	2400 IIIII	390—520 nm	390—520 nm	>420 nm	>420 nm	>400 nm	>400 nm	>410 nm	>420 nm	>420 nm	>420 mm	>420 nm	>420 nm	>420 nm	>420 nm	2420 nm	>420 mm	>420 nm	>420 nm	>420 nm	>420 nm	>420 nm	>420 nm	2420 nm	>400 nm >420 nm	7420 mm	>420 nm	>420 nm	visible light	>420 nm	>420 nm	>400 nm
r photocatalysts		light source	300-W Xe	300-W Xe	350-W Xe	300-W Xe	300-W Xe	500-W Hg	500-W Hg			500-W H	300-W Xe	300-W Xe	250-w Xe	250-W Xe	300-W H	350-W Ae	300-W Ae	200-W Ac	200-W Ac	500-W Xe	500-W Xe	300-W Xe	300-W Xe	250-W Xe	250-W Xe	900-W Xe	300-W Xe	300-W Xe	300-W Xe	300-W Xe	300-W Xe	300-W Xe	300-W Xe	300-W Xe	300-W Xe	300-W Xe	300-W Xe	300-W Xe	300-W Xe	300-w Xe	300-W Xe	300-W Xe	300-W Xe	500-W Xe	300-W Xe	300-W Xe	150-W Xe	300-W Xe	350-W Xe	300-W Xe
Semiconducto		mass (g)	0.3	0.3	0.1	0.3	0.5	0.5	0.5	0.1	0.1	0.2	0.1	0.5 -	٠.	¢	0.1	0.1	C.O.	5000	0.045	; I	I	0.1	0.1		1	0.2	0.4	0.5 5.0	C.O.O.		0.5	0.5	0.5	0.0	0.5	0.5	0.2	0.2	0.5	- ·	C.O	U.O 2. C	0.3	0.3	0.5	0.8	0.1	0.2	0.1	0.5
		synthetic method	solid-state reaction	solid-state reaction	sol—gel method	solid-state reaction	solid-state reaction	citrate complex method	citrate complex method	solid-state reaction	solid-state reaction	solid-state reaction	sol—gel method	solid-state reaction	sol—gel method	sol—gel method	Solid-state reaction	nydrotnermal method	Solid-state reaction	Sourcestate Teachon	copiecipitation memor	solid-state reaction	solid-state reaction	solid-state reaction	solid-state reaction	solid-state reaction	solid-state reaction	Puratronic WO <sub>3</sub>	ion-exchange method	solid-state reaction	solid-state reaction	precipitation	solid-state reaction	solid-state reaction	solid-state reaction	solid-state reaction	solid-state reaction	solid-state reaction	molten reaction	molten reaction	reflux method	solid-state reaction	reflux method	SOlid-State reaction	solid-state reaction	impregnation and calcination	solid-state reaction	solid-state reaction	sol-gel method	sol-gel method	hydrothermal method	ion-exchange method
		photocatalyst	SrTiO <sub>3</sub> :Mn	CaTiO <sub>3</sub> :Rh	CaTiO <sub>3</sub> :Cu	$Ca_3Ti_2O_7$ :Rh	$BaNi_{1/3}Nb_{2/3}O_3$	$\text{La}_2\text{Ti}_2\text{O}_7$ :Cr	$\text{La}_2\text{Ti}_2\text{O}_7$ :Fe	K <sub>2</sub> La <sub>2</sub> Ti <sub>3</sub> O <sub>10</sub> :Fe	K <sub>2</sub> La <sub>2</sub> Ti <sub>3</sub> O <sub>10</sub> :W	$K_4Nb_6O_{17}$ :Ni	Bi <sub>4</sub> Ti <sub>3</sub> O <sub>12</sub> :Cr	NaTaO <sub>3</sub> :La/Cr	K <sub>2</sub> La <sub>2</sub> I <sub>13</sub> O <sub>10</sub> : V	K <sub>2</sub> La <sub>2</sub> 11 <sub>3</sub> O <sub>10</sub> :Zn	III21105:V	Na(Bl <sub>x</sub>   a <sub>1-x</sub> )O <sub>3</sub>	CSLaSTIND2INIO9	Interviole 1110042	Mar I a Ta Co O	Nd-7r-0-	Sm.Zr.O.	K,Ce,TanOm	K-Ce-Nb <sub>10</sub> O <sub>30</sub>	Sm <sub>2</sub> InTaO <sub>7</sub>	Sm <sub>2</sub> InNbO <sub>7</sub>	$\mathrm{WO}_3$	$C_{S}-WO_{3}$	$Ca_{1-x}B_{1x}V_xMo_{1-x}O_4$	(A oNbO 2) 2 (SrTiO) 2 2	BiVO4	$Z_{\rm n_2,5}V{ m MoO_8}$	${ m Mg_{2.5}VMoO_8}$	$ m Mg_3V_2O_8$	Zn3V2O8 CaBiVMcO.	CaBivWO	AgInW <sub>2</sub> O <sub>8</sub>	$AgLi_{1/3}Ti_{2/3}O_2$	AgLi <sub>1/3</sub> Sn <sub>2/3</sub> O <sub>2</sub>	PbMoO <sub>4</sub> :Cr	Bi <sub>2</sub> WO <sub>6</sub>	Bi-Mo-O.	BI2MIO3O12	P-Shw O <sub>4</sub> Ag, VO,	Agg V O4 TiO::Rh	BiCu <sub>2</sub> VO <sub>6</sub>	BiZn <sub>2</sub> VO <sub>6</sub>	$Ca_2Fe_2O_5$	Nb2O5:C	$ZnFe_2O_4$	$\mathrm{Ag}_{3}\mathrm{PO}_{4}$

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		Semicon	ductor photocataly	sts for water splitti	Semiconductor photocatalysts for water splitting under visible-light irradiation	nt irradiation			
						activity $(\mu \text{mol} \cdot \text{h}^{-1} \cdot \text{g}^{-1})$	$(1 \cdot h^{-1} \cdot g^{-1})$		
photocatalyst	synthetic method	mass (g)	light source	incident light	aqueous reaction solution	cocatal./H <sub>2</sub>	cocatal./O <sub>2</sub>	QY (%)	reference
CdS	two-step precipitation	0.15	300-W Xe	>420 nm	$Na_2S + Na_2SO_3$	Pv27333		60 (420 nm)	281, 728, 730,
CdS CdS	precipitation precipitation and hydrothermal	0.1	300-W Xe 300-W Xe	>420 nm >420 nm	lactic acid $Na_2S + Na_2SO_3$	MoS <sub>2</sub> /5400 Pt-PdS/29 233		93 (420 nm)	782 782 733
CdS CdSe	precipitation low-temperature solution-	0.1	500-W Hg 700-W Hg	>420 nm >400 nm	Na2S + Na2SO3 $Na2S + Na2SO3$	WC/~1350 436	I	13.4 (>400 nm)	786 974
CdS:Ag	pnase synthesis precipitation hydrothermal method	0.4	$900-W Xe$ $300-W W-H^b$	>420 nm	$Na_2S + Na_2SO_3$ $Na_3S + K_3SO_3$	Pt/33480 Pt/2456		~25 (450 nm)	282
CdS:Mn CdS-ZnS	hydrothermal method precipitation	0.2	500-W Xe 300-W Hg	>420 nm >400 nm	$Na_{2}S + Na_{2}SO_{3}$ $Na_{2}S + Na_{2}SO_{3}$ $Na_{2}S + Na_{2}SO_{3}$	RuO <sub>x</sub> /1935 2007		7 (>420 nm) 0.60 (>400 nm)	367 373, 529,
CdS-ZnS	H <sub>2</sub> S thermal	0.2	350-W Xe	>430 nm	$\mathrm{Na_2S} + \mathrm{Na_2SO_3}$	006		10.2 (420 nm)	536 536
Cd <sub>0.1</sub> Zn <sub>0.9</sub> S:Ni	Summization hydrothermal method	0.2	350-W Xe	>420 nm	$Na_2S + Na_2SO_3$	Pt/585.5		15.9 (420 nm)	374
Cd <sub>0.1</sub> Cu <sub>0.01</sub> Zn <sub>0.89</sub> S	precipitation	0.3	350-W Xe	>430 nm	$Na_2S + Na_2SO_3$	1166.7 D+/2632.3		9.2 (420 nm)	368
(Zn0.95Cu0.05/0.67Cu0.332) CdS-ZnS:Ag	precipitation precipitation	0.5	900-W Xe	>420 IIIII	$Na_2S + Na_2SO_3 + Na_2SO_3$	Pt/40957.5		37 (450 nm)	282
ZnS:Cu	precipitation		300-W Xe	>420 nm	Na <sub>2</sub> SO <sub>3</sub>	450		3.7 (420 nm)	306
ZnS:Nı ZnS:Ni	precipitation ultrasonic spray	1 0.1	300-w Xe 300-w Xe	>420 nm >400 nm	$Na_2S + K_2SO_3 + K_2SO_3 + Na_2S$	280 160		1.3 (420 nm) 2.1 (430 nm)	307 365
ZnS:Pb/Cl	pyrolysis precipitation and	1	300-W Xe	>420 nm	$\mathrm{K}_{2}\mathrm{SO}_{3}+\mathrm{Na}_{2}\mathrm{S}$	93			527
ZnS:C	solvothermal and	0.1	500-W Hg	>420 nm	$Na_2S + Na_2SO_3$	06∼/1d			501
$\ln_2 S_3$	calcination hydrothermal method	0.05	300-W Xe	>400 nm	$Na_2S + Na_2SO_3$	Pd/960.2			891
$AgInZn_7S_9$	precipitation and calcination	0.3	300-W Xe	>420 nm	$Na_2S + Na_2SO_3$	Pt/3164.7		20 (420 nm)	537
$Cu_{0.09}In_{0.09}Zn_{1.82}S_2$	precipitation and	0.3	300-W Xe	>420 nm	$Na_2S + Na_2SO_3$	Pt/4090		12.5 (420 nm)	538
$Cu_{0.25}Ag_{0.25}In_{0.5}ZnS_2$	precipitation and calcination	0.3	300-W Xe	>420 nm	$Na_2S + Na_2SO_3$	Ru/7666.7		7.4 (520 nm)	540, 541
ZnIn <sub>2</sub> S <sub>4</sub>	nyuromerma memod surfactant-assisted hydrothermal	0.2	300-W Xe	>420 mm	$Na_2S + Na_2SO_3$ $Na_2S + Na_2SO_3$	FU231 PU562		18.4 (420 nm)	377, 915
ZnIn <sub>2</sub> S <sub>4</sub> :Cu	inemod surfactant-assisted hydrothermal method	0.2	300-W Xe	>430 nm	$Na_2S + Na_2SO_3$	Pt/757.5		14.2 (420 nm)	378
AgGaS <sub>2</sub>	solid-state reaction	0.1	500-W Hg	>420 nm	$Na_2S + Na_2SO_3$ $Na_2S + Na_2SO_3$	Pt/2960 Nis/~2800		12.4 (>420 nm)	496, 873
Ag2ZnSnS <sub>4</sub> Cu2ZnGeS <sub>4</sub>	precipitation and calcination precipitation and	0.3	300-W Xe 300-W Xe	>420 nm >420 nm	$Na_2S + K_2SO_3  Na_2S + K_2SO_3  Na_2S + K_2SO_3$	Ru/1607 Ru/1233		3 (500 nm)	542 542
CuGo. In. C.	calcination	03	300 W Ve	~ 02V	$N_{2}$ . $N + N_{2}$ .	Pb/10667		15 (560 nm)	5/3
AgGa2In3S8	solid-state reaction	0.3	300-W Xe	>420 mm	Na <sub>2</sub> S + K <sub>2</sub> SO <sub>3</sub>	Rh/3433		15 (490 nm)	543
${ m AgGa_{0.9}In_{0.1}S_2} \ { m ZnS-In_2S_3-Ag_2S}$	solid-state reaction hydrothermal method	0.1	500-W Hg 300-W Xe	>420 nm >400 nm	$Na_2S + Na_2SO_3$ $Na_2S + Na_2SO_3$	Pt/3500 220 000		19.8 (420 nm)	544 545
ZnS-In <sub>2</sub> S <sub>3</sub> -CuS NaInS,	hydrothermal method prepicipation and	0.01	300-W Xe 300-W Xe	>400 nm >420 nm	$Na_2S + Na_2SO_3$ $Na_2SO_3$	360 000 Pv671.4		22.6 (420 nm) 6 (440 nm)	546 500
S. ale	calcination	3 0	050 W Vs	.430	пол + з п	0909		17.1 (500 nm)	0101
$\operatorname{Cdm}_2 S_4 \\ \operatorname{AgIn}_5 S_8 \\ \widetilde{G}$	nydrounermal method precipitation, calcination	0.3	300-W Xe	>420 nm >420 nm	$R_2S + KOH$ $K_2SO_3 + Na_2S$	0900 PV200		5.3 (411.2 nm)	497
Culn5 <sub>2</sub> Culn <sub>5</sub> S <sub>8</sub>	solvothermal method precipitation, calcination	0.25	500-w Xe 400-W Xe	>420 nm	$Na_2S + Na_2SO_3$ $Na_2SO_3$	$_{ m Pt}^{ m VW84}$		<0.0002 (460 nm)	492, 493 492

Table 2. Continued

		Semi	conductor photo	catalysts for wa	Semiconductor photocatalysts for water splitting under visible-light irradiation	ole-light irradiatio	u		
						activity (µmol•h <sup>-1</sup> •g <sup>-1</sup> )	$\operatorname{ol} \cdot h^{-1} \cdot \mathrm{g}^{-1}$		
photocatalyst	synthetic method	mass (g)	light source	incident light	aqueous reaction solution	cocatal./H <sub>2</sub>	cocatal./O <sub>2</sub>	QY (%)	reference
Na <sub>14</sub> In <sub>17</sub> Cu <sub>3</sub> S <sub>35</sub> •xH <sub>2</sub> O Cu <sub>8</sub> Ge <sub>5</sub> S <sub>16</sub> <sup>4-</sup>	hydrothermal method solvothermal method	0.5	300-W Xe 300-W Xe	>420 nm >400 nm	$Na_2S$ $Na_2SO_3$	18 2.64		3.7 (420 nm)	954 956
TiO <sub>2</sub> :S	precursor heat-treatment	0.3	500-W Xe	>420 nm	ethanol	Pt/~0.5		(30,000)	440
In(OH)3:3/Zn Sm <sub>2</sub> Ti <sub>2</sub> S <sub>2</sub> O <sub>5</sub>	hydromermal method $H_2S$ thermal sulfurization or solid-state	0.3	300-W Xe	>420 mm >440 mm	$Na_2S + Na_2SO_3$ $Na_2S + Na_2SO_3/AgNO_3$	PU223.3 Pt/110	110	0.39 (420 nm) 0.3/0.7 (440– 650 nm)	228 485, 486, 488
	or some-state reaction							0.50 11111)	
$\mathrm{Gd}_{2}\mathrm{Ti}_{2}\mathrm{S}_{2}\mathrm{O}_{5}$	H <sub>2</sub> S thermal sulfurization	0.2	300-W Xe	>440 nm	$Na_2S + Na_2SO_3/AgNO_3$	Pt/105	120	0.1/— (440—650 nm)	488
La—In oxysulfide	solid-state reaction	0.1	300-W Xe	>420 nm	$Na_2S + Na_2SO_3/AgNO_3$	Pt/~85	$IrO_2/\sim 26$	0.2/0.1 (420–600 nm)	489, 490
La—Ga oxysulfide	solid-state reaction	0.1	300-W Xe	>420 nm	Na <sub>2</sub> S + Na <sub>2</sub> SO <sub>3</sub> /AgNO <sub>3</sub>	Pt/1080	$IrO_2/\sim 10$	2.5/0.2 (420-600 nm)	490
La <sub>10</sub> OS <sub>14</sub> La <sub>4</sub> In <sub>5</sub> S <sub>13</sub>	H <sub>2</sub> S thermal sulfurization H <sub>2</sub> S thermal sulfurization	0.2	500-W Xe	>500 nm	$Na_2S + Na_2SO_3$ $Na_2S + Na_2SO_3$	Ku/0.45 Pr/0.5			491 491
$\operatorname{ZnS}_{1-x-0.5v}\operatorname{O}_x(\operatorname{OH})_v$	coprecipitation, calcination	0.1	400-W H		$Na_2S + Na_2SO_3$	~460		3.0 (400-700 nm)	547
ZrW <sub>2</sub> O <sub>8</sub> :S	hydrothermal method	0.1	300-W Hg-Xe		AgNO <sub>3</sub>		$\sim$ 50		484
$(Ga_{0.88}Zn_{0.12})(N_{0.88}O_{0.12})$	NH <sub>3</sub> nitridation, postcalcination	0.3	450-W Hg	>400 nm	pure water + $H_2SO_4$ (nH = 4.5)	Rh <sub>2-y</sub> Cr <sub>y</sub> O <sub>3</sub> /2800	Rh <sub>2-y</sub> Cr <sub>y</sub> O <sub>3</sub> /1400	5.9 (420–440 nm)	563, 564, 567, 569—571, 575, 767—770, 773
$Zn_{1.6}GeN_{1.7}O_{0.5}$	NH3 nitridation,	0.1	450-W Hg	>400 nm	pure water	Rh <sub>2-y</sub> Cr <sub>y</sub> O <sub>3</sub> /3160	Rh <sub>2-y</sub> Cr <sub>y</sub> O <sub>3</sub> /1580	2.0 (420-440 nm)	577-580
NO	postcalcination	5	200 W V.	2420	OM Allowedto	0.700	0271	(2005 000 000 000 000 000 000 000 000 000	072 057 057 002
$Ta_3N_5$	NH3 muraauon NH3 nitridation	0.2	300-W Xe	>420 mm	methanol/AgNO3	Pt/~125	2100	-/10 (420-600 nm)	390, 470, 894
$LaTiO_2N$	NH <sub>3</sub> nitridation	0.2	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/~15	51	0.15/1.5	475, 476
Sr <sub>2</sub> Nb <sub>2</sub> O <sub>2</sub> :N	NH <sub>2</sub> nitridation	0.1	200-W Xe	>400 nm	methanol/AgNO <sub>2</sub>	Pt/93	trace	(420-600 nm)	399
$ m Y_2Ta_2O_5N_2$	NH <sub>3</sub> nitridation	0.3	300-W Xe	>420 nm	ethanol/AgNO3	Pt-Ru/833	467		393
$Zr_2ON_2$	NH <sub>3</sub> nitridation	0.2/0.05	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Pt/7	46		396
$Sn_{x}TO_{y}N_{z}$ (SrTiO <sub>3</sub> ) <sub>1-x</sub> •	NH3 nitridation NH3 nitridation	0.2	300-w Xe 300-w Xe	>420 nm >420 nm	methanol/AgNO <sub>3</sub> methanol/AgNO <sub>3</sub>	Pt/0.4 Pt/66.7	400 53.3		482 562
$(LaTiO_2N)_x$									
Ga-Zn-In mixed oxynitride	$\mathrm{NH}_3$ nitridation	0.2	450-W Hg	>400 nm	methanol/AgNO <sub>3</sub>	Rh/50	$CoO_x/\sim 100$		581
$CaTaO_2N$	NH <sub>3</sub> nitridation	0.2	300-W Xe	>420 nm	methanol	Pt/75			42
SrTaO <sub>2</sub> N	NH <sub>3</sub> nitridation	0.5	300-W Xe	>420 nm	methanol	Pt/100 Pt/75			42
Da1aO <sub>2</sub> IN I aTaON,	NH3 mundauon NH, nitridation	7.0	300-w Xe	>420 IIIII	memanor	Pt-R 1176 7			4 <i>2</i> 39 <i>4</i>
$Nb_2Zr_6O_{17-x}N_x$	NH, nitridation	0.5	450-W Xe	>420 nm	H <sub>2</sub> S+KOH	8566		13.5 (550 nm)	398
N-doped Zn-Ga	solid-state	0.02	125-W Hg	visible light	methanol	Rh-Cr <sub>2</sub> O <sub>3</sub> /37202		5.1%	483
mixed oxide Zr.TaiOi+.Ni	reaction NH2 nitridation	0.4	300-W Xe	>420 nm	methanol	$Ru/\sim 250$			868
TiN film	cathodic arc technique		300-W Xe	>400 nm	Na <sub>2</sub> S	$Pt/\sim 150^a$			731
TiO <sub>2</sub> :N	precursor heat-treatment	0.1	300-W Xe	>400 nm	$Na_2SO_3$	$Pt/\sim 250$			430, 431
TiO <sub>2</sub> :N	NH <sub>3</sub> nitridation	1.0	450-W W <sup>c</sup>	>420 nm	AgNO <sub>3</sub>	3	221	14 (>420 nm)	708
TiO <sub>2</sub> :B/N	sol—gel method	0.15	400-W H 300 W Ye	>420 nm	EDIA-2Na	Pt/34.9	~420		465 460—462
TiO,:S/N	precursor intribution thermal decomposition	0.35	300-W Xe	>400 nm	methanol	Pt/9.97	074		463
TiO <sub>2</sub> :In/N	polyol method	0.05	288-W F	quartz filter	methanol	Pd/75		0.4 (400 nm)	525
Ca <sub>0.25</sub> La <sub>0.75</sub>	NH <sub>3</sub> nitridation	0.2	300-W Xe	>420 nm	$AgNO_3$		$IrO_2/500$	5 (420–600 nm)	475
TiSi <sub>2</sub>	Alfa	2	Н	310-800 nm	pure water	~5	~2.5		719
graphitic C <sub>3</sub> N <sub>4</sub>	thermal condensation	0.1	300-W Xe	>420 nm	TEA/AgNO <sub>3</sub>	Pt/107	$RuO_2/\sim 8.3$	0.1/—(420—460 nm)	720, 763
$C_3N_{4-x}S_x$ harbituric acid modified $C_2N_x$	H <sub>2</sub> S thermal sulfurization conolymerization	0.1	300-W Xe 500-W HBO	>420 nm >420 nm	TEA TEA	Pt/~750 Pt/294			721 722
			:						

Table 2. Continued

		Semico	nductor photocata	llysts for water	Semiconductor photocatalysts for water splitting under visible-light irradiation	ht irradiation			
						activity $(\mu \text{mol} \cdot \text{h}^{-1} \cdot \text{g}^{-1})$	$(1 \cdot h^{-1} \cdot g^{-1})$		
photocatalyst	synthetic method	mass (g)	light source	incident light	aqueous reaction solution	cocatal./H <sub>2</sub>	cocatal./O <sub>2</sub>	QY (%)	reference
graphite oxide	modified Hummers'	0.5	400-W Hg	>400 nm	methanol	4		0.01	723
mesoporous graphitic $C_3N_4$	silica templated self- polymerization	0.1	500-W Hg	>420 nm	TEA	Pt/1490			952
$Cr-Ba_2In_2O_5/Cr-In_2O_3$	solid-state reaction	0.5	300-W Xe	>420 nm	methanol/AgNO3	Pt/15.8	0.7		819
$\mathrm{Ba_2 In_2 O_3 / In_2 O_3}$ $\mathrm{WO_3 / W/PhB_1 ^2 Nh_1 ^0 Ti_{0.1} O_0}$	coprecipitation method solid-state reaction chemical	0.3	300-w Xe 450-W Xe	>420 nm >420 nm	methanol methanol/A oNO <sub>2</sub>	Pt/16.4 Pt/49.3	741	6.06/41 (>420 nm)	820
60 1:01 - 6:10 - 7:70 - 1:11 - 6:0	vapor deposition				20110	2	-		
TaON/ZrO <sub>2</sub>	NH <sub>3</sub> nitridation	0.4	300-W Xe	>420 nm	methanol/AgNO <sub>3</sub>	Ru/~675	~312.5		897
n1 livb(1a)O5/Fe2O3	stepwise intercalation reaction	0.1	300-w ng	>470 IIIII	meuianon/AgivO <sub>3</sub>	uaces	0112		629
$HTaWO_d(Pt, TiO_2)$	stepwise intercalation	1	450-W Hg	>400 nm	methanol	029~		0.14 (>400 nm)	827
CdS/TiO <sub>2</sub>	precipitation, sol—	0.1	350-W Hg	>420 nm	$Na_2S + Na_2SO_3$	Pt/6400			803-805, 816
Bi,S,/TiO,	hydrothermal method		M M-009		Na <sub>2</sub> S <sub>2</sub> O <sub>3</sub>	2900			822
CdS/ZnO	two-step precipitation	0.2	300-W Xe		$Na_2S + Na_2SO_3$	Pt/3870		3.2 (300-600 nm)	797, 798
CdS/LaMnO <sub>3</sub>	reverse micelle method	0.1	300-W Xe	>420 nm	$Na_2S + Na_2SO_3 + NaOH$	375			799
CdS/KNbO <sub>3</sub>	ion adsorption, precipitation	0.2	500-W Hg-Xe	>400 nm	isopropanol	NiO <sub>x</sub> /203.5		8.8 (>400 nm)	800, 801
$TiO_{2-x}N_x/WO_3$	sol—gel	0.05	300-W Hg	>400 nm	$Na_2S + Na_2SO_3$	Pd/1005		0.45 (>400 nm)	817, 818
MWNT/TiO <sub>2</sub>	chemical vapor deposition	1005	150-W H 500 W H <sub>2</sub> -V <sub>2</sub>	380-780 nm	methanol	Ni/38.1 13 360			821
CdSe/CdS	unect deposition colloidal synthesis	0.00	300-W ng-Ae	7400 IIIII	$10a_2 3 + 10a_2 3 O_3 + 10aOH$ methanol	13 360 Pt/40 000		20 (450 nm)	824 824
$HLaNb_2O_7/(Pt, Fe_2O_3)$	ion-exchange,	1	100-W Hg	>400 nm	methanol	$\sim$ 372			838
i H	photodeposition	-	24 /41 000						0 0 0
$CdS/K_2Ti_{3.9}Nb_{0.1}O_9$	cation exchange, sulfurization	0.1	300-w Xe		$Na_2S$	Pt/~/31			840—842
(Pt,Cd <sub>0.8</sub> Zn <sub>0.2</sub> S)/HNbWO <sub>6</sub>	cation exchange, sulfurization	1	450-W Hg	>400 nm	$Na_2S$	~2678			843
$(Pt,Cd_{0.8}Zn_{0.2}S)/HLaNb_2O_7$	cation exchange, sulfurization	_	100-W Hg	>400 nm	$Na_2S$	$\sim 892.8$			845
CdS/ETS-4	ion exchange, sulfurization	0.1	300-W Xe	>420 nm	$Na_2S + Na_2SO_3 + NaOH$	Pt/175			847
CdS/ETS-10	ion exchange, sulfurization	0.1	300-W Xe	>420 nm	$Na_2S + Na_2SO_3 + NaOH$	Pt/133			847
CdS/11-MCM-41	ion exchange, sulfurization	0.7	350-W Xe	>430 nm	Na <sub>2</sub> SO <sub>3</sub>	Pt//6		2.6 (420 nm)	849
W 52/ 11O2	photodeposition	2.0	300 W Ye	>430 nm	N228	PU890 Dt//300		(min 001) C LC	851
Cu3/Z10,25 1.10,75r O4	sulfurization	7.0	300-W Ac	7430 IIIII	11423 T 1142303	LU2300		27.2 (420 mm)	0.71
Cu <sub>2</sub> O/CuMnO <sub>2</sub>	ion exchange	0.05	500-W H	Pyrex filter	$Na_2SO_3 + KOH$	424			898
CuAlO <sub>2</sub> /TiO <sub>2</sub>	direct mixing	0.375	200-W W-H	Pyrex filter	$Na_2S$	21 060			870
CuFeO <sub>2</sub> /SnO <sub>2</sub>	direct mixing	300	W W-009	Pyrex filter	$Na_2S_2O_3$	11 606 410 688		0.5	871
CdS/A GG 8.	solid-state reaction precinitation	0.23	450-W Hα	~420 nm	N3.8 + N3.80.	410 066 Pt/4730		19.7 (>420 nm)	873
TiO <sub>2</sub> /AgGaS <sub>2</sub>	solid-state reaction,	0.1	450-W Hg	>420 nm	$Na_2S + Na_2SO_3$	Pt/4200		17.5 (>420 nm)	874
	sol—gel method	0.3	450 W W	430 000	[C#04]	D*- D.:. O O		101	310
CdS/Na <sub>2</sub> Ti <sub>2</sub> O <sub>4</sub> (OH) <sub>2</sub>	potymetrzeu comprex memou hydrothermal, ion exchange, sulfurization	0.15	300-W Xe	>420 nm	$Na_2S + Na_2SO_3$	Pt/2680		43.4 (420 nm)	852–854
$Cr_2O_3/Na_2Ti_2O_4(OH)_2$	hydrothermal, impregnation	0.1	350-W Xe	>400 nm	$Na_2S + Na_2SO_3$	36.4			855
$CaFe_2O_4/PbBi_2Nb_{0.9}W_{0.1}O_9$	solid-state reaction,	0.3	450-W Xe	>420 nm	$AgNO_3$		675	38 (>420 nm)	876
Cu <sub>2</sub> O/WO <sub>3</sub>	electrodeposition	2	400-W Hg	>400 nm	pure water	$\sim 1.9$			298

		reference	633	633	633	611 70, 592, 605–607	609	610 613, 642–644	613, 642–644	613, 642–644	614, 615	614, 615 614, 615	616	617	618	619	604	623	623, 624	625, 626	627 630	632	639	641	645 645 646 647, 648, 650 651	649
		QY (%)	12.2 (300 nm)			90.0	~0.1	(300–630 mm) 1.8 (517 mm)	2.5 (440 nm)	4.5 (517 nm)					6.7 (>420 nm)	2.4 (350 nm)	10.5 (450 nm)	26 (450 nm)	22 (450 nm)	0.3 (400-500 nm)	0.5 (500 nm)		0.0034 (457.9 nm)	0.02 (450 nm)	10 (520 nm) 10.27 (>420 nm) 9.4 (>420 nm)	14.97 (>420 nm)
Dye-sensitized semiconductor photocatalysts for water splitting under visible-light irradiation	$ol \cdot h^{-1} \cdot g^{-1})$	cocatal./O <sub>2</sub>	284	257	45																					
	activity $(\mu \text{mol} \cdot \text{h}^{-1} \cdot \text{g}^{-1})$	cocatal./H <sub>2</sub>	575	513	145	312.5 5360	4000 134	71.6 420	300	1080	310	1020 695	1293	$06\sim$	712	9:589	840	2880	4320	1.5	10	$\sim$ 268	8920	34	2700 2560 1111 1185.5 850	1861.2
		aqueous reaction solution	pure water $+$ KOH (pH = 11)	pure water $+$ KOH (pH = 11)	pure water $+$ KOH (pH = 11)	pure water methyl viologen (MV $^{2+}$ )	EDTA EDTA	$\begin{array}{l} \text{EDTA} \\ \text{acetonitrile} + \text{NaI} \end{array}$	acetonitrile + NaI	acetonitrile + NaI	methanol	methanol methanol	EDTA	EDTA	TEA	$MV^{2+} + EDTA$	EDTA	EDTA	EDTA	KI	EDTA EDTA	$MV^{2+}$ + methanol	EDTA	TEA	TEA + I <sup>-</sup> TEA + I <sup>-</sup> TEA diethanolamine TEA	TEA
		incident light				visible light >400 nm	>420 nm >420 nm	>420 nm >410 nm	>410 nm	>410 nm	>420 nm	>420 nm >420 nm	>420 nm	>420 nm	>420 nm		>420 nm	>420 nm	>420 nm	400-500 nm	>430 nm >520 nm	sunlight	>435 nm	>430 nm	>420 nm >420 nm >460 nm >420 nm >420 nm	>420 nm
		light source	500-W Xe	500-W Xe	500-W Xe	250-W W-H 450-W Xe	300-W Xe 500-W Xe	500-W Xe 300-W Xe	300-W Xe	300-W Xe	500-W Xe	500-W Xe 500-W Xe	300-W Xe	300-W Xe	250-W W-H	150-W Xe	300-W Xe	300-W Xe	300-W Xe	200-W Hg-Xe	500-W Xe 1000-W Xe	direct	450-W Xe	400-W Hg	500-W Xe 500-W Xe 300-W Xe 200-W H 300-W W-H	300-W W-H
		mass (g)	0.1	0.1	0.1	0.04	0.002	0.05	0.05	0.05	0.04	0.04	0.025	0.01	0.005	0.08	0.005	0.005	0.005	0.1	0.02	0.125	0.005	0.05	0.03 0.03 0.02 0.02	0.04
		synthetic method	solid-state reaction	solid-state reaction	solid-state reaction	P25-TiO <sub>2</sub> thermal hydrolysis	P25 TiO <sub>2</sub> TiO-5 standard sample	TiO-5 standard sample Ishihara ST-01	Ishihara ST-01	Ishihara ST-01	hydrothermal method	hydrothermal method hydrothermal method	sol-gel, adsorption method	adsorption method	adsorption method	Aldrich	solid-state reaction	solid-state reaction,	solid-state reaction,	solid-state reaction,	P25 TiO <sub>2</sub> thermal hydrolysis	commercial anatase TiO <sub>2</sub>	Montedison TiO₂−U	commercial TiO <sub>2</sub>	Commercial ZnO commercial ZnO ST-01 TiO <sub>2</sub> P25 TiO <sub>2</sub> hydrothermal method, chemical	adsorption hydrothermal method, chemical adsorption
		photocatalyst	K <sub>0.95</sub> Ta <sub>0.92</sub> Zt <sub>0.08</sub> O <sub>3</sub> /Pt/	Cyanocobananin K <sub>0.95</sub> Ta <sub>0.92</sub> Zt <sub>0.08</sub> O <sub>3</sub> /Pt/	Ci—teu apnenyiporphytin K <sub>0.95</sub> Ta <sub>0.92</sub> Zi <sub>0.08</sub> O <sub>3</sub> /Pt/ Co totroopenyiporphytiin	CO-teu apiteu) poi piuj m TiO <sub>2</sub> /Pt = RuO <sub>2</sub> / TiO <sub>2</sub> /Pt = RuO <sub>2</sub> / Pudinani 2+	TiO <sub>2</sub> /Pt/tet/akis TiO <sub>2</sub> /Pt/tet/akis (4.cokoccakis	(4-carboxyphenyl)popnine TiO <sub>2</sub> /PVRu(bpym) <sub>3</sub> <sup>2+</sup> TiO <sub>2</sub> /PV	TiO <sub>2</sub> /Pt/	Counaim aye TiO <sub>2</sub> /Pt/ N3 dva	mesoporous TiO <sub>2</sub> /Pt/ Pu(dchw)_(MCs)_	http://pyst.nc.s/2 mesoporous TiO <sub>2</sub> /Pt/Ru <sub>2</sub> (bpy) <sub>4</sub> L <sub>1</sub> -PF <sub>6</sub> mesoporous TiO <sub>2</sub> /Pt/Ru <sub>2</sub> (bpy) <sub>4</sub> L <sub>1</sub> -PF <sub>6</sub> Pt-Ru <sub>2</sub> (ham) (tim) —NO	FUNU(0py)2(mm)2=10O3 Al <sub>2</sub> O3/TiO <sub>2</sub> / Pt/Ru(4ny)-2+	guanidinium/Nafion/TiO <sub>2</sub> / Pr/Du/(hay), $^{2+}$	Db [NiFeSe]-H/ P::P/T:O	SnO <sub>2</sub> /Pt-RuO <sub>2</sub> / Ru(hmv), <sup>2+</sup>	K4Nb <sub>6</sub> O <sub>17</sub> / Pt/Ru(hav), 2+	$H_4Nb_6O_{17}/D_4N_6O_{17}/D_6O_{17}/D_6O_{17}/D_6O_{17}/D_{17}$	$FORU(0Py)_2(+,+,+,+) = -(FO_3FI_2)_20Py)$ $FORU(0Py)_2(+,+,+,+) = -(FO_3FI_2)_2(+,+)$ $FORU(0Py)_2(+,+,+) = -(FO_3FI_2)_2(+,+)$	$K_{4-x}H_xNb_6O_{17} \cdot nH_2O/Pt/RuL_3(PF_6)_2$	TiO <sub>2</sub> /Pt(alizarine) <sub>2</sub> TiO <sub>2</sub> /Pt/zinc	polphymis TiO <sub>2</sub> /RuO <sub>2</sub> /copper phtholograpina	piniatocyanine TiO <sub>2</sub> /Pt-RuO <sub>2</sub> /8- bidroxyaninolina	nyuroxydumomie 1,1'-binaphthalene-2,2'-	unon or to the control of the contro	Na <sub>2</sub> Ti <sub>2</sub> O <sub>4</sub> (OH) <sub>2</sub> / Pt/Eosin-Y

Table 2. Continued

		reference	652	653	653	653	654	655	959	634, 635 637, 638
		QY (%)	12.01 (>420 nm)	6.65 (>420 nm)	17.36 (>420 nm)	15.04 (>420 nm)	19.1 (>420 nm)	10.4 (>420 nm)	12.14 (>420 nm)	20.0 (670 nm)
iation	$(ol \cdot h^{-1} \cdot g^{-1})$	cocatal./H <sub>2</sub> cocatal./O <sub>2</sub>								
Dye-sensitized semiconductor photocatalysts for water splitting under visible-light irradiation	activity (\(\mu\text{mol} \cdot \hat{h}^{-1} \cdot \mathbf{g}^{-1}\)	cocatal./H <sub>2</sub>	1027.5	1800	4690	4060	2750	716.7	3060	2730 ~62.5
		aqueous reaction solution	TEA	TEA	TEA	TEA	TEA	TEA	TEA	ethanol glycerol
		light source incident light	>420 nm	>420 nm	>420 nm	>420 nm	>420 nm	>420 nm	>420 nm	visible light >420 nm
		light source	300-W W-H	300-W W-H	300-W W-H	300-W W-H	400-W H	300-W W-H	300-W W-H	400-W W 300-W H
nsitized semi		mass (g)	0.04	0.02	0.02	0.02	0.1	90.0	0.04	0.075
Dye-sensit		synthetic method	hydrothermal method, chemical adsorption	hydrothermal method, chemical adsorption	hydrothermal method, chemical adsorption	hydrothermal method, chemical adsorption	chemical adsoprtion	chemical adsorption	refluxing treatment	impregnation method photocatalytic reduction
		photocatalyst	Ti-MCM-41/Pt/Eosin-Y	titanic acid short nanotubes/ Pt/Eosin-Y	titanic acid long nanotubes/ Pt/Eosin-Y	titanic acid nanorods/Pt/ Eosin-Y	TiO <sub>2</sub> /Pt/Fe <sup>3+</sup> -Eosin-Y	silica gel/Pt/Eosin-Y	multiwalled carbon nanotube/ Pt/Eosin-Y	Co-HPA/Pt/TiO <sub>2</sub> Wells-Dawson heteropoly blue/ Pt/TiO <sub>2</sub>

"The unit of  $H_2/O_2$  evolution rate is  $\mu$  mol·h<sup>-1</sup>, as the weight of photocatalyst has not been given in the reference. <sup>b</sup> Tungsten—halogen lamp. <sup>c</sup> Tungsten-arc lamp

implanted. The absorption maximum and minimum values stayed constant. Such a shift not only allowed the metalion-implanted TiO<sub>2</sub> to use solar irradiation more effectively but also initiated effective photocatalytic reactions, under both UV and visible-light irradiation. It was found that Pt<sup>4+</sup>and Ag<sup>+</sup>-doped TiO<sub>2</sub> nanoparticles also exhibited improved photocatalytic activities under visible-light or UV irradiation. This was explained as follows: the doping ions in these photocatalysts not only contributed to the visible-light absorption but also served as a recombination inhibitor by trapping electrons or holes, which in return promoted the charge separation required for the photocatalytic reaction. 328,329 However, in some cases, it was found that the metal-ion dopants could also serve as the recombination sites for photoinduced charges and not a recombination inhibitor. This resulted in relatively low photocatalytic activity even under UV irradiation.

In 1994, Choi et al. investigated the effects of metal ion doping on the photocatalytic activity of TiO<sub>2</sub>. <sup>330</sup> They found that the photocatalytic activity was related to the electron configuration of the dopant ion. Of the 21 metal-ion dopants studied, Fe, Mo, Ru, Os, Re, V, and Rh ion doping significantly increased the photocatalytic activity, whereas Co and Al ion doping caused detrimental effects. The nature of the metal-ion dopant in the TiO<sub>2</sub> significantly affected the charge recombination and electron transfer rates. An electron paramagnetic resonance (EPR) study of doped TiO<sub>2</sub> colloids showed that Fe- or V-doped TiO<sub>2</sub> caused the growth of the Ti<sup>3+</sup> signal. These changes were attributed to an inhibition of the hole-electron recombination by Fe<sup>3+</sup> or V<sup>4+</sup> dopant. On the other hand, Mo<sup>6+</sup> dopant behaved as an irreversible electron trap.331

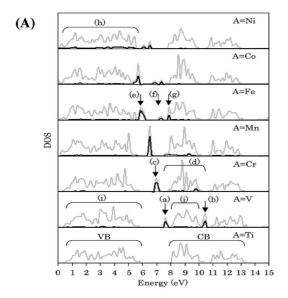
Kudo and co-workers reported that Ni<sup>2+</sup>-, Rh<sup>3+</sup>-, or Cr<sup>3+</sup>doped TiO2 exhibited virtually no photocatalytic activity, whereas TiO<sub>2</sub> codoped by Ni<sup>2+</sup>, Rh<sup>3+</sup>, or Cr<sup>3+</sup> together with Ta<sup>5+</sup>, Nb<sup>5+</sup>, or Sb<sup>5+</sup> led to efficient O<sub>2</sub> evolution from water. 303,332-335 Reaction conditions required Ag<sup>+</sup> as an electron acceptor and visible-light irradiation. Doping by Ni<sup>2+</sup>, Rh<sup>3+</sup>, or Cr<sup>3+</sup> created donor levels in the forbidden band and made TiO<sub>2</sub> responsive to visible light. The charge balance was maintained by codoping of Ta<sup>5+</sup>, Nb<sup>5+</sup>, or Sb<sup>5+</sup>, which was needed to suppress the recombination between photogenerated electrons and holes. Ohno et al. expanded the effective wavelength of TiO<sub>2</sub> photocatalyst into the visible region by Ru doping.<sup>336</sup> With this particular photocatalyst, the oxygen evolution reaction occurred under the irradiation of visible light at wavelengths longer than 440 nm. Iron(III) ions were used as the electron acceptor. Kahn and co-workers found that Pt-, Ir-, and Co-ionized titania nanotubes prepared by an ion-exchange method were effective photocatalysts for the production of stoichiometric hydrogen and oxygen by water splitting under visible-light irradiation. 337,338 Pt, Ir, and Co ionization all reduced the band gap energy as indicated by UV-vis absorption spectra, resulting in the visible photoresponse ability. Visible-light-responsive TiO<sub>2</sub>, obtained by self (Ti<sup>3+</sup>) doping, was found to decompose water in methanol or silver nitrate solution under visible-light irradiation. The observed absorption and photocatalytic ability in the visible-light region was ascribed to the defect levels present within the band gap of these samples.<sup>339–342</sup>

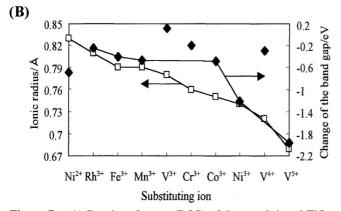
Attention has also been paid to other oxide semiconductors as host photocatalysts for metal-ion doping. Cr-doped  $SrTi_{1-x}Cr_xO_3$  (x = 0.00, 0.02, 0.05, 0.10) was prepared by a solvothermal method and showed increasing photocatalytic (A)

H<sup>+</sup>/H<sub>2</sub>

O2/H2O

Figure 6. Donor level (A) and acceptor level (B) formed by metal ion doping.





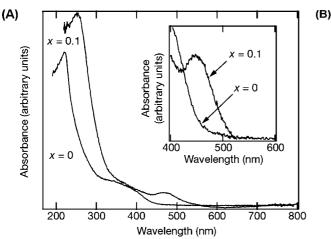
**Figure 7.** (A) Density of states (DOS) of the metal-doped  $\text{TiO}_2$  ( $\text{Ti}_{1-x}A_xO_2$ : A = V, Cr, Mn, Fe, Co, or Ni). Gray solid lines: total DOS. Black solid lines: dopant's DOS. The states are labeled a-j. Reprinted with permission from ref 315. Copyright 2002 Elsevier. (B) The relationship between the ionic radius and the change of the band gap (anatase). Except for some cations, the band gaps decrease with decreasing cation size. Reprinted with permission from ref 316. Copyright 1999 The Chemical Society of Japan.

activities for hydrogen production under both UV and visible light with increasing amounts of chromium. The new band gap in the visible-light range obtained by Cr doping was attributed to the band transition from the Cr 3d to the Cr 3d + Ti 3d hybrid orbital. Wang et al. studied the photophysical and photocatalytic properties of SrTiO<sub>3</sub> doped with Cr cations on different sites. Harrestingly, (Sr<sub>0.95</sub>Cr<sub>0.05</sub>)TiO<sub>3</sub> with Cr cations (Cr<sup>3+</sup>) doped at the Sr<sup>2+</sup> sites showed higher

photocatalytic activity of  $H_2$  evolution under visible-light irradiation than  $Sr(Ti_{0.95}Cr_{0.05})O_3$  with Cr cations (mixed  $Cr^3+$  and  $Cr^{6+}$ ) doped at the  $Ti^{4+}$  sites. It was concluded that, because the  $Cr^{6+}$  cations created the potential level of the empty  $Cr^{6+}$  lower than that for  $H_2$  evolution and thus behaved as the trapping center for photoinduced electrons, they should be avoided in Cr-containing visible-light-driven photocatalysts intended for water splitting.

Kudo and co-workers found that Mn- and Ru-doped SrTiO<sub>3</sub> showed photocatalytic activities for O<sub>2</sub> evolution from an aqueous silver nitrate solution. 344 Ru-, Rh-, and Ir-doped SrTiO<sub>3</sub> loaded with Pt cocatalysts produced H<sub>2</sub> from an aqueous methanol solution under visible-light irradiation ( $\lambda$ > 440 nm). In particular, the Rh (1%)-doped SrTiO<sub>3</sub> gave a quantum yield of 5.2% at 420 nm for H<sub>2</sub> evolution. The visible-light response of Rh-doped SrTiO<sub>3</sub> was due to the transition from the electron donor level formed by the Rh ions to the conduction band composed of Ti 3d orbitals. When SrTiO<sub>3</sub> was codoped with Sb<sup>5+</sup>/Cr<sup>3+</sup>, Ta<sup>5+</sup>/Cr<sup>3+</sup>, or Ta<sup>5+</sup>/Ni<sup>2+</sup>, all the resulting compounds displayed efficient photocatalytic activities for hydrogen production from aqueous methanol solutions under visible-light irradiation ( $\lambda$  > 420 nm).  $^{303,333-345}$  When Ti<sup>4+</sup> was replaced by Ni<sup>2+</sup> or Cr<sup>3+</sup> in SrTiO<sub>3</sub>, the results were similar to those for Sb<sup>5+</sup>/Cr<sup>3+</sup>codoped TiO<sub>2</sub>. The charge became unbalanced, and recombination centers formed. When a second metal ion such as Ta<sup>5+</sup> or Sb<sup>5+</sup> was codoped to compensate for the charge unbalance and suppress the formation of the recombination centers, the visible-light absorption was maintained.

Miyake and co-workers<sup>346</sup> and Chen and co-workers<sup>347</sup> demonstrated that both Rh<sup>3+</sup>- and Cu<sup>2+</sup>-doped CaTiO<sub>3</sub> were good candidates for visible-light-driven oxide photocatalysis for hydrogen evolution. Reaction occurred under visible-light irradiation with methanol as the electron donor. Miyake and co-workers also found that Rh3+-doped Ca3Ti2O7 was active in photocatalytic hydrogen generation under visible light.<sup>348</sup> Moreover, it was found that Rh<sup>3+</sup>-doped Ca<sub>3</sub>Ti<sub>2</sub>O<sub>7</sub> with a layered structure showed higher stability in air than Rh<sup>3+</sup>doped CaTiO<sub>3</sub> with a simple perovskite structure. Lee and co-workers studied the effects of both Cr and Fe cation substituion on the photophysical and photocatalytic properties of the layered perovskite La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub>. 305,349 The contribution of these dopants led to the formation of a partially filled 3d band, which served as the electron donor level in the band gap of La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub>. It also caused the excitation of electrons from this localized interband to the conduction band of La<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub> and was responsible for visible-light absorption and the H<sub>2</sub> evolution from water under visible light. Suzuki and co-workers reported that Fe<sup>3+</sup> and W<sup>6+</sup> substitution for Ti<sup>4+</sup> in K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub> resulted in a small activity for hydrogen



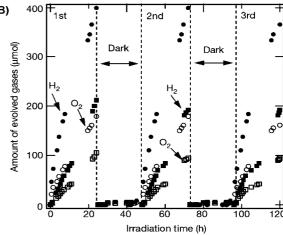


Figure 8. (A) Optical properties of the photocatalyst. The main panel shows the ultraviolet—visible diffuse reflectance spectra of  $In_{1-x}Ni_xTaO_4$  (x = 0 and 0.1) at room temperature, with the inset providing an expanded view of the spectra in the wavelength region from 400 to 600 nm. (B) Photocatalytic  $H_2$  and  $O_2$  generation. Shown are the evolution of  $H_2$  and  $O_2$  from pure water using as catalyst a suspension of  $NiO_y/In_{0.9}Ni_{0.1}TaO_4$  (solid circles,  $H_2$ ; open circles,  $O_2$ ) and  $RuO_2/In_{0.9}Ni_{0.1}TaO_4$  (solid squares,  $H_2$ ; open squares,  $O_2$ ). Experiments were done using 0.5 g of catalyst powder suspended in 250 mL of pure water in a Pyrex glass cell under visible-light irradiation (>420 nm). Light source: 300-W Xe lamp. The gases evolved were determined by thermal conductivity detector (TCD) gas chromatograph. The measurement uncertainties were  $\sim 0.05\%$ . Reprinted with permission from ref 353. Copyright 2001 Nature Publishing Group.

production under visible-light irradiation.<sup>350</sup> Under the same condition, pure K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub> showed no activity.

Recently, it was reported that both V-doped and Zn-doped K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub> exhibited high photocatalytic activities of hydrogen production under visible light. 140,351 The hybridization of either V 3d or Zn 3d and O 2p electron orbitals resulted in a new localized energy level. The catalyst was easily excited with lower energy, which in turn improved the photoactivity of K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub> for water splitting. The band gap and photocatalytic activity of In<sub>2</sub>TiO<sub>5</sub> also underwent significant changes as a result of V doping. These enabled the absorption of radiation from the entire visible region of 400-800 nm and led to the improvement of photocatalytic hydrogen production under visible light.352 It is worth mentioning that Zou and co-workers investigated the doping effects of different metal ions (Mn, Fe, Co, Ni, Cu) on the structural and photocatalytic properties of a InTaO<sub>4</sub> photocatalyst, 353-357 as shown in Figure 8. Of these, In<sub>0.9</sub>Ni<sub>0.1</sub>TaO<sub>4</sub> showed the highest photocatalytic activity, which induced direct splitting of water into stoichiometric amounts of oxygen and hydrogen under visible-light irradiation with a quantum yield of  $\sim 0.66\%$  at 402 nm. The narrower band gap of Ni-doped InTaO4 was attributed to the Ni 3d donor level formed in the forbidden band. 353 Ni-modified K<sub>4</sub>Nb<sub>6</sub>O<sub>10</sub> prepared by a solid-state reaction also exhibited increased visible-light absorption and photocatalytic activity for hydrogen evolution under visible-light irradiation. This was believed to be due to the effect of the Ni<sup>2+</sup> doping. <sup>358</sup> Cu 3d donor level formed above the valence band of BiTaO<sub>4</sub> by Cu<sup>2+</sup> doping. This contributed to the increased photocatalytic activity for hydrogen evolution under visible light compared to BiTaO<sub>4</sub>.359 The strong photoabsorption and good performance of H<sub>2</sub> evolution in the visible-light region shown by Cr<sup>3+</sup>-doped Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub> was largely attributed to the occurrence of the impurity level Cr 3d in both conduction and valence bands.360

Kudo and co-workers controlled the oxidation numbers of Ir when it substituted for Nb<sup>5+</sup> and Ta<sup>5+</sup> ions in the B sites of a perovskite structure for NaBO<sub>3</sub> (B = Nb, Ta).<sup>361</sup> Their strategy consisted of codoping of alkaline earth metal and lanthanum ions for Na<sup>+</sup> ions at the A sites. This

contributed to maintaining the charge balance in NaBO3. The resulting NaNbO3:Ir/A (A = Sr, Ba, and La) showed H2 or O2 evolution, whereas NaTaO3:Ir/A showed H2 evolution under visible light. Yang et al. reported that La³+/Cr³+codoped NaTaO3 showed intense visible-light absorption and H2 production in the presence of methanol under visible-light irradiation. The photocatalytic activity was much higher than that of monodoped NaTaO3. A high visible-light photocatalytic activity for O2 evolution from an aqueous solution containing an electron acceptor (quantum yield of 6% at 420 nm) was found for Cr<sup>6+</sup>-doped PbMoO4. It should be noted that the formation of an electron-acceptor level of Cr 3d orbitals below conduction band as a result of the Cr<sup>6+</sup> replacement was believed to have given rise to the absorption bands and photocatalytic activity in visible-light region.  $^{363}$ 

Although ZnS is a highly efficient photocatalyst for  $H_2$  evolution because of its high conduction band level, it is only active in the UV light region. Photocatalyst Hence, it would be quite interesting to see if ZnS becomes visible-light responsive while maintaining its high  $H_2$  evolution activity following some modification. With this in view, a series of Cuand Ni-doped ZnS photocatalysts were prepared. The results showed high activities for  $H_2$  evolution from the aqueous solutions containing sulfite ions under visible-light irradiation ( $\lambda > 420$  nm), even in the absence of a platinum cocatalyst. The visible-light response was a result of the formation of 3d donor levels by the doped Cu<sup>2+</sup> and Ni<sup>2+</sup> in the wide band gap of the ZnS host photocatalyst.

In contrast to the doping modifications described previously, which focused on UV-light-active photocatalysts, some visible-light-driven photocatalysts were also modified by metal-ion doping with the aim of narrowing their band gaps in order to utilize visible light in further longer-wavelength regions to produce more efficient photocatalytic water splitting. Reber and Meier modified the visible-light active photocatalyst CdS by doping it with  $Ag^+$  in order to extend the spectral response up to 620 nm. A significant enhancement in the photocatalytic activity for hydrogen evolution was obtained with CdS powders containing 0.77%  $Ag^+$ . Ren et al. reported two series of photocatalysts based on CdS as the base with a little of MInS<sub>2</sub> (M = Cu, Ag) as the dopant.



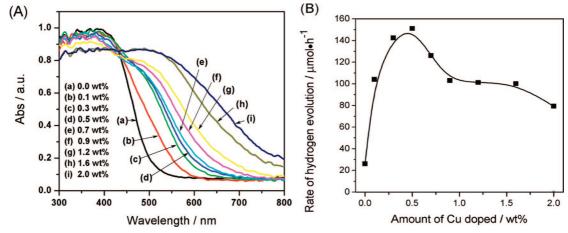


Figure 9. (A) Diffuse reflectance spectra of Cu-doped ZnIn<sub>2</sub>S<sub>4</sub>. (B) Dependence of photocatalytic activity for H<sub>2</sub> evolution over Cu-doped  $ZnIn_2S_4$ . The concentrations of  $Cu^{2+}$  were (a) 0.0 wt %, (b) 0.1 wt %, (c) 0.3 wt %, (d) 0.5 wt %, (e) 0.7 wt %, (f) 0.9 wt %, (g) 1.2 wt %, (h) 1.6 wt %, and (i) 2.0 wt %, respectively. Reprinted with permission from ref 378. Copyright 2008 American Chemical Society.

The resulting compounds had narrower band gaps and higher photocatalytic activities than undoped CdS. More recently, a large degree of Mn doping was discovered to be effective in improving the photocatalytic activity and stability of CdS for visible-light hydrogen production. 367

In independent studies, Liu et al.<sup>368</sup> and Xu and coworkers<sup>369-371</sup> reported that the Cu-doped  $Zn_xCd_{1-x}S$  solidsolution photocatalyst with the absorption edge shifted to a lower energy region displayed higher water-splitting efficiency to produce hydrogen than Zn<sub>x</sub>Cd<sub>1-x</sub>S, which itself had been shown to be an excellent visible-light-driven photocatalyst.372,373 Zhang et al. found Ni2+ doping also greatly improved the photocatalytic activity of  $Zn_xCd_{1-x}S$ for hydrogen production.<sup>374</sup> For some time it has been known that ZnIn<sub>2</sub>S<sub>4</sub> can efficiently produce hydrogen from some aqueous solutions with sulfite and sulfide ions as the electron donor under visible-light irradiation.<sup>375–377</sup> Shen et al. hydrothermally synthesized a series of Cu-doped ZnIn<sub>2</sub>S<sub>4</sub> photocatalysts with broader absorptions in the visible-light region than the corresponding undoped ZnIn<sub>2</sub>S<sub>4</sub>. <sup>378</sup> As shown in Figure 9, they found that the photocatalytic activity of ZnIn<sub>2</sub>S<sub>4</sub> was remarkably enhanced by Cu doping. The 0.5 wt % Cu-doped ZnIn<sub>2</sub>S<sub>4</sub> photocatalyst showed the highest activity for hydrogen evolution under visible-light irradiation. The surplus doped Cu2+ ions served as the recombination sites for the photogenerated electrons and holes.<sup>379</sup>

#### 4.1.2. Nonmetal-Ion Doping

Nonmetal ion doping is another approach used to modify UV-light-active photocatalysts. It has been widely used to narrow the band gap and improve the visible-light-driven photocatalytic activity. Unlike metal-ion dopants, nonmetalion dopants are less likely to form donor levels in the forbidden band but instead shift the valence band edge upward. This results in a narrowing of band gap as indicated in Figure 10. The technology of nonmetal-ion doping has been widely used to modify some UV-light-active oxide photocatalysts, such as Ti-based oxides, 380-387 Ta-based oxides, 388-395 Zr-based oxides, 396-398 Nb-based oxides, 398-402

Various nonmetal ions (such as C, N, S, etc.) were used to dope TiO<sub>2</sub>, and the products were studied for their optical and photocatalytic properties. Nonmetal-ion doped TiO<sub>2</sub>, in which the absorption spectra were red-shifted to longer wavelengths, exhibited improved photocatalytic activities

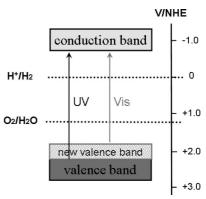


Figure 10. New valence band formation by doping of nonmetal

compared to those for pure TiO<sub>2</sub>, especially in the visiblelight region. 403-416 Chen and co-workers used X-ray photoelectron spectroscopy (XPS) to show that additional electronic states exist above the valence band edge of pure TiO<sub>2</sub> for C-, N-, and S-doped TiO<sub>2</sub> (Figure 11A). 417,418 This additional electron density of states can explain the redshifted absorption of these potential photocatalysts, as observed in the "shoulder" and "tail-like" features in the UV-vis spectra (Figure 11B). Asahi et al. studied the substitutional doping of C, N, F, P, and S for O in anatase TiO<sub>2</sub>. 419 Using full-potential linearized augmented plane wave (FLAPW) method, they calculated the electronic band structures of anatase TiO2 with different substitutional dopants as shown in Figure 12. They found that the substitution of N for O, which leads to the mixing of the 2p states of N with the 2p of O, was the most effective. It led to the band gap narrowing by shifting the valence band edge upward, which in return resulted in N-doped TiO2 having a much higher photocatalytic activity than pure TiO<sub>2</sub> in the visible-light region. Chen et al. found that the ease and degree of nitrogen doping was responsible for the significant increase in photocatalytic activity observed in the TiO2 nanocolloid versus the nitrided commercial nanopowder. 414 Braun et al. revealed an additional eg resonance in the valence band of TiO<sub>2</sub> formed by nitrogen doping in the oxygen 1s NEXAFS pre-edge. 420 This extra resonance was found to bear coresponsibility for the photocatalytic performance of N-doped TiO<sub>2</sub> at visible-light wavelengths.

Up until now, N-doped TiO<sub>2</sub> has been both widely investigated and successfully prepared by many different

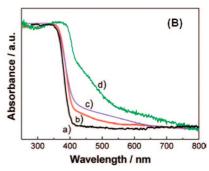
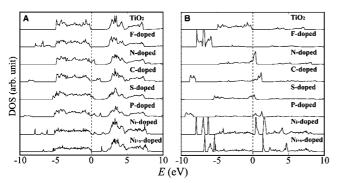


Figure 11. VB XPS spectra (A) and diffuse reflectance spectra (B) of (a) pure TiO<sub>2</sub>, (b) C-TiO<sub>2</sub>, (c) S-TiO<sub>2</sub>, and (d) N-TiO<sub>2</sub>. Reprinted with permission from ref 418. Copyright 2008 American Chemical Society.



**Figure 12.** (A) Total DOSs of doped TiO<sub>2</sub> and (B) the projected DOSs into the doped anion sites, calculated by FLAPW. The dopants F, N, C, S, and P were located at a substitutional site for an O atom in the anatase TiO<sub>2</sub> crystal (the eight TiO<sub>2</sub> units per cell). The results for N doping at an interstitial site (N<sub>i</sub>-doped) and that at both substitutional and interstitial sites (N<sub>i+s</sub>-doped) are also shown. Reprinted with permission from ref 419. Copyright 2001 AAAS.

methods. These include physical/chemical vapor deposition, 421,422 the heating of titanium hydroxide and urea, 423 reactive magnetron sputtering, 424,425 a hydrothermal/solvothermal process, 426,427 and treating TiO<sub>2</sub> powder in a NH<sub>3</sub> gas flow. 428,429 Yuan et al. synthesized N-doped TiO2 by heating a mixture of urea and TiO<sub>2</sub> at 350-700 °C in air. 430 XPS confirmed that both the molecularly chemisorbed  $N_2$ and the substitutional N contributed to the material response to visible light. However, the substitutional N was considered the predominate factor in improving the photocatalytic activity of the water splitting under visible light. Lin et al. demonstrated that the N-doped TiO<sub>2</sub> photocatalyst, synthesized by a two-microemulsion technique, showed favorable photocatalytic hydrogen evolution at neutral pH range in methanol/water solutions. 431 Pillai and co-workers reported that the chemical modification of titanium isopropoxide using different nonmetallic chemical reagents such as urea, sulfuric acid, and trifluoroacetic acid improved both the anatase stability and the photocatalytic activity by doping with nitrogen, sulfur, and fluorine, respectively. 432-434

The photocatalytic activity of S-doped TiO<sub>2</sub> has also been studied in depth. Using different synthetic methods, anionic sulfur can be doped into TiO<sub>2</sub> to replace the lattice oxygen or as an cation to replace the Ti ion.<sup>407,435–438</sup> It was found that S-doped TiO<sub>2</sub> showed different photocatalytic activities under visible light.<sup>439</sup> In Nishijima and co-workers' study of S-doped TiO<sub>2</sub>, a higher activity level for hydrogen evolution than for N-doped TiO<sub>2</sub> photocatalysts under visible light was reported.<sup>440</sup> Kahn et al. prepared C-doped TiO<sub>2</sub> with main rutile structure by pyrolyzing Ti metal in a natural gas flame. The C-doped TiO<sub>2</sub> possessed lower band gap

energy and displayed a much higher photoactivity in water splitting than pure  ${\rm TiO_2}$  with the mixed phase of antanse and rutile. He was found that C-doped  ${\rm TiO_2}$  nanotubes also displayed a high photoactivity for water splitting. He bandgap reduction and the new intragap band formation in C-doped  ${\rm TiO_2}$  nanotubes extended its utilization of solar energy up to the visible to infrared region.

To further enhance the visible-light activity, nonmetal codoping of TiO<sub>2</sub> such as F/B-codoping, 443,444 F/N-codoping, 445,446 S/F-codoping, 447 F/C-codoping, 448 C/N-codoping, 449,450 S/N-codoping, 451,452 N/Br-codoping, 458 N/Si-codoping, 456 C/S-codoping, 457 and P/Fcodoping<sup>458</sup> have all been studied. Results suggested that nonmetal-codoped TiO<sub>2</sub> compounds had significant visible photocatalytic activities due to the synergetic effect of the codoping. Luo et al. found that Cl/Br-codoped TiO<sub>2</sub> displayed a much higher photocatalytic activity for water splitting than either Cl- or Br-doped TiO<sub>2</sub>. 459 Domen and co-workers found that N/F-codoped TiO<sub>2</sub> had high visible-light photocatalytic activity for water oxidation due to the absorption band of  $TiN_xO_yF_z$  in the visible region. 460-462 N/S-codoped  $TiO_2$ showed considerable activity in the photocatalytic water splitting, and under visible light produced H<sub>2</sub>. This was attributed to the visible-light photoexcitation of the electrons from the isolated energy levels in the band gap formed by the doped N or S to the conduction band. 463 Liu et al. synthesized B/N-codoped TiO<sub>2</sub> with high visible-light photocatalytic activity. 464 They proposed that the synergistic effects of B/N codoping created visible-light absorption by N doping, and the lifetime of the photoinduced carriers was prolonged by B doping. Subsequently, Li et al. found that the visible-light-driven photocatalytic activity of B/Ncodoped TiO<sub>2</sub> for hydrogen evolution increased greatly compared to that of N-doped TiO2. 465 They believed that the codoping of boron contributed to the excellent activity of B/N-codoped TiO<sub>2</sub>, because the doped boron could act as shallow traps for electrons, which prolong the life of photoinduced electrons and holes. Recently, OCN-doped TiO<sub>2</sub> nanoparticles were prepared for photocatalysis in the visible-light region of 380-550 nm as indicated by UV-vis absorption spetra. 466 In this doped TiO2, the OCN group bonding to titanium atoms produces a weakening of its C=O double bond and a hardening of CN and NO bonds, which enables visible absorption and related photocatalytic activity.466

Similar to the case of N-doped TiO<sub>2</sub>, N-doping in Ta<sub>2</sub>O<sub>5</sub> also caused a valence band upward shift to a higher energy level, which narrows the band gap.<sup>391,467</sup> TaON and Ta<sub>3</sub>N<sub>5</sub>, prepared from reacting Ta<sub>2</sub>O<sub>5</sub> with NH<sub>3</sub>, were found to be active for water splitting into hydrogen and oxygen under visible-light irradiation.<sup>390,468–472</sup> In particular, in the presence

of the sacrificial electron acceptor (Ag<sup>+</sup>), TaON functioned as a very efficient photocatalyst for the oxidation of water into O<sub>2</sub> (quantum yield of 34%). 468 Recently, nonmetal-iondoped titanates and tantalates have also been studied for photocatalytic reduction and oxidation under visible-light irradiation. Wang and co-workers found that, in the visiblelight range and the near-ultraviolet range, N-doped and N/Scodoped SrTiO3 displayed much higher photocatalytic activities for NO elimination than pure SrTiO<sub>3</sub>. 473,474 Kasahara and co-workers found that under visible-light irradiation LaTiO<sub>2</sub>N reduced H<sup>+</sup> into H<sub>2</sub> and oxidized H<sub>2</sub>O into O<sub>2</sub> in the presence of a sacrificial electron donor (methanol) or acceptor (Ag<sup>+</sup>) using the band gap transition (2.1 eV). 475,476 Moreover, LaTiO2N could split water without any sacrificial reagent under visible-light irradiation when used as a photoanode in the photoelectrochemical cell.<sup>477</sup> The visiblelight absorption was attributed to the new valence band composed of O 2p orbitals for the lower-energy side and N 2p orbitals for the higher-energy side. Similarly, because of N-doping reducing the band gap energy, N-doped KTa<sub>0.92</sub>Zr<sub>0.08</sub>O<sub>3</sub> was reported to achieve a complete splitting of water under visible-light irradiation, when Pt was loaded as a cocatalyst. 478 Yamasita and co-workers prepared  $MTaO_2N$  (M = Ca, Sr, Ba) by nitriding  $M_2Ta_2O_7$  at 1123 K for 15 h. MTaO<sub>2</sub>N had small band gap energies (2.5-2.0 eV) and could absorb visible light at 500-630 nm via the N 2p orbitals of the upper regions of the valence bands. 42,395 Under visible-light irradiation, MTaO<sub>2</sub>N reduced H<sup>+</sup> into H<sub>2</sub> in the presence of a sacrificial electron donor. These oxynitrides, however, did not function in the oxidation of water due to the fact that the valence band did not have sufficient overpotentials for the oxidation potential of water. Liu et al. reported Y<sub>2</sub>Ta<sub>2</sub>O<sub>5</sub>N<sub>2</sub> as a novel photocatalyst with high activity for water splitting into hydrogen and oxygen under visible-light irradiation in the presence of appropriate sacrificial reagents.<sup>393</sup> The smaller bandgap energy of Y<sub>2</sub>Ta<sub>2</sub>O<sub>5</sub>N<sub>2</sub> was attributed to the partial replacement of O<sup>2-</sup> by N3- in YTaO4 and the predominant population of the valence band by a hybrid orbital between N 2p and O 2p.

In addition to Ti-based and Ta-based oxides as the host photocatalysts, some other oxides with wide band gaps were also modified by using nonmetal-ion doping and were found to improve visible-light-driven photocatalytic activities for water splitting. Ji et al. studied photocatalytic water splitting using nitrogen-doped Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub> under visible-light irradiation.<sup>399</sup> After nitrogen doping, the Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub> band gap energy was reduced and subsequently induced photocatalytic activity for hydrogen production from methanol-water mixtures in the visible-light region. Recently, N-doped ZrO<sub>2</sub> and In<sub>2</sub>O<sub>3</sub>, C-doped In<sub>2</sub>O<sub>3</sub>, and Nb<sub>2</sub>O<sub>5</sub> were also reported to show enhanced photocatalytic activities for water splitting under visible-light irradiation compared to the undoped oxide hosts. 396,479-481 Hisatomi et al. found that the novel spineltype d<sup>0</sup>-d<sup>10</sup> complex oxynitride photocatalyst Zn<sub>x</sub>TiO<sub>y</sub>N<sub>z</sub> displayed photocatalytic activity for both the reduction and oxidation of water in the presence of either a sacrificial electron donor or acceptor under irradiation at visible wavelengths. 482 Li et al. demonstrated for the first time the development of a simple lamellar-solid acid photocatalyst. 400 N-doped HNb<sub>3</sub>O<sub>8</sub> displyed superior visible-light-responsive photocatalytic activity in comparison to N-doped Nb<sub>2</sub>O<sub>5</sub> and TiO<sub>2</sub>. Kanade et al. reported that N-doped Nb<sub>2</sub>Zr<sub>6</sub>O<sub>17</sub> gave a quantum yield of 13.5% for photocatalytic hydrogen production from the decomposition of hydrogen sulfide in KOH aqueous solution under visible-light irradiation.<sup>398</sup> N-doped Ga-Zn mixed oxides with a band gap of 2.6 eV were capable of producing hydrogen from a methanol aqueous solution with an apparent quantum efficiency of 5.1% under visible-light illumination. 483 Rh/Cr<sub>2</sub>O<sub>3</sub> was loaded as the cocatalyst.

Some other S<sup>2-</sup> substituted metal oxides have also received considerable attention as visible-light-responsive photocatalysts for water reduction and/or oxidation. For example, the absorption edge of ZrW2O8 was significantly extended to the visible-light region by S doping. 484 H<sub>2</sub> and O<sub>2</sub> were evolved under irradiation of up to 360 and 510 nm, respectively. The visible-light sensitization was attributed to the S 3p states, which increased the width of the valence band itself and caused a decrease in the band gap energy.<sup>484</sup> Ishikawa and co-workers were the first to report a Ti-based oxysulfide, Sm<sub>2</sub>Ti<sub>2</sub>S<sub>2</sub>O<sub>5</sub>, as a visible-light-driven photocatalyst for hydrogen or oxygen production from aqueous solutions containing a sacrificial electron donor or acceptor. 485-487 It was found that the S 3p orbitals constituted the upper part of the valence band and that these orbitals made an essential contribution to the small band gap energy. Subsequently, it was proposed that the members of the series  $Ln_2Ti_2S_2O_5$  (Ln = Pr, Nd, Sm, Gd, Tb, Dy, Ho, and Er) would also function as visible-light-responsive photocatalysts for water splitting. The photocatalytic activity of [Pr,Nd,Er]<sub>2</sub>Ti<sub>2</sub>S<sub>2</sub>O<sub>5</sub>, containing sulfur defects and Ti as Ti<sup>3+</sup> was lower than that of the other Ln<sub>2</sub>Ti<sub>2</sub>S<sub>2</sub>O<sub>5</sub> forms. It appeared that the electronic band structure was strongly affected by the lanthanoid ions.488

Recently, a number of La-based oxysulfides were shown to catalyze H<sup>+</sup> reduction to form H<sub>2</sub> and/or water oxidation to form O<sub>2</sub> under visible irradiation in the presence of a sacrificial electron donor (methanol, Na<sub>2</sub>S-Na<sub>2</sub>SO<sub>3</sub>) and acceptor (Ag<sup>+</sup>), respectively. The valence bands of these photocatalysts involved overlap of the O 2p and S 3p orbitals, where the higher-energy region mainly consisted of S 3p orbitals and the lower-energy region was mainly composed of O 2p orbitals. 489-491 Thus, it was deduced that the valence bands of photocatalysts consisting of S 3p orbitals, instead of O 2p orbitals, will result in the formation of narrow band gaps. From the point of view of total substitution of  $O^{2-}$  by S<sup>2-</sup> in oxide semiconductors, some novel ternary sulfide systems, including Zn(Cu)-In-S,  $^{375,376,492-495}$  Ag-Ga(In)-S,  $^{496,497}$  Cu-Ga-S(Se),  $^{498,499}$  and Na-In-S,  $^{500}$  have been developed as visible-light-driven photocatalysts for hydrogen production from aqueous solution containing a sacrificial reagent, such as Na<sub>2</sub>S/Na<sub>2</sub>SO<sub>3</sub>. In follow-up studies, different modification technologies were used to improve the photocatalytic activities of these single-phase sulfides.

Jang et al. were the first to obtain C-doped ZnS by the calcination of a  $ZnS(en)_{0.5}$  complex (en = ethylenediamine), in which carbon interacted with Zn through Zn-C bond formation.<sup>501</sup> The participation of C 2p orbitals in the formation of the upper valence band should be responsible for the reduced band gap and the visible-light absorption of ZnS. This was confirmed by the fact that C-doped ZnS exhibited photoactivity for hydrogen production via water splitting under visible-light irradiation.

## 4.1.3. Metal/Nonmetal-Ion Codoping

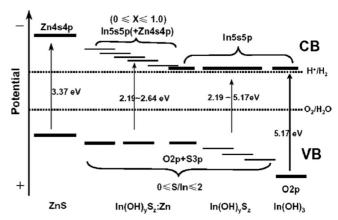
Recently, metal/nonmetal-ion-codoped semiconductor systems have been employed as photocatalysts with the objective of improving their photocatalytic activity under visible-light irradiation. The modification of TiO<sub>2</sub> by codoping with metal and nonmetal ions was frequently used to improve the photocatalytic activity, especially for pollutant degradation. For instance, (Ce + C, I, N, or B),  $^{502-505}$  (Fe + N or C),  $^{506,507}$  (Bi + S, C, or N),  $^{508,509}$  (Ni + B or N),  $^{510,511}$  (La + N, I, or S), $^{512-514}$  and (Eu, Ta, Mo, Pt, or W + N) $^{515-519}$  codoped TiO<sub>2</sub>, without exception, exhibited enhanced visible-light photocatalytic activity compared to their undoped counterparts. SrTiO<sub>3</sub> codoped with La and N was also reported as having greater visible-light photocatalytic activity than the unmodified SrTiO<sub>3</sub>.520-522 However, only a few studies reported metal/nonmetal ion codoped photocatalyst systems with modified conduction and valence band used for visiblelight hydrogen production from water. Gai et al. proposed to dope TiO<sub>2</sub> using charge-compensated donor-acceptor pairs such as (N + V), (Nb + N), (Cr + C), and (Mo + C).  $^{523}$  Among all these systems,  $TiO_2$ : (Mo + C) has the highest positive effect on photocatalytic water splitting. This is because it reduces the band gap to the ideal visible-light region. It does not, however, have much effect on the conduction band position. Using density-functional theory (DFT) calculations, Yin et al. predicted that (Mo, 2N) and (W, 2N) were the best donor-acceptor combinations in the low-alloy concentration regime whereas (Nb, N) and (Ta, N) were the best choice of donor-acceptor pairs in the highalloy concentration regime for solar-driven photoelectrochemical water-splitting.<sup>524</sup>

Sasikala et al. found codoping of TiO<sub>2</sub> with In and N in cationic and anionic sites, respectively, resulted in the narrowing of the band gap of TiO2 compared to In or N doping alone. 525 As a result, the In- and N-codoped samples exhibited enhanced absorption of visible light and improved photocatalytic activity for hydrogen production. Because of the charge-compensation effect from the donor-acceptor codoping, (N, Al)-codoped ZnO obtained by radio frequency (RF) magnetron sputtering method displayed significantly reduced band gap and enhanced photocurrents under visiblelight irradiation, when compared to ZnO and Al-doped ZnO.<sup>526</sup> Tsuji and Kudo synthesized Pb and halogen-codoped ZnS as an active visible-light-driven photocatalyst for H<sub>2</sub> evolution without cocatalysts.<sup>527</sup> Pb doping was responsible for the visible-light absorption, and halogen doping suppressed the formation of nonradiative transition sites in which recombination of photogenerated electrons and holes could occur.

Lei et al. synthesized sulfur-substituted and Zn-doped  $In(OH)_3$  in an aqueous solution of ethylenediamine using the hydrothermal method and investigated its photoactivity for  $H_2$  production under visible-light irradiation. <sup>528</sup> As shown in Figure 13, the band gap of  $In(OH)_3$  was narrowed by the substitution of  $S^{2-}$  for  $OH^-$ . The valence band was composed mainly of S 3p orbitals hybridized with O 2p orbitals. Meanwhile,  $Zn^{2+}$  doping leveled up the conduction band consisting of In 5s5p and In 4s4p orbitals and provided a large thermodynamic driving force for the reduction of water to produce In2.

# 4.2. Controlling Band Structure through Solid Solutions

In addition to using foreign elements for doping, forming solid solutions between wide and narrow band gap semiconductors is another promising method for controlling photocatalyst band structure. Both the band gap and position



**Figure 13.** Proposed band structure of In(OH)<sub>3</sub>, In(OH)<sub>y</sub>S<sub>z</sub>, and In(OH)<sub>y</sub>S<sub>z</sub>:Zn. Reprinted with permission from ref 528. Copyright 2006 Elsevier.

can be adjusted by varying the ratio of the compositions of the narrow and the wide band gap semiconductor in the solid solution. Figure 14 shows the controllable band formation by making a solid solution.

## 4.2.1. (Oxy)sulfide Solid Solutions

 $Cd_xZn_{1-x}S$  (1 < x < 1) sulfide solid solutions, formed by combining the wide band gap semiconductor ZnS and the narrow band gap semiconductor CdS, have attracted much research interest for visible-light-driven photocatalysis applications.  $^{529-533}$  It is known that  $Cd_xZn_{1-x}S$  prepared by a coprecipitation method shows quite good photocatalytic activity for hydrogen production from aqueous solutions containing sacrificial reagents S<sup>2-</sup> and SO<sub>3</sub><sup>2-</sup> under visiblelight irradiation.  $^{534,535}$  Zhang et al. found that  $Cd_{1-x}Zn_xS$  solid solution innovatively prepared by the thermal sulfuration of the corresponding mixed oxide precursors displayed a much higher photoactivity than that prepared by the conventional coprecipitation method.<sup>536</sup> The  $Cd_{1-x}Zn_xS$  (x = 0.2) solid solution exhibited the highest quantum yield of 10.23% at 420 nm for hydrogen production even without a noble metal as a cocatalyst.

Since 2004, Kudo et al. have focused their attentions on the development of novel photocatalysts from ZnS (wide band gap) and MInS<sub>2</sub> (M = Cu, Ag; narrow band gap) sulfide solid solutions.<sup>537–539</sup> The energy structures of the solid solutions were found to be controllable by changing the composition. Taking ZnS-AgInS<sub>2</sub> as an example, the diffuse reflectance spectra, as well as the conduction and valence

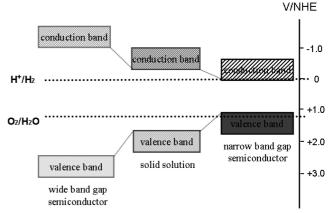


Figure 14. Band structure controlled by making a solid solution.

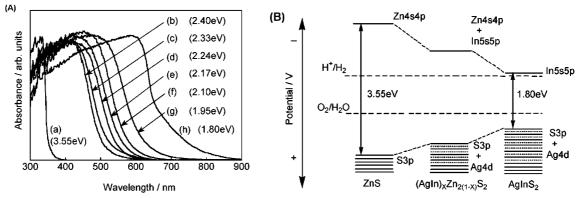


Figure 15. (A) Diffuse reflection spectra of  $(AgIn)_xZn_{2(1-x)}S_2$  solid solutions; the values of x are (a) 0, (b) 0.17, (c) 0.22, (d) 0.29, (e) 0.33, (f) 0.40, (g) 0.5, and (h) 1. (B) Band structures of  $(AgIn)_xZn_{2(1-x)}S_2$  solid solutions, ZnS, and AgInS<sub>2</sub>. Reprinted with permission from ref 537. Copyright 2004 American Chemical Society.

band levels, shifted monotonically with the composition of the solid solution as shown in Figure 15.537 The solid solutions showed high photocatalytic activities for hydrogen evolution under visible-light irradiation. The apparent quantum yields of Pt-loaded (AgIn)<sub>0.22</sub>Zn<sub>1.56</sub>S<sub>2</sub> ( $E_g = 2.33 \text{ eV}$ ) and Pt-loaded (CuIn)<sub>0.09</sub>Zn<sub>1.82</sub>S<sub>2</sub> ( $E_{\rm g}$  = 2.35 eV) amounted to 20%<sup>537</sup> and 12.5% at 420 nm,<sup>538</sup> respectively. Subsequently, they developed a ZnS-CuInS2-AgInS2 solid solution, which proved a highly active visible-light-driven photocatalyst. It was demonstrated that the ZnS-CuInS<sub>2</sub>-AgInS<sub>2</sub> solid solution exhibited an absorption band over a wider wavelength range and a higher photocatalytic activity for hydrogen evolution under the irradiation from a solar simulator compared to the previously reported  $ZnS-MInS_2$  (M = Cu or Ag) solid solutions. This was probably due to its narrower band gap ( $Cu_{0.25}Ag_{0.25}In_{0.5}ZnS_2$ :  $E_{\rm g} = 1.77 \text{ eV}$ ) derived from interactions between the Cu 3d and Ag 4d orbitals involved in the valence band. 540,541

Another series of novel stannite-type complex sulfide solid solutions with ZnS as the wide band gap semiconductor,  $A_2^I - Zn - A^{IV} - S_4$  ( $A^I = Cu$  and Ag;  $A^{IV} = Sn$  and Ge) was investigated by the same group, also for photocatalytic hydrogen evolution under visible-light irradiation. Among them, Ag<sub>2</sub>ZnSnS<sub>4</sub> was the most active, with a quantum yield of ~3% at 500 nm. Cu<sub>2</sub>ZnGeS<sub>4</sub> also showed a high activity and had greater stability. 542 The recently reported AGa<sub>2</sub>In<sub>3</sub>S<sub>8</sub> solid solutions formed from  $AGa_5S_8$  and  $AIn_5S_8$  (A = Cu or Ag) gave relatively high quantum yields. They were active for solar H<sub>2</sub> production from aqueous solutions containing electron donors. The apparent quantum yields of Rh(1.5 wt %)/CuGa<sub>2</sub>In<sub>3</sub>S<sub>8</sub> and Rh(0.5 wt %)/AgGa<sub>2</sub>In<sub>3</sub>S<sub>8</sub> were 15% at 560 and 460 nm, respectively.<sup>543</sup> Jang et al. developed novel  $AgGa_{1-x}In_xS_2$  solid solutions as visible-light-driven photocatalysts for hydrogen evolution in the presence of an electrolyte (Na<sub>2</sub>S/Na<sub>2</sub>SO<sub>3</sub>).<sup>544</sup> The concentration-dependent band gap tuning was achieved in AgGa<sub>1-x</sub>In<sub>x</sub>S<sub>2</sub> (0 < x < 1), which was found to vary from 2.6 to 1.9 eV due to the participation of the In 5s5p orbitals in the formation of the lower region of the conduction band. The photocatalytic activity for hydrogen production over  $AgGa_{1-x}In_xS_2$  showed the maximum value for AgGa<sub>0.9</sub>In<sub>0.1</sub>S<sub>2</sub>. Chen and co-workers synthesized two nanoporous solid solutions, i.e.,  $ZnS-In_2S_3-Ag_2S^{545}$  and  $ZnS-In_2S_3-CuS$ ,  $^{546}$  using a facile template-free method. Both of them showed relatively high visible-light activities for photocatalytic H2 evolution from water containing sacrificial reagents of Na<sub>2</sub>SO<sub>3</sub> and Na<sub>2</sub>S even in the absence of cocatalysts. The quantum yields were 19.8% and 22.6% at 420 nm, respectively. 545,546 Li et al. successfully prepared a zincoxysulfide  $(ZnS_{1-x-0.5y}O_x(OH)_y)$ solid solution with the valence band formed by S 3p and O 2p hybrid orbitals. 547 It displayed excellent photocatalytic activity for hydrogen evolution, and even without a noble metal cocatalyst, the apparent quantum yield was ca. 3.0% under visible-light irradiation.

#### 4.2.2. Oxide Solid Solutions

The photophysical and photocatalytic properties of oxide semiconductors with similar crystal structures were also studied to explore the possibility of energy structure control using solid solutions. <sup>157,256,548–550</sup> Kudo and Mikami reported the successive changes in absorption, photoluminescence spectra, and conduction band levels of In<sub>2</sub>O<sub>3</sub>-Ga<sub>2</sub>O<sub>3</sub> solid solutions.<sup>256</sup> Moreover, they found that both the band gap and conduction band level of the  $Sr_2(Ta_{1-x}Nb_x)_2O_7$  solid solution with a layered perovskite structure could be controlled by changing the ratio of niobium to tantalum. All the  $Sr_2(Ta_{1-x}Nb_x)_2O_7$  photocatalysts studied had band gaps larger than 3.9 eV and only exhibited photocatalytic activities for water splitting into H<sub>2</sub> and O<sub>2</sub> under UV irradiation.<sup>223</sup> Meanwhile, Zou and co-workers prepared BiTa<sub>1-x</sub>Nb<sub>x</sub>O<sub>4</sub> (0 < x < 1) solid-solution photocatalysts using solid-state reactions. 200,551 These produced hydrogen both from aqueous CH<sub>3</sub>OH/H<sub>2</sub>O solutions and from pure water under UV irradiation. Although none of these photocatalysts exhibited visible-light-driven activities, these studies provided some important information for developing novel visible-lightdriven photocatalysts derived from oxide solid solutions that could be used for water splitting.

Luan et al. prepared  $Bi_xIn_{1-x}TaO_4$  (0 < x < 1) solid solutions using a solid-state reaction. 552 The band gaps of the  $Bi_xIn_{1-x}TaO_4$  (x = 0.2, 0.5, and 0.8) photocatalysts were estimated at about 2.86, 2.71, and 2.74 eV, respectively. It was suggested that the band structure consisted of a conduction band of mainly Ta 5d, In 5p, and In 5s orbitals, whereas the valence band was principally O 2p, Bi 6s, and Bi 6p orbitals. Under visible-light irradiation ( $\lambda > 420$  nm), H<sub>2</sub> and O<sub>2</sub> were evolved from CH<sub>3</sub>OH and AgNO<sub>3</sub> aqueous solutions, respectively, using the  $Bi_xIn_{1-x}TaO_4$  photocatalysts. Yi and Ye prepared  $Na_{1-x}La_xTa_{1-x}Co_xO_3$  monophase solid solutions by conventional solid-state reactions.<sup>553</sup> The Na<sub>0.9</sub>La<sub>0.1</sub>Ta<sub>0.9</sub>Co<sub>0.1</sub>O<sub>3</sub> photocatalyst exhibited the highest performance of H<sub>2</sub> evolution (4.34 µmol/h) under visiblelight irradiation. The photocatalytic activities of  $Na_{1-x}La_x$ - $Ta_{1-x}Co_xO_3$  series were mainly attributed to the hybridization of the Co 3d and O 2p orbitals.

**Figure 16.** (A) Schematic band structures. Reprinted with permission from ref 569. Copyright 2005 American Chemical Society. (B) UV—visible diffuse reflectance spectra of  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$ . Reprinted with permission from ref 51. Copyright 2007 American Chemical Society.

Since the divalent metals in scheelite-type molybdates are easily exchangeable with other elements, Yao and Ye developed new scheelite solid-solution photocatalysts by combining the scheelite molybdates (CaMoO<sub>4</sub> and (Na<sub>0.5</sub>Bi<sub>0.5</sub>)MoO<sub>4</sub>) with the narrow band gap semiconductor BiVO<sub>4</sub>.554,555 These crystallized with similar crystal structures. Both the resulting CaMoO<sub>4</sub>-BiVO<sub>4</sub> and (Na<sub>0.5</sub>Bi<sub>0.5</sub>)-MoO<sub>4</sub>-BiVO<sub>4</sub> solid solutions showed high activities for photocatalytic  $O_2$  evolution under visible-light irradiation. In fact, they were even better than that of monoclinic BiVO<sub>4</sub>, which itself is a well-known efficient visible-light-driven photocatalyst. In the study of Li et al., another series of novel solid-solution photocatalysts Na(Bi<sub>x</sub>Ta<sub>1-x</sub>)O<sub>3</sub> were successfully prepared using a simple hydrothermal method, and Na(Bi<sub>0.08</sub>Ta<sub>0.92</sub>)O<sub>3</sub> showed the highest photocatalytic activity for hydrogen production under visible light.<sup>556</sup> This was attributed to the band structure formed by a hybrid conduction band of the (Bi 6s + 6p + Ta 5d) orbitals.

Wang et al. developed a novel series of solid-solution semiconductors  $(AgNbO_3)_{1-x}(SrTiO_3)_x$  (0 < x < 1) as visiblelight-active photocatalysts for efficient O2 evolution and decomposition of organic pollutants. 557,558 The modulation of the band structure (band gap energy, band edge positions, etc.) depends on the extent of the orbital hybridization between both the Ag 4d and O 2p orbitals, as well as that between the Nb 4d and Ti 3d orbitals. As a result of competition between the absorption ability to visible light and the reductive/oxidative abilities, the highest visible-light activities for both O<sub>2</sub> evolution and decomposition of gaseous 2-propanol were realized over (AgNbO<sub>3</sub>)<sub>0.75</sub>(SrTiO<sub>3</sub>)<sub>0.25</sub>. The solid solution of Y<sub>2</sub>WO<sub>6</sub> and Bi<sub>2</sub>WO<sub>6</sub>, BiYWO<sub>6</sub>, was found by Liu et al. to have the photocatalytic ability to completely and stoichiometrically split water into H<sub>2</sub> and O<sub>2</sub> ratio under visible light up to  $\lambda < 470$  nm.<sup>559</sup> It was suggested that the Bi 6s and Y 4d orbitals contributed to a new valence band and conduction band, respectively. It was expected that, because of the flexible structure of the solid solution, the activity would be favorably promoted by changing the ratio of Y<sub>2</sub>WO<sub>6</sub> and Bi<sub>2</sub>WO<sub>6</sub>. Soon after, they discovered another Bi<sub>0.5</sub>Dy<sub>0.5</sub>VO<sub>4</sub> solid solution composed of BiVO<sub>4</sub> and DyVO<sub>4</sub> that responded to visible light up to 450 nm and completely split water into  $H_2$  and  $O_2$ .<sup>560</sup>

#### 4.2.3. Oxynitride Solid Solutions

Starting from the colored oxynitride LaTiO<sub>2</sub>N, new perovskite-type solid solutions LaTiO<sub>2</sub>N-ATiO<sub>3</sub> (A = Sr, Ba) were prepared by the thermal ammonolysis method. Set Narrowed band gaps were achieved by both lowering the bottom of the conduction band and raising the top of the valence band as x increased. Thus,  $(SrTiO_3)_{1-x} \cdot (LaTiO_2N)_x$ 

presented suitable band positions for photocatalytic water splitting into hydrogen and oxygen under visible-light irradiation.

Since 2005, Domen's group has systematically investigated on GaN-ZnO solid solutions as potentially new efficient photocatalysts capable of decomposing water into hydrogen and oxygen stoichiometrically under visible-light irradiation. 563-572 The solid solution of GaN and ZnO,  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$ , should have band gaps greater than 3 eV, because of the large band gap energies of both GaN and ZnO (>3 eV). However, the p-d repulsion between the N 2p and Zn 3d orbitals shifted the valence-band maximum upward without affecting the conduction-band minimum. This resulted in a narrowing of the band gap of GaN-ZnO solid solution as schematically depicted in Figure 16.569,573 The result, therefore, was that the band positions of  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  were suitable for overall water splitting under visible-light irradiation. The quantum efficiency at 420–440 nm was  $\sim 2.5\%$ , when  $Rh_{2-\nu}Cr_{\nu}O_3$  was loaded as a cocatalyst. 564 The photocatalytic performance of Rh<sub>2-y</sub>Cr<sub>y</sub>O<sub>3</sub>/  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  was due to the fact that the charge recombination was prevented, and there was enhanced reactivity of photoexcited holes in the O<sub>2</sub> evolution reaction.<sup>574</sup> Moreover, the visible-light-driven photocatalytic activity of  $Rh_{2-\nu}Cr_{\nu}O_3/(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  was further improved by a postcalcination treatment through reduction of the density of the zinc- and/or oxygen-related defects that functioned as recombination centers for photogenerated electrons and holes in the material. The maximum quantum efficiency obtained by postcalcination treatment is ca. 5.9% at 420-440 nm.<sup>575</sup> This was about an order of magnitude higher than the photocatalytic activity of previously discussed photocatalysts used in overall water splitting under visible light (Ni-InTaO<sub>4</sub>: 0.66% at 402 nm;<sup>353</sup> BiYWO<sub>6</sub>: 0.17% at 420 nm<sup>559</sup>). In addition, an appropriate amount of electrolyte (e.g., NaCl and Na<sub>2</sub>SO<sub>4</sub>) in the photocatalytic reactant solution proved quite effective in enhancing overall water splitting using  $Rh_{2-\nu}Cr_{\nu}O_3$ -loaded  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$ . 576

Following the development of the  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  solid solution,  $(Zn_{1+x}Ge)(N_2O_x)$ ,  $^{577-580}$  the solid solution of Zn-GeN<sub>2</sub> and ZnO was developed by Domen's group and found to be another active and stable photocatalyst for overall water splitting under visible-light irradiation. Similar to the  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  solid solution, the band gap narrowing of  $(Zn_{1+x}Ge)(N_2O_x)$  was also attributed to the p-d repulsion between the N 2p and Zn 3d orbitals. Thus, the visible-light-driven overall water splitting catalyzed by  $(Zn_{1+x}Ge)(N_2O_x)$  proceeded via band gap photoexcitation from the valence band formed by the N 2p, O 2p, and Zn 3d orbitals to the conduction band consisting of Ge 4s and 4p hybridized

orbitals. Instead of the successful formation of the ideal solid solution between  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  and InN, the obtained Ga-Zn-In mixed oxynitride also behaved as a visible-lightdriven photocatalyst for water splitting, but it photocatalyzed  $H_2$  and  $O_2$  evolution only in the presence of an appropriate electron donor or acceptor.<sup>581</sup>

## 4.3. Dye Sensitization to Harvest Visible Light

Dye sensitization is another powerful strategy to enable the visible-light harvesting of wide band gap semiconductors.<sup>582–584</sup> It has been actively studied in solar energy conversion systems involving dye-sensitized solar cells<sup>585–591</sup> and dye-sensitized photocatalytic reactions. 592-595 Since the breakthrough work by O'Regan and Grätzel in 1991,596 the dye-sensitized solar cell has attracted much attention, owing to its stability, low cost, and device efficiency. Powerconversion efficiencies of >11% have been achieved. 597-600 The operating principle of dye-sensitizated solar cells usually involves the excitation of the dye and then charge transfer from the dye to the semiconductor. 601 In addition, the application of dye-sensitized solar cells to water splitting has also been achieved. 602 Since the electron-transfer step resembles that in dye-sensitized solar cells, dye-sensitized semiconductors could also function as photocatalysts for water splitting under visible-light irradiation.

Figure 17 illustrates the basic principle of dye-sensitized photocatalytic H<sub>2</sub> production from water. Photoexcitation of the dye adsorbed onto the semiconductor leads to the injection of electrons into the conduction band of the semiconductor. The electrons are consumed by the reduction of water to produce H<sub>2</sub>. The oxidized dye molecules are subsequently reduced and then regenerated by accepting electrons from the electron donor. 603,604

### 4.3.1. Sensitization Using Ruthenium Complex Dyes

Photocatalytic hydrogen production systems in which ruthenium(II) complex dyes sensitize wide band gap semiconductors to visible light have been the focus of intensive research for many years. In the early 1980s, Grätzel and co-workers<sup>70,592,605–607</sup> succeeded in decomposing water by visible light using Ru(bpy)<sub>3</sub><sup>2+</sup> and its amphiphilic derivatives as sensitizers. Pt/RuO2-loaded TiO2 particles proved particularly effective in those systems, acting as photocatalysts for water-splitting process.

Nakahira et al. found that Pt/TiO2 sensitized with a polymer-pendant Ru(bpy)<sub>3</sub><sup>2+</sup> complex was effective in H<sub>2</sub> evolution in the presence of the sacrificial donor (EDTA) under visible-light irradiation. 608 Adsorption of the Ru(dcbpy)<sub>3</sub> sensitizer onto platinized TiO<sub>2</sub> particles (Pt/TiO<sub>2</sub>) also

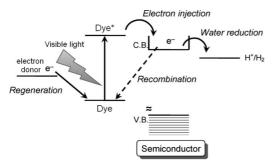


Figure 17. Basic principle of dye-sensitized photocatalytic H<sub>2</sub> production from water. Reprinted with permission from ref 604. Copyright 2008 American Chemical Society.

led to H<sub>2</sub> evolution by visible light, again with a sacrificial donor.<sup>609</sup> Hirano et al. found that, when Ru(bpym)<sub>3</sub><sup>2+</sup> was used as the visible-light sensitizer, Pt/TiO2 showed much higher efficiency for  $H_2$  production than  $Ru(bpy)_3^{2+.610}$  This may be due to the greater affinity of  $Ru(bpym)_3^{2+}$  on the TiO<sub>2</sub> surface. Dhanalakshmi et al. found that, when TiO<sub>2</sub> and ZnO were sensitized with a new sensitizer ([Ru-(dcbpy)<sub>2</sub>(dpq)]<sup>2+</sup>), they displayed extremely stable and effcient photocatalytic activity for hydrogen production under visible-light irradiation from water even in the absence of an electron donor.<sup>611</sup> However, interestingly, the photocatalytic activities of [Ru(dcbpy)<sub>2</sub>(dpq)]<sup>2+</sup>-sensitized Pt/ZnO and Cu/ZnO were found to be 10 times lower than those of [Ru(dcbpy)<sub>2</sub>(dpq)]<sup>2+</sup>-sensitized Pt/TiO<sub>2</sub> and Cu/TiO<sub>2</sub>, respectviely. Bae et al. compared the photocatalytic activities and stabilities of two Ru(bpy)<sub>3</sub><sup>2+</sup>-sensitized TiO<sub>2</sub> photocatalysts with different surface-anchoring groups (carboxylate and phosphonate), i.e.,  $c-Ru(bpy)_3^{2+}$  and  $p-Ru(bpy)_3^{2+}$ , respectively. Pt/TiO<sub>2</sub>/p-Ru(bpy)<sub>3</sub><sup>2+</sup> anchored through phosphonate groups exhibited higher photocatalytic activity for hydrogen production from water than Pt/TiO<sub>2</sub>/c-Ru(bpy)<sub>3</sub><sup>2+</sup> anchored through carboxylate groups.<sup>612</sup> Because of the more rapid regeneration of p-Ru(bpy)<sub>3</sub><sup>2+</sup> than c-Ru(bpy)<sub>3</sub><sup>2+</sup>, the phosphonate group seemed to be better than the carboxylate group as a ruthenium sensitizer linkage to the TiO<sub>2</sub> surface in aqueous environment. However, both of them turned out to be unstable in water, not only under visible-light illumination but also in the dark.

Abe et al. studied the effect of the energy gap between the I<sup>3-</sup>/I<sup>-</sup> redox potential and the highest occupied molecular orbital (HOMO) level of the dyes on the photocatalytic activity of Ru complex dye-sensitized Pt/TiO2 in wateracetonitrile solutions. 613 The system included I as an electron donor. It was found that a sufficiently positive HOMO level, compared to the oxidation potential of I<sup>-</sup>, was necessary in order to produce a relatively high rate of H<sub>2</sub> evolution over Ru complex dye-sensitized Pt/TiO<sub>2</sub>. Peng and co-workers investigated the influence of different ruthenium(II) bipyridyl complexes on the photocatalytic H<sub>2</sub> evolution over TiO<sub>2</sub> under visible light.<sup>614,615</sup> When compared to Ru(bpy)<sub>2</sub>(him)<sub>2</sub>-NO<sub>3</sub>- and Ru(dcbpy)<sub>2</sub>(NCS)<sub>2</sub>-sensitized Pt/TiO<sub>2</sub>, Ru<sub>2</sub>(bpy)<sub>4</sub>L<sub>1</sub>-PF<sub>6</sub>-sensitized Pt/TiO<sub>2</sub> displayed higher photocatalytic efficiency and preferable durability for H<sub>2</sub> evolution. This may be related to the dynamic equilibrium between the linkage of the ground dye to TiO2 and the separation of the oxidized dye from TiO<sub>2</sub>. Choi and coworkers investigated the addition of a thin Al<sub>2</sub>O<sub>3</sub> overlayer on the Ru(bpy)<sub>3</sub><sup>2+</sup>-sensitized TiO<sub>2</sub> and found that it significantly increased the visible-light-sensitized activity for hydrogen production. 616 As shown in Figure 18, it was proven that the main role of the Al<sub>2</sub>O<sub>3</sub> barrier layer was to retard the charge recombination between the electron injected from the excited dye and the oxidized dye. They also found that guanidinium cations adsorbed on nafion-coated Ru(bpy)<sub>3</sub><sup>2+</sup>sensitized TiO<sub>2</sub> retarded the charge recombination between the electrons injected into the TiO<sub>2</sub> conduction band and the oxidized dye molecules. This resulted in enhanced photocatalytic hydrogen production under visible light.<sup>617</sup>

Reisner et al. constructed a special system consisting of [NiFeSe]—hydrogenase attached to Ru dye-sensitized TiO<sub>2</sub>, with triethanolamine (TEA) as the sacrificial electron donor.618 It showed a high and stable photocatalytic activity for hydrogen generation under visible light. Gurunathan et al. found that the photocatalytic efficiency of SnO<sub>2</sub> could be

**Figure 18.** Schematic illustration of electron transfer/recombination processes occurring on a dye-sensitized  $TiO_2$  particle with  $Al_2O_3$  overlayer for  $H_2$  production. The numbered paths indicate the following: (1) dye excitation; (2) electron injection from the excited dye to  $TiO_2$  CB; (3) electron trapping on Pt; (4) regeneration of the oxidized dye by electron donor (D); and (5) back electron transfer to the oxidized dye (recombination). Reprinted with permission from ref 616. Copyright 2009 American Chemical Society.

greatly improved using  $Ru(bpy)_3^{2+}$  sensitization.<sup>619</sup> The highest photocatalytic activity for hydrogen evolution, with quantum yield of 2.40% at 350 nm, was observed with the  $Pt/SnO_2/RuO_2-Ru(bpy)_3^{2+}-MV^{2+}-EDTA$  system. This may be due to a number of contributing factors, which include the loading of the bifunctional redox catalyst onto  $SnO_2$ , electron injection from the excited  $Ru(bpy)_3^{2+}$  to the conduction band of  $SnO_2$ , electron mediation by  $MV^{2+}$ , and electron donation by EDTA to scavenge the valence band holes.

Furube et al. investigated the dynamics of photoexcited  $Ru(bpy)_3^{2+}$  intercalated into the  $K_4Nb_6O_{17}$  interlayers.<sup>620</sup> Because of the fast and efficient electron transfer between  $Ru(bpy)_3^{2+}$  and  $K_4Nb_6O_{17}$ , the transient bleaching of the  $Ru(bpy)_3^{2+}$  band showed fast and nonexponential decay. This differed from the behavior of Ru(bpy)<sub>3</sub><sup>2+</sup> in water, and led to the photocatalytic hydrogen evolution over Ru(bpy)<sub>3</sub><sup>2+</sup> intercalated K<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub> from aqueous solution containing an electron donor.621 Later, Mallouk and co-workers studied different layered oxide semiconductors as the building blocks for visible-light H<sub>2</sub> production from water using a ruthenium complex as the photosensitizer. 604,622-626 The Ru(bpy)<sub>3</sub><sup>2+</sup>sensitized K<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub> nanoscrolls exhibited higher photocatalytic activity than Ru(bpy)<sub>3</sub><sup>2+</sup>-sensitized lamellar K<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub> and TiO<sub>2</sub>. This was primarily due to the high surface area of the nanoscrolls and their excellent ability to bind Ru- $(bpy)_3^{2+}$ . This led to the facile transfer of electrons from the sensitizer to the Pt catalyst islands via the single-crystalline nanoscrolls. The authors claimed that the platinized H<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub> nanoscrolls were a slightly better electron transfer mediator than acid-restacked HCa<sub>2</sub>Nb<sub>3</sub>O<sub>10</sub> nanosheets. The apparent quantum yield of visible-light photocatalytic hydrogen production over Pt/H<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub> nanoscrolls was ~25% at 450  $\pm$  20 nm, when sensitized by Ru(bpy)<sub>3</sub><sup>2+</sup> complexes. 623,624

## 4.3.2. Sensitization Using Other Transition-Metal Complex Dyes

Many other transition-metal complexes such as polypyridine complexes, alizarine, phthalocyanine, and metalloporphyrins with metal centers including Pt(II), Co(II), Zn(II), and Cr(III) have also been widely used as sensitizers for wide band gap photocatalysts to improve their optical and photocatalytic properties for hydrogen evolution in the visible-light region.

In the study of Zakharenko et al.,  $^{627}$  the Pt(alizarine)<sub>2</sub> complex adsorbed onto the surface of titanium dioxide was found to cause spectral sensitization of TiO<sub>2</sub> and lead to photocatalytic hydrogen production from water in the visible-light region. Two new platinum(II) terpyridyl acetylide complexes were synthesized and used by Jarosz et al.  $^{628}$  They were used to sensitize platinized TiO<sub>2</sub> for the photogeneration of H<sub>2</sub> using visible light. However, the long-term ability of these complexes to act as sensitizers in this capacity was limited by decomposition after oxidation.

Malinka and co-workers found that  $H_2$  evolved from aqueous suspensions of zinc-porphyrin-sensitized Pt/TiO<sub>2</sub> in the presence of an electron donor, such as EDTA, TEA, and ascorbic and oxalic acids, under irradiation with visible light  $(\lambda > 520 \text{ nm}).^{629,630}$  Zinc porphyrins displayed increased photostability when absorbed onto the semiconductor surface, and the photocatalytic activity of zinc-porphyrin-sensitized Pt/TiO<sub>2</sub> was greatly affected by factors such as concentrations of platinum, sensitizer and electron donor, the pH and temperature of the suspension. Shimidzu et al. found that the visible-light illumination  $(500 < \lambda < 700 \text{ nm})$  of a water-soluble porphyrin—zinc complex and TEA in aqueous solution with suspended particulate ZnO/Pt resulted in efficient hydrogen production with a quantum yield higher than 3.0% at  $548 \pm 3 \text{ nm}.^{631}$ 

Nada et al. found that, when copper phthalocyanine was used as a photosensitizer, it was effective for hydrogen production over RuO<sub>2</sub>/TiO<sub>2</sub> using UV/solar light irradiation. 632 In fact, copper phthalocyanine exhibited higher efficiency compared to other sensitizers such as ruthenium bipyridyl. Since the splitting of pure water using a dyesensitized oxide semiconductor has not been achieved previously, it is worth mentioning that Hagiwara et al. were the first to report the sensitizing effects of porphyrinoids on the photocatalytic splitting of H<sub>2</sub>O into H<sub>2</sub> and O<sub>2</sub> by Pt/ KTa(Zr)O<sub>3</sub>.<sup>633</sup> The addition of porphyrinoids to KTa(Zr)O<sub>3</sub> effectively increased the activity of this compound for pure water splitting. Among the porphyrinoids examined, cyanocobalamin showed the strongest effect on photocatalytic water splitting. Moreover, the metal cation of the porphyrin complex had a large influence on the production rates of H<sub>2</sub> and O<sub>2</sub>. For example, Pt/KTa(Zr)O<sub>3</sub> sensitized with Crtetraphenylporphyrin showed much higher photoactivity than a catalyst sensitized with Co-tetraphenylporphyrin.

Rayalu and co-workers reported that, because of the synergistic effect of Co<sup>2+</sup> and heteropolyacid (HPA), HPA/Co<sup>2+</sup> sensitized TiO<sub>2</sub> showed excellent photocatalytic activity for hydrogen evolution under visible light.<sup>634–636</sup> The presence of Co<sup>2+</sup> not only enhanced the visible-light absorption property but also played a very important role in triggering the overall catalytic efficiency. Recently, a TiO<sub>2</sub> photocatalyst sensitized by grafting surface complexes of lacunary Wells–Dawson W-contained heteropoly blue (HPB) exhibited 20.0% quantum efficiency for H<sub>2</sub> evolution under 670 nm irradiation in the presence of glycerol as the electron donor.<sup>637,638</sup>

#### 4.3.3. Sensitization Using Metal-Free Dyes

Because some transition-metal complexes, especially Ru complexes, are quite expensive, metal-free dyes, such as porphine dyes, xanthene dyes, melocyanine dyes, and coumarin dyes, have all been investigated.

Houlding and Grätzel reported the excellent ability of 8-hydroxyquinoline-sensitized anatase TiO<sub>2</sub> powders to

Figure 19. Eosin Y-sensitized TiO<sub>2</sub> for H<sub>2</sub> evolution system in two-step water splitting. Reprinted with permission from ref 646. Copyright 2000 Elsevier.

generate hydrogen in a sacrificial water-reduction system containing EDTA as the electron donor using visible light.<sup>639</sup> Gurunathan et al. investigated the photocatalytic activities for hydrogen production by Pt/SnO<sub>2</sub> sensitized with various metal-free photosensitizers. 619 These included acriflavin, Eosin Blue, rhodamine B, Rose Bengal, and fluorescein. Of all the photosensitizers used, Eosin Blue showed the maximum efficiency to sensitize SnO<sub>2</sub> and evolved a greater amount of hydrogen, even higher than Ru(bpy)<sub>3</sub><sup>2+</sup>. Hirano et al. found that some porphine derivatives, such as tetrakis-(4-carboxyphenyl)porphine (TPPC), tetrakis-(4-sulfonicphenyl)porphine (TPPS), and tetrakis-(4-pyridylphenyl)porphine (TPPPy) were all effective as photosensitizers. 610 It was also found that TPPC/TiO<sub>2</sub>/Pt exhibited the highest activity for hydrogen evolution from water in the presence of EDTA as the electron donor. The principal reasons proposed for this enhanced activity include the degree of adsorption onto TiO<sub>2</sub>, the different electron densities in excited states, or redox potential. Chatterjee claimed that the efficacy of the photocatalytic system toward hydrogen evolution was quite possibly governed by the excited-state redox properties of the dyes adsorbed onto the surface of the TiO2 photocatalyst.<sup>640</sup> A binaphthol complex was applied by Ikeda et al. to sensitize TiO<sub>2</sub> for photocatalytic hydrogen evolution from TEA solutions under visible-light irradiation.<sup>641</sup> The photonic efficiency of the photocatalytic reaction at 450 nm was estimated to be 0.02% for the binaphthol-modified TiO<sub>2</sub> loaded with 0.1 wt % Pt.

Abe et al. investigated the significant influence of solvent on hydrogen production over metal-free dye sensitized Pt/ TiO<sub>2</sub> photocatalysts from a water—acetonitrile mixed solution containing an iodide electron donor under visible-light irradiation. 613,642-644 The rates of H<sub>2</sub> evolution decreased with increasing water content in the mixed solutions. This was due to the decreasing energy gap between the redox potential of I<sub>3</sub><sup>-</sup>/I<sup>-</sup> and the HOMO level of the dye. However, when compared to merocyanine sensitized Pt/TiO2, a relatively high rate of H<sub>2</sub> evolution was still observed over a coumarinsensitized Pt/TiO<sub>2</sub> photocatalyst even in aqueous solution. This was because coumarin had a sufficiently positive HOMO level compared to the oxidation potential of I<sup>-</sup>.

Shimidzu et al. reported extensive studies on visible-lightinduced photocatalytic water reduction using some xanthene dye-sensitized particulate Pt-loaded semiconductor catalysts.<sup>645</sup> They found that heavy halogenated xanthene dyes, e.g., Rose Bengal, erythrosine, and eosine bluish, showed high quantum yields of hydrogen production for sensitized platinized semiconductors. However, they tended to photodehalogenate, whereas nonhalogenated xanthene dyes, e.g., uranine and rhodamine B, exhibited high durability against any photodeterioration but only moderate catalytic activities. Interestingly, it was also found that the addition of external halide ions improved the photocatalytic activities of nonhalogenated xanthene dye-sensitized platinized semiconductors. Abe et al. constructed a stable dye-sensitized photocatalyst system in water by a chemical fixation of xanthene dyes onto platinized TiO<sub>2</sub> particles using a silane-coupling reagent.<sup>646</sup> The Eosin Y-fixed Pt-TiO<sub>2</sub> exhibited high efficiency and quite steady H<sub>2</sub> production from aqueous TEA solution under visible-light irradiation for long periods. The quantum yield at 520 nm was determined to be  $\sim$ 10%. Figure 19 shows the Eosin Y-sensitized TiO<sub>2</sub> in a two-step water-splitting system for H<sub>2</sub> evolution.

Lu and co-workers systematically studied the Eosin Y-sensitized Ti-based photocatalysts for hydrogen generation from di- or triethanolamine aqueous solution under visiblelight irradiation ( $\lambda \ge 420$  nm). <sup>647–653</sup> Eosin Y-sensitized Rh/ TiO<sub>2</sub>,<sup>647</sup> Pt/Na<sub>2</sub>Ti<sub>2</sub>O<sub>4</sub>(OH)<sub>2</sub>,<sup>649</sup> and Pt/Ti-MCM-41<sup>652</sup> showed relatively excellent photocatalytic activities and long-term stability for hydrogen production. Average quantum yields higher than 10% in the visible-light region were found. In particular, Eosin Y-sensitized Pt/Na<sub>2</sub>Ti<sub>2</sub>O<sub>4</sub>(OH)<sub>2</sub>, with a quantum yield up to 14.97%, showed good photocatalytic stability for hydrogen production over 100 h after 10 consecutive runs. 649 Subsequently, a much higher quantum yield of 17.36% was obtained over Eosin Y-sensitized Ptloaded long titanic acid nanotubes. However, the destruction of the nanotube structure after long-time reaction led to a decrease in the hydrogen generation rate. 653 Additionally, the multilayer-Eosin Y-sensitized TiO<sub>2</sub> via Fe<sup>3+</sup> coupling was found to have a high light harvesting efficiency and photocatalytic activity for hydrogen evolution under visible-light irradiation. The maximum apparent quantum yield for hydrogen evolution is 19.1% from aqueous TEA solution. 654 They also achieved efficient photocatalytic water reduction for hydrogen generation over Eosin Y-sensitized Pt/SiO<sub>2</sub><sup>655</sup> and Pt/MWCNT (multiwalled carbon nanotube)<sup>656</sup> in the presence of TEA as the electron donor. Quantum yields up to 10.4% and 12.14%, respectively, were observed.

## 4.4. Developing Novel Single-Phase Visible-Light-Responsive Photocatalysts

Even without using the modification technologies discussed above (ion doping, making solid solutions, and dye sensitization), many novel single-phase visible-light-responsive photocatalysts, mostly heterometallic oxides, have been fabricated successfully using metal-mediated band structure

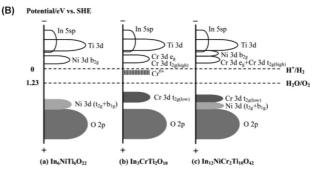


Figure 20. (A) UV—vis absorption spectra of  $In_6NiTi_6O_{22}$ ,  $In_3CrTi_2O_{10}$ , and  $In_{12}NiCr_2Ti_{10}O_{42}$  powder samples at room temperature. The dotted line shows the absorption spectrum of  $In_{12}ZnGa_2Ti_{10}O_{42}$ . (B) Schematic band structures of (a)  $In_6NiTi_6O_{22}$ , (b)  $In_3CrTi_2O_{10}$ , and (c)  $In_{12}NiCr_2Ti_{10}O_{42}$ . Reprinted with permission from ref 666. Copyright 2007 American Chemical Society.

engineering. In this section, we will focus on recent progress in the design of single-phase heterometallic oxides as novel visible-light-responsive photocatalysts. Of particular interest are materials with band structures mediated and band gap narrowed by d-, s-, and f-orbitals from the outer layer orbital configurations of d-block, p-block, and f-block metal ions, respectively.

## 4.4.1. d-block Metal Oxides

The versatile features of the d orbitals of some transition metals for band gap narrowing have been demonstrated by doping such d-block transition metals into wide band gap photocatalysts to form a separate donor level in the forbidden band. 353,354,525 The activities of the resulting doped photocatalysts are usually not high because the doping level can also serve as a recombination center for photogenerated electrons and holes. It is expected, therefore, that d-block transition metals could also contribute to the band gap narrowing of single-phase heterometallic oxide photocatalysts.

In the study of Yin et al., the single-phase visible-lightdriven photocatalyst BaCrO<sub>4</sub>, with a band gap determined to be 2.63 eV, was synthesized by a solid-state reaction method. 657 Since the Cr<sup>6+</sup> 3d orbital was empty, the band structure of BaCrO<sub>4</sub> was defined by the 3d level of Cr<sup>6+</sup> and the O 2p level of the ligand O atom. As a result, the visiblelight transition was ascribed to the electronic excitation from the valence band composed of the O 2p orbitals to the conduction band composed of the Cr<sup>6+</sup> 3d orbitals. Unlike the Cr 3d orbitals in BaCrO<sub>4</sub>, Cr<sup>3+</sup> 3d orbitals in BaCr<sub>2</sub>O<sub>4</sub> were partially filled and split into Cr 3d-t<sub>2</sub> g and Cr 3d-e<sub>g</sub> orbitals. This split in band structure resulted in two cases of electron photoexcitations, namely, the photoexcitation from Cr 3d-t<sub>2 g</sub> to Cr 3d-e<sub>g</sub> and the photoexcitation from O 2p to Cr 3d-t<sub>2</sub> g, evolving H<sub>2</sub> and O<sub>2</sub>, respectively. <sup>658</sup> Further reports of Cr-based oxides with different electronic structures of Cr 3d orbitals were conducted by Ouyang et al.<sup>659</sup> In AgAlO<sub>2</sub>, the valence band and the conduction band were mainly constructed by Ag 4d and O 2p orbitals, and Ag 5s and 5p orbitals, respectively. The band gap was estimated to be 2.95 eV from the UV-vis spectra. When Al element in AgAlO<sub>2</sub> was replaced by Cr, both AgCrO<sub>2</sub> and Ag<sub>2</sub>CrO<sub>4</sub> showed an obvious red shift of the absorption edge. The trivalent Cr ions in AgCrO<sub>2</sub> owned three 3d-electrons, which were split to two groups, whereas the hexavalent Cr ions in Ag<sub>2</sub>CrO<sub>4</sub> had no 3d-electrons. This was the reason AgCrO<sub>2</sub> had double absorption bands and Ag<sub>2</sub>CrO<sub>4</sub> only had a single absorption band in the visible-light region. Cr ions with different 3d orbitals had different effects on the band structure. Cr 3d orbitals in Ag<sub>2</sub>CrO<sub>4</sub> only contributed to the conduction band, whereas Cr 3d orbitals in AgCrO<sub>2</sub> contributed to both valence and conduction bands. This was analogous to the previous reports on BaCrO<sub>4</sub><sup>657</sup> and BaCr<sub>2</sub>O<sub>4</sub>.<sup>658</sup>

Lv et al. developed a new photocatalyst, namely, LiCr- $(WO_4)_2$ , in which the Cr3d level contributed similarly to the formation of both the conduction and valence bands.<sup>660</sup> It showed photocatalytic activity for H<sub>2</sub> evolution under visiblelight irradiation. Ye et al. developed a new series of 3d transition metal Ni-containing photocatalysts, NiM<sub>2</sub>O<sub>6</sub> (M = Nb, Ta), which proved active for water splitting under visible-light irradiation.<sup>661</sup> For NiM<sub>2</sub>O<sub>6</sub>, which contains two kinds of octahedra, NiO<sub>6</sub> and MO<sub>6</sub>, the valence band energy was assumed to arise from both the O 2p orbitals of the NiO<sub>6</sub> and MO<sub>6</sub> octahedra. The valence band showed a value about 1.0 eV more positive than 2.94 (standard hydrogen electrode (SHE)) of O 2p levels. This resulted in the band gap narrowing of NiM<sub>2</sub>O<sub>6</sub>. It is believed that the Ni d-d transition between the Ni 3d-t<sub>2 g</sub> and Ni 3d-e<sub>g</sub> levels also played an important role in the visible-light photoexicitaion and photocatalytic activity of NiM2O6. Similar Ni d-d transitions were also observed in other Ni-containing heterometallic oxides such as BaNi<sub>1/3</sub>Nb(Ta)<sub>2/3</sub>O<sub>3</sub>, <sup>178,225</sup>  $MLaSrNb_2NiO_9$  (M = Na, Cs, H),  $^{662,663}$  and  $Ca_2NiWO_6$ .  $^{664}$ In all these cases, such transitions appeared to be responsible for their visible-light absorption and visible-light-induced photocatalytic activities.

Wang and co-workers systematically investigated the band structures of three oxide semiconductor photocatalysts,  $In_6NiTi_6O_{22}$ ,  $In_3CrTi_2O_{10}$ , and  $In_{12}NiCr_2Ti_{10}O_{42}$ . 665,666 They used different 3d transition metals with partially filled 3d orbitals to clarify the effects of the transition metal cations (Ni and/or Cr) on the photophysical and photocatalytic properties. As shown in Figure 20, the split Ni 3d orbitals in In<sub>6</sub>NiTi<sub>6</sub>O<sub>22</sub> or the split Cr 3d orbitals in In<sub>3</sub>CrTi<sub>2</sub>O<sub>10</sub> formed only a discrete band, whereas the coexistence of both Ni<sup>2+</sup> and Cr<sup>3+</sup> in In<sub>12</sub>NiCr<sub>2</sub>Ti<sub>10</sub>O<sub>42</sub> enabled the formation of continuous conduction and valence bands through the hybridization of the split Ni 3d and Cr 3d orbitals with the Ti 3d/In 5s5p and O 2p orbitals. The formation of the continuous conduction and valence bands not only decreased the band gap energy but also increased the mobility of the photoinduced charge carriers. Thus, the photocatalytic activity of In<sub>12</sub>NiCr<sub>2</sub>Ti<sub>10</sub>O<sub>42</sub> for hydrogen evolution under visiblelight irradiation was improved compared to those of In<sub>6</sub>NiTi<sub>6</sub>O<sub>22</sub> and In<sub>3</sub>CrTi<sub>2</sub>O<sub>10</sub>. Silva et al. synthesized a series of Zn/Ti, Zn/Ce, and Zn/Cr layered double hydroxides (LDH) and tested them for the visible-light photocatalytic oxygen generation.<sup>667</sup> Because of the chemical composition and in contrast to the (Zn/M)LDH (M = Ti, Ce), the (Zn/Cr)LDH

containing Cr exhibited much greater light absorption in the visible region. The most active material was found to be the (Zn/Cr)LDH. Its apparent quantum yields for oxygen generation were 60.9% and 12.2% at 410 and 570 nm, respectively.

Ye et al. found that the 3d transition metal V compound, InVO<sub>4</sub>, was effective for H<sub>2</sub> evolution from pure water under visible-light irradiation. 668 The InVO<sub>4</sub> showed photocatalytic activity up to 600 nm, which is a much wider wavelength range than previously reported for In-based oxides such as InTaO<sub>4</sub> and InNbO<sub>4</sub>. This was as expected from the conduction band composed of the less negative V 3d orbitals in InVO<sub>4</sub>, when compared to the Nb 4d orbitals in InNbO<sub>4</sub> and the Ta 5d in InTaO<sub>4</sub>. It was reported that another vanadate, BiVO<sub>4</sub>, showed visible-light photocatalytic activity for O<sub>2</sub> evolution from an aqueous solution containing either  $Ag^{+}$  or  $Fe^{3+}$  as electron acceptor. <sup>669–674</sup> However, BiVO<sub>4</sub> did not display photocatalytic activity for H<sub>2</sub> evolution. This is because the conduction band composed of the V 3d orbitals was located at less negative potential than for H<sup>+</sup> reduction to H<sub>2</sub>. Judging from the above results, it is possible that, depending on the chemical components and crystal structures of different vanadates, the V 3d level is either less negative or more negative than the H<sub>2</sub> evolution potential. Some other V-containing oxide semiconductors, such as M<sub>2.5</sub>VMoO<sub>8</sub> (M = Mg, Zn), $^{675}$  M<sub>3</sub>V<sub>2</sub>O<sub>8</sub> (M = Mg, Ni, Zn), $^{676}$  CaBiVMO<sub>8</sub> (M = W and Mo), $^{677}$  Bi<sub>2</sub>GaVO<sub>7</sub>, $^{251}$  Bi<sub>2</sub>YVO<sub>8</sub>, $^{252}$  and  $BiM_2VO_6$  (M = Cu and Zn), 678,679 also displayed efficient visible-light absorption and photocatalytic activities. This is because the V 3d orbitals involved in the band structures act to lower down the conduction band, leading to band gap narrowing of these V-containing oxide photocatalysts.

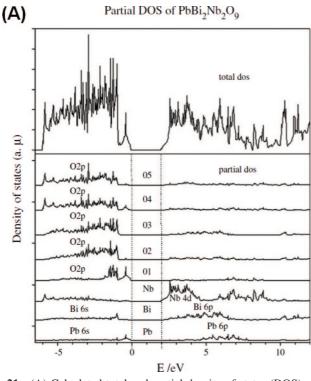
Yin et al. investigated a new series of visible-light-driven photocatalysts  $MCo_{1/3}Nb_{2/3}O_3$  (M = Ca, Sr, and Ba) with an ABO3-type perovskite structure, in which the B site is occupied by Co<sup>2+</sup> and Nb<sup>5+</sup> randomly.<sup>680,681</sup> In these compounds, the Co<sup>2+</sup> 3d states, which were split into two main peaks, the lower t<sub>2 g</sub> state and the higher e<sub>g</sub> state, were strongly hybridized with the O 2p states to form the valence bands. As a result, the observed visible-light absorption for MCo<sub>1/3</sub>Nb<sub>2/3</sub>O<sub>3</sub> was ascribed to the electronic excitations from the Co<sup>2+</sup> t<sub>2 g</sub> state to the Nb<sup>5+</sup> 4d state. Sun et al. reported that, in Bi<sub>5</sub>FeTi<sub>3</sub>O<sub>15</sub>, the Fe 3d band split into two main parts, corresponding to the Fe t<sub>2 g</sub> and the Fe e<sub>g</sub>, respectively.<sup>682</sup> The photoexcitation from the hybridized O 2p + Fe  $t_{2g}$  + Bi 6s orbitals to the Fe eg orbitals were most probably responsible for the visible-light absorption of Bi<sub>5</sub>FeTi<sub>3</sub>O<sub>15</sub>. Because of the Fe 3d orbitals involved in the band structure, Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub> was sensitive to visible light. Its band gap was estimated to be  $\sim$ 1.86 eV from opitical absorption spectra. This could split pure water to generate H<sub>2</sub> under visiblelight irradiation, after Ca<sub>2</sub>Fe<sub>2</sub>O<sub>5</sub> was loaded with NiO.<sup>683</sup>

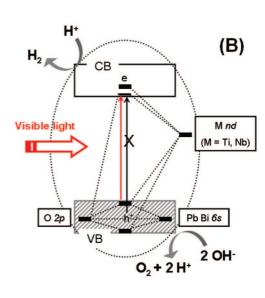
Kato et al. found that the band gaps of AgTaO3 and AgNbO<sub>3</sub> were 0.6 eV smaller than those of NaTaO<sub>3</sub> and NaNbO<sub>3</sub>, respectively, although their crystal structures of  $AgMO_3$  (M = Ta, Nb) were similar to those of NaMO<sub>3</sub>.<sup>242</sup> DFT calculations showed that a hybrid Ag 4d and O 2p orbital formed a valence band at a higher energy level than O 2p orbitals, and this resulted in the decrease in the band gaps of  $AgMO_3$  (M = Ta, Nb). These findings made it possible to develop AgNbO<sub>3</sub> as a good visible-light-driven photocatalyst with the ability to evolve H<sub>2</sub> or O<sub>2</sub> from water in the presence of sacrificial reagents. Similarly, the contribution of Ag<sup>+</sup> to the valence band formation for α-AgVO<sub>3</sub> led to a smaller band gap than in α-NaVO<sub>3</sub>, which was responsible for the unnegligible photocatalytic activity for O<sub>2</sub> evolution under visible-light irradiation.<sup>684</sup> Keeping in mind the valence bands consisting of hybridized Ag 4d and O 2p orbitals, some noval Ag-based metal oxides, such as  ${\rm AgInW_2O_8,^{685}\,\alpha\text{-}AgGaO_2,^{686}\,Ag_2ZnGeO_4,^{687}}$  and  ${\rm AgLi_{1/3}M_{2/3}O_2}$  (M = Ti and Sn),  $^{688}$  were developed as visible-lightdriven photocatalysts. The valence bands of these metal oxides are located at higher energy levels than those consisting of only O 2p orbitals. This is what leads to the narrowed band gaps and hence to the proposed visible-lightdriven photocatalytic activities.

#### 4.4.2. p-block Metal Oxides

In addition to the d-block transition metals, s-orbitals from the outer layer-orbital configurations of p-block metal ions are also effective to mediate the band structures. Tang and co-workers synthesized  $MIn_2O_4$  (M = Ca, Sr, Ba) semiconductors as potential visible-light-photoactive materials containing the InO<sub>6</sub> octahedral structure. 689,690 For these Incontaining oxides, the highest occupied band was composed of O 2p orbitals, which corresponded to the valence band. The lowest unoccupied band which was mainly composed of the In 5s orbitals, which corresponds to the conduction band. Taking this into consideration, the visible-light absorption was attributed to the photoexcition from the O 2p orbital to the In 5s orbital. However, for  $InMO_4$  (M = Ta, Nb), which contains two kinds of octahedra, InO<sub>6</sub> and MO<sub>6</sub>, the valence band was assumed to be a combination of both the O 2p levels of InO<sub>6</sub> and NbO<sub>6</sub> octahedrons. It showed  $\sim$ 1.1 eV more positive than that of the O 2p levels. <sup>691–693</sup> This is quite similar to the case in which the valence band of NiNb<sub>2</sub>O<sub>6</sub> containing NiO<sub>6</sub> and NbO<sub>6</sub> octahedra was attributable to both the O 2p levels of NiO<sub>6</sub> and NbO<sub>6</sub> octahedra. 661,694 The relatively positive valence band energy suggested that both InTaO<sub>4</sub> and InNbO<sub>4</sub> had suitable band structures and could respond to visible light. This was quite different from the visible-light photoexcition from O 2p orbital to In 5s orbital in  $MIn_2O_4$  (M = Ca, Sr, Ba).

Zhou et al. prepared a visible-light-responsive polycrystalline Bi<sub>12</sub>TiO<sub>20</sub> photocatalyst using a simple solid-state reaction between Bi<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> powders.<sup>695</sup> Its band gap was narrowed to be 2.78 eV through the introduction of the p-block transition metal Bi. The band structure of Bi<sub>12</sub>TiO<sub>20</sub> originated from the Ti 3d band as well as the Bi 6s and O 2p hybridized band. This hybridization shifted the valence band upward, causing the band gap of Bi<sub>12</sub>TiO<sub>20</sub> to be narrowed. Similarly, for some other Bi(III)-containing oxides, such as  $CaBi_2O_4$ ,  $^{696,697}$   $Bi_2WO_6$ ,  $^{145,698,699}$   $Bi_2MoO_6$ ,  $^{700}$   $Bi_2O_4$ ,  $^{701}$  and  $Bi_2MNbO_7$  (M = Al, Ga, In),  $^{702,703}$  the proposed band structure of the valence bands also involved the hybridized orbitals between the Bi 6s and the O 2p orbitals. These played an important role in their visible-light absorption and photocatalytic activities. Kako et al. investigated the photophysical properties and band structure of NaBiO<sub>3</sub> as a visible-light-driven oxide Bi (V) photocatalyst. 704 They found that the band structure of the valence band in NaBiO<sub>3</sub>, mainly composed of the O 2p orbitals, was quite different from that of Bi(III)-containing oxides. The latter was composed of hybridized orbitals between the Bi 6s and the O 2p orbitals at the top of the valence band. This situation arose because the contribution of the empty Bi<sup>5+</sup> 6s orbitals in NaBiO3 to the valence band was much smaller than the filled Bi<sup>3+</sup> 6s orbitals in the Bi (III)-containing oxides.





**Figure 21.** (A) Calculated total and partial density of states (DOS) of PbBi<sub>2</sub>Nb<sub>2</sub>O<sub>9</sub>. The conduction and valence bands of the compound consist of empty Nb 4d and occupied O 2p orbital, with the latter hybridized with Pb6s and/or Bi6s, giving the smaller band gap compared to compounds that do not contain Pb and Bi in their structure. (B) Schematic band energy diagram of lead- or bismuth-substituted perovskite-related oxides. Reprinted with permission from ref 707. Copyright 2006 Elsevier.

Li and co-workers developed the two binary lead niobates, Pb<sub>3</sub>Nb<sub>2</sub>O<sub>8</sub> and Pb<sub>3</sub>Nb<sub>4</sub>O<sub>13</sub>, as visible-light-sensitive photocatalysts. 705,706 Compared to pure Nb<sub>2</sub>O<sub>5</sub>, the hybridization of the Pb 6s and O 2p orbitals in these lead niabates raised the position of the valence band edge and narrowed the band gap, leading to the observed visible-light responses. Kim and co-workers investigated the substitution effects of lead on the photophysical and photocatalytic properties of a range of UV-light-active photocatalysts: CaBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub>, CaBi<sub>2</sub>Nb<sub>2</sub>O<sub>9</sub>, K<sub>0.5</sub>La<sub>0.5</sub>Ca<sub>1.5</sub>Nb<sub>3</sub>O<sub>10</sub>, and Sr<sub>3</sub>Ti<sub>2</sub>O<sub>7</sub>. They found that their lead-containing analogues (PbBi<sub>4</sub>Ti<sub>4</sub>O<sub>15</sub>, PbBi<sub>2</sub>Nb<sub>2</sub>O<sub>9</sub>,  $K_{0.5}La_{0.5}Ca_{0.75}Pb_{0.75}Nb_3O_{10},\ K_{0.5}La_{0.25}Bi_{0.25}Ca_{0.75}Pb_{0.75}Nb_3O_{10},$ and PbTiO<sub>3</sub>) absorbed visible light and exhibited good photocatalytic activities for water decomposition under visible-light irradiation. This was attributed to the additional hybridization of the occupied Pb 6s and O 2p orbitals in such lead-containing compounds. This pushed up the position of the valence band, giving a smaller band gap compared to their respective lead-free counterparts as shown in Figure

Hosogi and co-workers found that SnNb<sub>2</sub>O<sub>6</sub><sup>243,709</sup> and SnNbO<sub>2</sub>,<sup>710</sup> which contain Sn<sup>2+</sup> with a 5s<sup>2</sup> electron configuration, showed the photocatalytic activity under visible-light irradiation for H<sub>2</sub> or O<sub>2</sub> evolution from an aqueous solution containing methanol or AgNO<sub>3</sub>. The valence band level formed from a hybrid orbital of Sn 5s and O 2p was higher than that consisting of only O 2p orbitals. This made SnNb<sub>2</sub>O<sub>6</sub> responsive to visible light. Furthermore, they developed a novel series of Sn<sup>2+</sup>-exchanged layered metal oxides, Sn<sup>2+</sup>/K<sub>2</sub>Ti<sub>2</sub>O<sub>5</sub>, Sn<sup>2+</sup>/K<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub>, Sn<sup>2+</sup>/Cs<sub>2</sub>Ti<sub>6</sub>O<sub>13</sub>. These Sn<sup>2+</sup>-exchanged layered metal oxides had visible-light absorption bands and showed activities for H<sub>2</sub> or O<sub>2</sub> evolution from an aqueous solution containing a sacrificial reagent under visible-light irradiation.<sup>711</sup> This was due to the

electronic transition from an electron donor level consisting of Sn 5s orbitals to conduction bands consisting of Ti 3d and Nb 4d orbitals. Of particular note is the contribution of the Sn 5s filled orbitals to the valence band structure of  $\beta$ -SnWO<sub>4</sub>, which causes a decrease in the band gap, leading to a novel visible-light active photocatalyst for H<sub>2</sub> evolution from an aqueous methanol solution under visible-light irradiation.<sup>712</sup>

#### 4.4.3. f-block Metal Oxides

The introduction of rare earth 4f orbitals with a view to tuning the electronic structure has also been investigated for developing new visible-light-driven photocatalysts. Zou et al. investigated the effects of the rare earth elements on the band structures and photophysical properties of Bi<sub>2</sub>RNbO<sub>7</sub> (R = Y, rare earth). They found that the R 4f orbitals of the R<sup>3+</sup> ions were very important for the band structures of these photocatalysts. The visible-light absorption was found to be based on the R 4f and Nb 4d band transition between the R<sup>3+</sup> and Nb<sup>5+</sup> ions. That is to say, the partly filled R 4f level in the rare earths could form a new energy level in the band structure, and the 4f-d transition resulted in the narrowed band gaps of these photocatalysts. Hwang et al. also found that the 4f band of rare earths greatly contributed to the band structures of  $R_2Ti_2O_7$  (R = rare earth: La, Pr, Nd). 127 The R 4f level in R<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub> was shifted to a lower energy as the number of 4f electrons increased. This decreased the band gap energy of both Pr<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub> and Nd<sub>2</sub>Ti<sub>2</sub>O<sub>7</sub>. A similar phenomenon was also discovered by Uno and co-workers for the  $R_2M_2O_7$  (R = La, Pr, Nd, Sm, Gd, Dy, Ho, Er, and Yb; M = Ti, Zr) photocatalysts. <sup>132,154</sup> Because of the effect of the 4f orbital electrons, Sm<sub>2</sub>M<sub>2</sub>O<sub>7</sub> (M = Ti, Zr) and  $Nd_2Zr_2O_7$  were considered possible

candidates capable of responding to visible light for photocatalytic hydrogen production from water.

Tian et al. developed a series of Ce-containing metal oxide photocatalysts  $K_4Ce_2M_{10}O_{30}$  (M = Ta, Nb), capable of evolving H<sub>2</sub> and O<sub>2</sub> from aqueous solutions, under visiblelight irradiation ( $\lambda > 420$  nm), containing a sacrificial electron donor and acceptor, respectively.714 It was demonstrated that the conduction bands of  $K_4Ce_2M_{10}O_{30}$  (M = Ta, Nb) were mainly attributed to the Ta 5d (or Nb 4d) orbitals, while the valence bands were composed of hybridized O 2p + Ta 5d (or Nb 4d) and occupied Ce 4f orbitals. The latter made an essential contribution to the small band gap energy of these photocatalysts. The pyrochlore-type compound Sm<sub>2</sub>InMO<sub>7</sub> (M = Ta, Nb), with a  $4f-d^{10}-d^0$  configuration, was found to be a new stable visible-light-responsive photocatalyst for  $H_2$  evolution from water. The  $4f-d^{10}-d^0$  configuration was also proven useful for visible-light photocatalysis from the viewpoint of both the crystal and electronic structure. 715,716

#### 4.4.4. Miscellaneous Photocatalysts

Recently, some phosphate semiconductors were reported to show good photocatalytic activity for water splitting under visible-light irradiation. For example, Ag<sub>3</sub>PO<sub>4</sub> showed an extremely high quantum yield of ca. 80% at wavelengths less than  $\sim$ 480 nm for O<sub>2</sub> evolution from an aqueous silver nitrate solution.<sup>717</sup> It was significantly higher than previously reported values.667 Li<sub>9</sub>Fe<sub>3</sub>(P<sub>2</sub>O<sub>7</sub>)<sub>3</sub>(PO<sub>4</sub>)<sub>2</sub>, prepared by a hydrothermal method, was proven to be active for photocatalytic H<sub>2</sub> evolution under visible-light irradiation, where I<sup>-</sup> was used as the electron donor.718

TiSi<sub>2</sub>, with a band gap range from 3.4 to 1.5 eV, was reported as a promising semiconducting catalyst for visiblelight-driven overall water splitting. The evolution of oxygen and hydrogen was facilitated with the formation of the catalytic centers on the surface of TiSi<sub>2</sub> by photooxidation. Moreover, the reversible photoadsorption of oxygen onto the TiSi<sub>2</sub> catalyst allowed the convenient separation of the hydrogen and oxygen products.<sup>719</sup>

A metal-free polymeric photocatalyst, carbon nitride (g- $C_3N_4$ ), was developed by thermal condensation for efficient hydrogen or oxygen production from water under visiblelight irradiation with the assistance of a sacrificial reagent. By varying the thermal condensation conditions, the fine adjustment of the electronic and optical properties was possible. The band gap of the fully condensed graphitic carbon nitride was estimated to be 2.7 eV. Whereas the valence band was a combination of the HOMO levels of the melem monomer derived from the nitrogen p<sub>z</sub> orbitals, the conduction band was similarly connected to the LUMO of the melem monomer predominantly from the carbon p<sub>7</sub> orbitals.<sup>720</sup> Homogeneous substitution of sulfur for lattice nitrogen within g-C<sub>3</sub>N<sub>4</sub> gave rise to an increased valence bandwidth in combination with an elevated conduction band minimum and a slightly reduced absorbance. The resulting  $C_3N_{4-x}S_x$  showed a  $H_2$  evolution rate 8.0 times higher than  $C_3N_4$  under the irradiation of  $\lambda > 420$  nm.<sup>721</sup> The optical absorption of g-C<sub>3</sub>N<sub>4</sub> was extended into the visible region up to about 750 nm after being copolymerized with barbituric acid. 722 The modified g-C<sub>3</sub>N<sub>4</sub> showed a remarkable improvement in H<sub>2</sub> evolution activity compared to g-C<sub>3</sub>N<sub>4</sub> under both UV and visible-light irradiation.<sup>722</sup>

A graphite oxide semiconductor photocatalyst was reported to steadily produce H<sub>2</sub> under visible-light irradiation from an aqueous methanol solution, even in the absence of Pt as the cocatalyst. 723 The band gap energy of graphite oxide was dependant on the number of oxygenated sites and was estimated from optical absorption spectra to be 2.4–4.3 eV. The valence band edge of graphite oxide was mainly composed of the O 2p orbitals. The conduction band edge was mainly formed by the antibonding  $\pi^*$ -orbitals and had a higher energy level than that needed for H<sub>2</sub> generation.

## 5. Approaches for Efficient Photogenerated Charge Separation

While visible-light-driven photocatalysts with proper band structures are currently developed using some modification technology or band engineering approaches as discussed above, the issue of photogenerated charge separation is another key factor strongly affecting the efficiency of the photocatalytic water-splitting process. Clearly, in order to increase the utilization rate of the photogenerated charges and obtain high photocatalytic water-splitting activities, the photogenerated charges must be efficiently separated to avoid bulk/surface charge recombination and transfer to the separated active sites on the surface of the photocatalysts. This is to ensure depression of the backward reaction of hydrogen and oxygen to form water. In this section, we will review the primary approaches and achievements to date for the efficient separation of photogenerated charges in the water-splitting process. We aim to present the numerous experimental and theoretical research results that have appeared over the past decades.

## 5.1. Cocatalyst Loading

Transtition metals, especially the noble metals, are widely used as effective cocatalysts for photocatalytic water splitting. Taking Pt as an example, the processes of charge transfer between cocatalyst and host photocatalyst are described in Figure 22. When the noble metal is loaded onto the surface of the photocatalyst, the photogenerated electrons migrate to the surface of the host photocatalyst and are entrapped by the noble metal cocatalyst, because the Fermi energy level of noble metal is always lower than that of the semiconductor photocatalyst. Meanwhile, the photogenerated holes stay at the host photocatalyst and migrate to its surface. This results in the efficient separation of the photogenerated electrons and holes. Subsequently, the separately localized electrons and holes become involved in their roles as the reducer and oxidizer, respectively, in the photocatalytic reaction. Overall, the role played by the cocatalysts dispersed on the surface of the photocatalysts is extremely important. It improves the

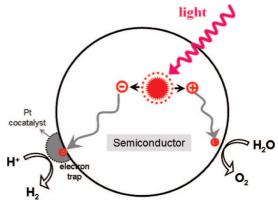


Figure 22. Processes of charge transfer between host photocatalyst and cocatalyst, taking Pt as the example of cocatalyst.

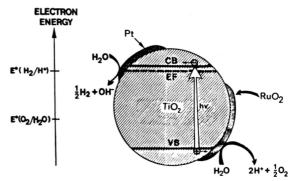
overall photocatalytic activity of the water splitting because it helps to promote charge separation, which in return reduces both bulk and surface electron/hole recombination. It also accelerates the surface chemical reaction by inhibiting the backward reaction.

#### 5.1.1. Noble Metal Cocatalysts

As one of the noble metals, Pt has been widely used as the cocatalyst in photocatalytic water splitting over many different kinds of semiconductors: oxides, 71,80,92,186,724–726 (oxy)sulfides, 489,490,727–730 and (oxy)nitrides. 390,430,468,476,731 All have been shown to greatly enhance the photocatalytic activity for hydrogen evolution. Up until now, the highest photocatalytic activities for hydrogen production from water using visible-light irradiation are from photocatalysts loaded with Pt as the cocatalyst. 732,733 Some other noble metals, such as Au, 734–739 Ru, 540,541,740–742 Pd, 735,743–745 Ag, 746–750 and Rh, 751–754 have also been reported as efficient cocatalysts.

Iwase et al. found that fine gold nanoparticles played an important role in the creation of active sites for H<sub>2</sub> evolution and the enhancement of charge separation. 167 In addition, the back-reaction between H2 and O2 to produce water on the Au cocatalyst was negligible in comparison to that on a Pt cocatalyst. This ultimately led to improved photocatalytic activities of some titanate, niobate, and tantalate photocatalysts for overall water splitting. Hara et al. reported the unusual enhancement of H2 evolution by Ru on a TaON photocatalyst under visible-light irradiation.<sup>740</sup> It should be noted that, under identical conditions, other noble metals, such as Pt, Ir, and Rh, were ineffective in promoting any evolution of H<sub>2</sub>. The authors deduced that the electronic structure of the interface between the Ru particles and TaON possibly facilitated electron transfer from TaON to Ru. This ultimately gave an impetus to charge separation.

Wu et al. investigated H<sub>2</sub> production with low CO selectivity from the photocatalytic reforming of glucose in water on metal/ $TiO_2$  catalysts (metal = Pt, Rh, Ru, Ir, Au, Ni, and Cu).<sup>755</sup> The loaded metals, in particular Rh, were found to greatly enhance the rate of H<sub>2</sub> production. This was attributed to the fact that the Schottky barrier formed at the metal and TiO2 interface could serve as an efficient electron trap, thus preventing photogenerated electron-hole recombination. The effect of the nature of the metal cocatalyst was interpreted in terms of different electronic interactions between the metal nanoparticles and the TiO<sub>2</sub> surface.<sup>756</sup> It was also reported that the smaller the Schottky barrier height at the metal/semiconductor junction, the greater was the electron flow from semiconductor to metal, thus leading to higher photocatalytic activity for hydrogen production.<sup>757</sup> On the other hand, the synergistic enhancement effect of the Pt and Ru coloading on hydrogen evolution compared to that of either Pt or Ru alone was observed by Liu and co-workers. 393,394 They considered that the superior activity of Pt/Ru-loaded LaTaON<sub>2</sub> (or Y<sub>2</sub>Ta<sub>2</sub>O<sub>5</sub>N<sub>2</sub>) for H<sub>2</sub> evolution was related to enhanced electron transfer from the conduction band of the photocatalyst to the Pt/Ru cocatalysts. This retards the possibility of electron-hole recombination in the valence band and improves the charge separation efficiency. When cocatalysts consisting of Au/Pd bimetallic nanoparticles with core/shell structures were loaded onto the TiO<sub>2</sub> surface, selective H2 permeability by the Pd shell was believed to contribute to the selective donation of the photogenerated electrons to the protons. The result is greatly



**Figure 23.** Schematic illustration of the photoinduced events leading to water decomposition on Pt/RuO<sub>2</sub>—TiO<sub>2</sub> composite catalyst. Reprinted with permission from ref 70. Copyright 1981 American Chemical Society.

enhanced photocatalytic production of  $H_2$  from aqueous ethanol solutions.  $^{758}$ 

#### 5.1.2. Transition-Metal Oxide Cocatalysts

Some transition-metal oxides such as  $RuO_2^{187,244,257,277,577,759,760}$  and  $NiO^{109,110,175,206,761,762}$  are also well-known as efficient cocatalysts in water splitting. Domen and co-workers found that  $\beta$ -Ge<sub>3</sub>N<sub>4</sub> alone exhibited little photocatalytic activity for water decomposition. <sup>24,277</sup> However, when loaded with RuO<sub>2</sub>, the material became photocatalytically active under UV irradiation ( $\lambda$  > 200 nm). Maeda et al. found that the presence of RuO<sub>2</sub> cocatalysts loaded on the g-C<sub>3</sub>N<sub>4</sub> surface was indispensable not only for enhancing O<sub>2</sub> evolution activity but also for suppressing the oxidative decomposition of the nitride catalyst. <sup>763</sup> Similarly, IrO<sub>2</sub> was also found to act as a water oxidation cocatalyst for water splitting. <sup>622,764,765</sup>

Borgarello and co-workers noted a pronounced synergistic effect in catalytic activity when both RuO<sub>2</sub> and Pt are codeposited onto the TiO<sub>2</sub> particles. <sup>308,606,766</sup> Figure 23 depicts the photosplitting of water on Pt/RuO2-TiO2. It has been proposed that Pt is likely to yield an ohmic contact whereas a Schottky barrier may be formed by RuO2. This would direct the electron flow to the Pt sites while the holes would be trapped by RuO<sub>2</sub>,<sup>70</sup> resulting in efficient charge separation and improved photocatalytic activity. Moreover, Teramura et al. found that the high dispersion of RuO<sub>2</sub> nanoclusters on the catalyst surface was essential for improving the photoactivity for H<sub>2</sub> and O<sub>2</sub> evolution in overall water splitting.<sup>767</sup> It was demonstrated that both Ru(IV) species and bulk RuO2 exhibited less activity for overall water splitting, whereas with RuO<sub>2</sub> nanoclusters as cocatalyst,  $(Ga_{1-x}Zn_x)$ - $(N_{1-x}O_x)$  displayed an improvement of  $H_2$  and  $O_2$  evolution.

In a series of studies using  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$ , Maeda and co-workers loaded Cr-M (M being one of the transition metals, Fe, Co, Ni, Cu, Ru, Rh, Pd, Ag, Ir, or Pt) as a mixed-oxide cocatalyst onto  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  using a coimpregnation method. The largest improvement in activity was obtained by loading the Rh-Cr mixed oxide (1 wt % Rh and 1.5 wt % Cr) with calcination at 623 K. It was proposed that loading the Cr-Rh mixed oxide facilitated the charge transfer from the host photocatalyst to the cocatalyst. It is also possible that, in addition, the loading promoted the creation of active sites for hydrogen evolution, which resulted in inhibited charge recombination and enhanced photocatalytic activity. When and co-workers also developed noble metal/ $Cr_2O_3$  core/shell nanoparticles as a cocatalyst for overall photocatalytic water splitting. The property of the transition of

**Figure 24.** Schematic reaction mechanism of overall water splitting on Rh/Cr<sub>2</sub>O<sub>3</sub>-core/shell-loaded  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  and the corresponding processes on supported Rh nanoparticles and Cr<sub>2</sub>O<sub>3</sub> nanoparticles. The impregnation of GaN:ZnO with Cr<sub>2</sub>O<sub>3</sub> nanoparticles alone does not promote overall water splitting. H<sub>2</sub> evolution occurs on Rh nanoparticles loaded on GaN:ZnO; however, in the presence of O<sub>2</sub>, this H<sub>2</sub> reacts immediately with O<sub>2</sub> on the Rh nanoparticles to form H<sub>2</sub>O. For Rh/Cr<sub>2</sub>O<sub>3</sub>-core/shell-loaded  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$ , H<sub>2</sub> evolution occurs on the Cr<sub>2</sub>O<sub>3</sub> shell, and O<sub>2</sub> evolution occurs on the surface of GaN:ZnO bulk, thereby preventing water formation from H<sub>2</sub> and O<sub>2</sub> on Rh nanoparticles. Reprinted with permission from ref 773. Copyright 2006 Wiley-VCH.

a schematic reaction mechanism for overall water splitting on Rh/Cr<sub>2</sub>O<sub>3</sub>-core/shell-loaded ( $Ga_{1-x}Zn_x$ )( $N_{1-x}O_x$ ). The mode of operation is quite different to cocatalysts such as the Rh-Cr mixed oxide, in which at least two roles are played simultaneously. In the Rh/Cr<sub>2</sub>O<sub>3</sub> core/shell cocatalyst, the Rh core and the Cr<sub>2</sub>O<sub>3</sub> shell worked independently. The Rh core promotes charge separation in  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  as well as tunneling electron transfer to the Cr<sub>2</sub>O<sub>3</sub> shell, whereas the Cr<sub>2</sub>O<sub>3</sub> shell provides catalytic active sites for H<sub>2</sub> production, thereby preventing water formation from H<sub>2</sub> and O<sub>2</sub> on the Rh. The Rh core could be further replaced by other noble metals (Pd, Pt) as well as metal oxides (NiO<sub>x</sub>, RuO<sub>2</sub>, Rh<sub>2</sub>O<sub>3</sub>). This also resulted in enhanced photocatalytic activity for overall water splitting under visible-light irradiation.<sup>776</sup> This enhancement in activity was primarily due to the suppression of undesirable reverse reactions (H<sub>2</sub>-O<sub>2</sub> recombination and/or O<sub>2</sub> photoreduction) and possibly the protection of the core component from chemical corrosion. This depended on the core type. Among the core materials examined, Rh species exhibited relatively high performance. Interestingly, with assistance of Mn<sub>3</sub>O<sub>4</sub> nanoparticles coloading, the photocatalytic activity for overall water splitting was improved for  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$ , which was modified with core/shell-structured Rh/Cr<sub>2</sub>O<sub>3</sub> nanoparticles. The Mn<sub>3</sub>O<sub>4</sub> nanoparticles functioned as O<sub>2</sub> evolution sites, and Rh/Cr<sub>2</sub>O<sub>3</sub> nanoparticles hosted H<sub>2</sub> evolution sites.<sup>777</sup>

When NiO acts as the cocatalyst for photocatalytic water splitting, an activation pretreatment is generally necessary for NiO-loaded photocatalysts in order to obtain high activities. A reduction—reoxidation treatment produces a double-layered NiO/Ni structure on the surface of the photocatalyst. Because NiO has a lower Fermi level in comparison to metallic Ni, this double-layered structure facilitates the electron transfer from the photocatalyst material to the Ni layer and then onto the NiO layer of the H<sub>2</sub>

evolution sites. The result is the efficient charge separation for water splitting and the suppression of the backward reaction of water splitting on the metallic Ni surface.

Tian et al. investigated the effects of nanosized Pt, RuO<sub>2</sub>, and NiO<sub>x</sub> loading onto the visible-light-driven photocatalysts  $K_4Ce_2M_{10}O_{30}$  (M = Ta, Nb) for hydrogen evolution from water.<sup>779</sup> The photocatalytic activities of evolving H<sub>2</sub> were markedly promoted by cocatalysts loading, in particular, the NiO<sub>x</sub> loading with the formation of the NiO/Ni doublelayered structure. This was attributable to enhanced electron migration from the conduction band of the photocatalyst to the NiO/Ni cocatalyst. Hwang et al. prepared a series of metal oxide ( $MO_x$ , M = Ni, Pt, Cs, Bi, Fe, Pb, Ce)-loaded  $Sr_2Nb_2O_7$ photocatalysts using the impregnation method and following reduction—oxidation treatment. 780 Of these metal oxides, the  $NiO_x$  cocatalyst showed the highest activity for water splitting. The reduction—oxidation pretreatment for the formation of a double-layered structure was important to achieve the high activity especially for NiO<sub>x</sub>-loaded catalysts. This could be ascribed to the more efficient electron—hole separation as a result of the role of nickel played in the p-type/n-type junction between NiO<sub>x</sub> and Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub> as formed in the reduction-oxidation pretreatment. However, this marked dependence of photocatalytic activity on pretreatment conditions was not observed for any other metal oxides in this study. To improve the photocatalytic activity of H<sub>2</sub>O decomposition, various kinds of metal oxides ( $MO_x$ : M =Cr, Mn, Fe, Co, Cu, Ru, Ag, Ce, Sm, and Pb) were introduced into the  $NiO_y-K_2La_2Ti_3O_{10}$  catalyst as a second component of the cocatalyst together with NiO<sub>v</sub>. Only the CrO<sub>x</sub>-NiO<sub>y</sub>-K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub> catalyst showed higher photocatalytic activity and durability for longer periods of irradiation than the parent NiO<sub>v</sub>-K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub>.<sup>781</sup>

#### 5.1.3. Nonmetal-Oxide Cocatalysts

In addition to the noble metals and the transition-metal oxides, some other compounds, such as the transition-metal sulfides, have also been developed as novel cocatalysts. Zong and co-workers developed MoS2 as a novel cocatalyst for photocatalytic hydrogen evolution. 782,783 They found that the CdS activity was enormously increased by loading with MoS<sub>2</sub>, even higher than that of Pt-loaded CdS under the same reaction conditions. The better coupling between the structures and electronic configurations of MoS<sub>2</sub> and CdS together, and the formation of junctions between CdS and MoS<sub>2</sub>, improved the charge separation and were mainly responsible for the high activity of this MoS<sub>2</sub>/CdS catalyst. MoS<sub>2</sub> nanoparticles cocatalysts were also found to be effective in H<sub>2</sub> evolution over CdSe nanoribbons. <sup>784</sup> The MoS<sub>2</sub> activated the photocatalyst by reducing the electrochemical proton reduction overpotential.<sup>784</sup> Li and co-workers found that the photocatalytic activity of CdS could be enhanced significantly by loading a small amount of a noble metal sulfide (PdS, Rh<sub>2</sub>S<sub>3</sub>, Ru<sub>2</sub>S<sub>3</sub>), as well as a noble metal (Pt, Pd, Ru, Rh). <sup>733,785</sup> Moreover, codeposition of Pt with other noble metal sulfides on CdS demonstrated further enhanced photoactivity in hydrogen production; the photocatalytic activity of CdS codeposited with Pt and PdS was greater than those of Pt/ CdS and PdS/CdS. This synergistic effect between the Pt and PdS cocatalysts could be explained by the fact that the two cocatalysts may facilitate separation of the photogenerated electrons and holes on the photocatalyst.

Tabata et al. found that dispersion of transition metal sulfides such as NiS, FeS, Ru<sub>2</sub>S<sub>3</sub>, Ag<sub>2</sub>S, CoS, and PdS, into

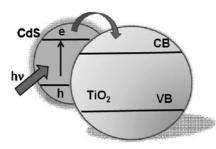
the CuGa<sub>3</sub>S<sub>5</sub> photocatalyst solution also increased the photocatalytic activity. <sup>498</sup> The transition metal sulfides were believed to accept electrons from the excited state of CuGa<sub>3</sub>S<sub>5</sub> and reduce H<sup>+</sup> into H<sub>2</sub> when transition metal sulfides and CuGa<sub>3</sub>S<sub>5</sub> particles collided with each other. Jang et al. proved that tungsten carbide (WC) had potential as an alternative cocatalyst for photocatalytic hydrogen production. <sup>786</sup> Under visible-light irradiation, the WC/CdS photocatalyst showed excellent photocatalytic activity for hydrogen production from water, comparable to that of conventional Pt/CdS. This was because WC provided active sites for hydrogen evolution and promoted fast migration of the photogenerated electrons from CdS to WC. This resulted in efficient charge separation and enhanced photocatalytic activity.

#### 5.2. Semiconductor Combinations

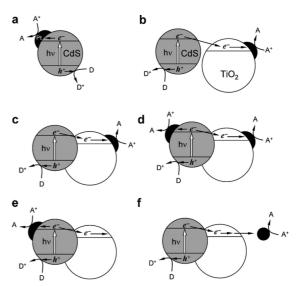
The semiconductor combination approach has been shown to be another effective method for improving photocatalytic activity through better photogenerated charge separation with a formation of a heterojunction structure.

As is well-known, CdS, which is a fascinating visiblelight-driven photocatalyst for hydrogen production, is very unstable toward photocorrosion, as a result of serious selfoxidation by the photogenerated holes in the valence band. To improve its photocatalytic activity and stability, CdS has been embedded in different kinds of inert matrices, 787-793 as well as combined with other semiconductors of different band energies, such as TiO<sub>2</sub>, <sup>794,614</sup> ZnO, <sup>614,796–798</sup> LaMnO<sub>3</sub>, <sup>799</sup> and KNbO<sub>4</sub>.800,801 Among these composites, CdS-based semiconductor combination systems, CdS/TiO2 has attracted the most extensive research. 802-810 Charge separation in a CdS/ TiO<sub>2</sub> system, as shown in Figure 25, 811,812 has been investigated in several detailed studies. 614,811–814 Both the conduction and valence band edges of CdS are at more negative potentials than those of TiO<sub>2</sub>. Under visible-light irradiation, the photogenerated electrons in the CdS particles quickly transfer to TiO<sub>2</sub> particles, whereas photogenerated holes stay in CdS. This facilitates the electron-hole separation and prevents the charge recombination, improving the photocatalytic activity.

Tada et al. developed an anisotropic CdS-Au-TiO<sub>2</sub> heterojunction, in which CdS, TiO<sub>2</sub>, and the electron-transfer medium (Au) were all spatially fixed.<sup>815</sup> This three-component system exhibited a high photocatalytic activity, far exceeding those of the single- and two-component systems. This is a result of vectorial electron transfer driven by the two-step excitation of TiO<sub>2</sub> and CdS. Park et al. investigated the effects on visible-light-driven hydrogen production of varied combinations of CdS, TiO<sub>2</sub>, and Pt in hybrid photocatalysts (Figure 26).<sup>816</sup> They found that direct particle-to-particle contact of CdS and TiO<sub>2</sub>, and photodepo-



**Figure 25.** Charge separation in CdS/TiO<sub>2</sub> semiconductor combination system under visible-light irradiation. Reprinted with permission from ref 812. Copyright 2009 Wiley-VCH.

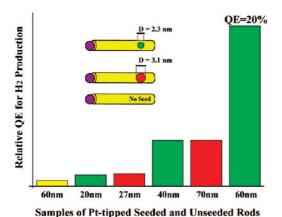


**Figure 26.** Illustrative diagrams of the electron transfers in the hybrid photocatalysts. (a) Pt-CdS; (b) CdS + Pt-TiO<sub>2</sub> (physically mixed); (c) CdS/(Pt-TiO<sub>2</sub>); (d) Pt-(CdS/TiO<sub>2</sub>); (e) TiO<sub>2</sub>/(Pt-CdS); (f) CdS/TiO<sub>2</sub> + Pt. Reprinted with permission from ref 816. Copyright 2008 The Royal Society of Chemistry.

sition of Pt on the  $TiO_2$  particle surface, which resulted in the vectorial electron transfer of  $CdS \rightarrow TiO_2 \rightarrow Pt$  (Figure 26c), was necessary to achieve efficient charge separation and transfer and hence observed the highest photoactivity of the  $CdS/(Pt-TiO_2)$  hybrid catalyst.

Because of the fact that the semiconductor combination has been proven to play a beneficial role in the improvement of photogenerated charge separation, many other different kinds of semiconductor combination systems have been developed for photocatalytic hydrogen production from water. Wang et al. developed a novel Zn-doped Lu<sub>2</sub>O<sub>3</sub>/Ga<sub>2</sub>O<sub>3</sub> composite photocatalyst for stoichiometric water splitting.<sup>255</sup> It was proved to have great advantage over the individual precursor materials owing to its enhanced charge-separation ability. Similarly, the TiO<sub>2</sub>/SnO<sub>2</sub> mixed oxide showed enhanced photocatalytic activity for hydrogen generation compared to pure TiO<sub>2</sub>.83 Guo and co-workers found that the TiO<sub>2-x</sub>N<sub>x</sub>-WO<sub>3</sub> composite photocatalyst showed higher hydrogen production activity under visible-light irradiation than either TiO<sub>2-x</sub>N<sub>x</sub> or WO<sub>3</sub> alone. 817,818 Again this was attributed to the efficient charge separation of photogenerated electrons and holes.

Zou and co-workers reported a novel Cr-doped Ba<sub>2</sub>In<sub>2</sub>O<sub>5</sub>/ In<sub>2</sub>O<sub>3</sub> system, which was properly bridged by an ohmic contact. 819 It turned out to be a novel composite photocatalyst with enhanced activity for water splitting when compared to the individual components. The photoinduced charge carrier separation and migration were promoted by the potential offsets between the band edges of individual materials Cr-Ba<sub>2</sub>In<sub>2</sub>O<sub>5</sub> and Cr-In<sub>2</sub>O<sub>3</sub>. On the basis of a similar electron transfer mechanism, In<sub>2</sub>O<sub>3</sub>/NaNbO<sub>3</sub> was also found to be advantageous for both photocatalytic H<sub>2</sub> evolution under visible-light irradiation and pure water splitting under UV irradiation.820 Ou et al. successfully synthesized a visible-light-driven MWNT-TiO<sub>2</sub> composite catalyst using MWNTs dispersively grown on the surface of TiO2.821 In this system, efficient charge separation and effective photocatalytic H<sub>2</sub> evolution resulted when the photogenerated electrons of the MWNTs transferred quickly to the conduction band of the TiO<sub>2</sub> under visible-light irradiation. Similarly, in a Bi<sub>2</sub>S<sub>3</sub>/TiO<sub>2</sub> system, the enhanced photoactivity was



**Figure 27.** Relative quantum efficiency for hydrogen production, obtained from platinum-tipped unseeded CdS rods (yellow), and five different samples of platinum-tipped seeded rods, with seed diameters of 3.1 (red) or 2.3 nm (green). Underneath each bar is the corresponding average sample length. Reprinted with permission from ref 824. Copyright 2010 American Chemical Society.

ascribed to photoelectron transfer from the conduction band of Bi<sub>2</sub>S<sub>3</sub> to the conduction band of TiO<sub>2</sub>, resulting in water reduction.822

Silva et al. developed a novel Pt-interlinked hybrid composite of cubic-phase CdS (c-CdS) and hexagonal-phase CdS (hex-CdS).823 In order to explain the relative order of photocatalytic activity found, c-CdS/Pt/hex-CdS > Pt/c-CdS/ hex-CdS > Pt/hex-CdS > hex-CdS > c-CdS/hex-CdS > c-CdS, they concluded that the formation of a potential gradient at the interface between the cubic and hexagonal CdS, and the interlinkage of quantum-sized cubic CdS and bulk-phase hexagonal CdS with Pt, were necessary to achieve the efficient charge separation and transfer. This was quite similar to the CdS/Pt/TiO2 semiconductor combination system investigated in their previous study.816

Recently, Amirav and Alivisatos designed a multicomponent nanoheterostructure for efficient photocatalytic hydrogen production composed of a Pt-tipped CdS rod with an embedded CdSe seed.<sup>824</sup> In such structures, holes were threedimensionally confined to the CdSe seed, whereas the delocalized electrons were transferred to the metal tip. Consequently, the electrons were separated from the holes over three different components and by the tunable physical length of the CdS rod. By tuning the nanorod heterostructure length and the seed size, as shown in Figure 27, the hydrogen production activity was significantly increased compared to that of the unseeded rods. This structure was found to be highly active for hydrogen production, with an apparent quantum yield of 20% at 450 nm. There was also a demonstrated improved stability compared to CdS rods without CdSe seeding.

The incorporation of a guest semiconductor into the interlayers of a lamellar compound for fabricating an intercalated composite semiconductor has been considered as a promising method for the enhancement of the photocatalytic activity. The idea is that in this guest-host semiconductor combination system, the recombination between the photoinduced charge carriers is effectively suppressed due to the quick charge transfer from the guest to the host semiconductor. 825–829 In the past few years, some transition metal oxides with semiconducting host layers have been shown to be generally promising candidates for use as intercalation hosts in photocatalysts.830-834 Wu and coworkers reported the intercalation of nanosized oxides (i.e.,

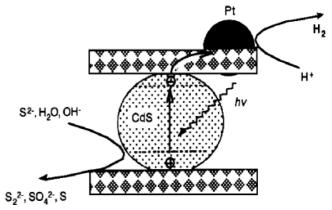


Figure 28. Mechanism of photocatalytic H<sub>2</sub> evolution on CdSintercalated layered composites loaded with Pt. Photogenerated electrons in CdS quickly transferred to host layers through the nanostructure, and the recombination between the photoinduced electrons and holes was effectively suppressed. Reprinted with permission from ref 842. Copyright 2007 IOP Publishing Ltd.

Fe<sub>2</sub>O<sub>3</sub> or TiO<sub>2</sub>) into the interlayer of various layered metal oxides (i.e., H<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub>, HNb(Ta)WO<sub>6</sub>, or HLaNb<sub>2</sub>O<sub>7</sub>). 835-838 When compared to the unintercalated materials, these intercalated composite semiconductors possessed higher photocatalytic activities for hydrogen production in the presence of methanol as a sacrificial agent under UV or even visible-light irradiation. This was ascribed to an effective separation of the photogenerated electrons and holes by electron transfer from guest semiconductor to the host layer. Subsequently, Jang et al. intercalated nanosized Fe<sub>2</sub>O<sub>3</sub> particles into the interlayers of HTiNbO<sub>5</sub> and HTiTaO<sub>5</sub> using successive intercalation reactions.<sup>839</sup> They found that the strong electronic interaction between the guest nanoparticles and the host layers, which was derived from the high electron deficiency of the nanosized Fe<sub>2</sub>O<sub>3</sub> particles, could work favorably for photocatalysis because of an efficient electron hole separation. However, in contrast to the results of Wu et al., 838 this Fe<sub>2</sub>O<sub>3</sub>-intercalated layered composite semiconductor did not lead to photocatalytic H<sub>2</sub> production from water reduction but instead to a high activity for oxygen production from water oxidation under visible-light irradiation.

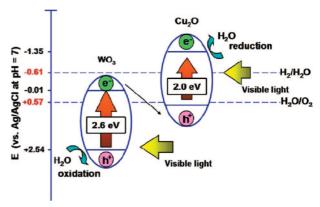
To improve their photocatalytic stability and photoactivity, sulfides, such as CdS, have also been intercalcated into the interlayers of some layered metal oxides. Shangguan et al. prepared a series of CdS-intercalated metal oxides using direct Cd<sup>2+</sup> exchange followed by a sulfurization process.<sup>840–842</sup> They found that the photocatalytic activities for hydrogen evolution of such CdS-intercalated composites were superior to those of CdS alone or to a physical mixture of CdS and the metal oxides. This activation effect was attributed to the quick transfer of the photogenerated electrons from CdS to the metal oxides through the layered nanostructure, as shown in Figure 28. The photogenerated charge recombination was effectively suppressed. Sato and co-workers investigated the photocatalytic activities of intercalated materials based on layered metal oxides (i.e., Ca<sub>2-x</sub>La<sub>x</sub>Nb<sub>3</sub>O<sub>10</sub>, HNbWO<sub>6</sub>, or  $HLaNb_2O_7)$  as the host and  $Cd_{0.8}Zn_{0.2}S$  as the guest.  $^{843-845}$ Fabrication involved successive intercalation and exchange reactions. Under visible-light irradiation, the hydrogen production activities of these Cd<sub>0.8</sub>Zn<sub>0.2</sub>S-intercalated composite photocatalysts were superior to those of unsupported Cd<sub>0.8</sub>Zn<sub>0.2</sub>S. They were further enhanced by the coincorporation of Pt. Clearly efficient charge separation resulted from the heterogeneous electron transfer from guest Cd<sub>0.8</sub>Zn<sub>0.2</sub>S

to host oxide layers, and this played an important role in the enhancement of the photocatalytic activity for hydrogen evolution.

It has also been noted that the encapsulation of guest semiconductors into the micro/mesostructured semiconducting host materials is effective for photogenerated charge separation and thus leads to the improved photocatalytic activity of such semiconductor combination systems. Using ion-exchange and sulfurization processes, Guan and coworkers successfully incorporated CdS nanoparticles into the microchannels of ETS-4 and ETS-10, which have the photocatalytically active -Ti-O-Ti- quantum wires in their frameworks. 846,847 The resulting CdS/ETS composite system showed better photocatalytic activity and stability than CdS nanoparticles for hydrogen production under visiblelight irradiation. This was interpreted in terms of more efficient charge separation due to the transfer of the photogenerated electrons from the CdS nanopaticles to the -Ti-O-Ti-O- nanowires. Similarly, Shen and Guo encapsulated CdS and In<sub>2</sub>S<sub>3</sub> nanoparticles into the mesopores of Ti-MCM-41.848,849 Under visible-light irradiation, the composite materials exhibited greater and more stable photocatalytic activity for hydrogen evolution than either bulk CdS or In<sub>2</sub>S<sub>3</sub>. Jing and Guo reported the deposition of quantum-sized WS2 onto the channel surface of crystalline mesoporous TiO<sub>2</sub>.850 It was postulated that under visible light only electrons transferred from nanosized WS<sub>2</sub> to the TiO<sub>2</sub> host. This resulted in the effective charge separation of electrons and holes photogenerated in the WS<sub>2</sub>, and thus the efficient photocatalytic activity for hydrogen production on the WS<sub>2</sub>/TiO<sub>2</sub> semiconductor combination system. Shortly afterward, they developed a novel composite CdS/meoporous zirconium titanium phosphate (ZTP) photocatalyst for working under visible light by a two-step thermal sulfidation procedure. In this system, the conduction band of ZTP could be continuously adjusted by choosing different Zr/Ti ratios. Therefore, by adjusting the conduction band of the photoactive ZTP to achieve an optimal optical match between the CdS guest and the ZTP host, highly efficient charge separation and hydrogen production were achieved over the designed CdS/ZTP composite photocatalyst. The highest quantum yield for this composite photocatalyst (with a Zr/ Ti ratio of 1:3 in the ZTP host) at 420 nm as measured in experiments reached 27.2%.851

Narrow band gap semiconductors, such as  $CdS^{852-854}$  and  $Cr_2O_3$ ,  $^{855}$  have also been incorporated into the nanotubular semiconducting host, generally titanate nanotubes ( $TiO_2NTs$ ), to develop high-efficiency composite photocatalysts for hydrogen production. Among them, the  $CdS/TiO_2NTs$  composite photocatalyst, in which homogeneously distributed CdS was incorporated inside the  $TiO_2NTs$ , showed a high photoactivity for hydrogen production from water containing sulfide and sulfite ions as hole scavengers under visible-light irradiation, with the apparent quantum yield of  $\sim$ 43.4% at  $\sim$ 420 nm. The potential gradient at the interface between the CdS nanoparticles and  $TiO_2NTs$  helped to facilitate the photoelectron diffusion from CdS particles toward the  $TiO_2NTs$  and led to high photocatalytic activity of hydrogen production.

It has been demonstrated that, in both photovoltaic and photoelectrochemical (PEC) cells, improved performance and higher conversion efficiency can be obtained by using diode structures. These were combined from suitable n-type photoanodes and p-type photocathodes<sup>856–858</sup> and led to

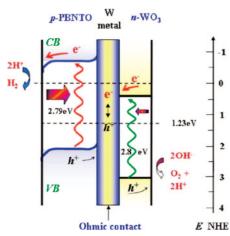


**Figure 29.** Reaction scheme of the photoinduced water splitting over the coupled p-type Cu<sub>2</sub>O and n-type WO<sub>3</sub> photocatalysts in a suspension system. Reprinted with permission from ref 867. Copyright 2008 Elsevier.

efficient electron—hole separation with the suppression of energy-wasteful charge recombination. 859–866 A p-Cu<sub>2</sub>O/n-WO<sub>3</sub> coupling system was developed by Hu et al. with a view to avoiding back-reactions of the photoinduced charges. 867 It resulted in higher photocatalytic hydrogen production. The reaction scheme of photoinduced water splitting over coupled p-type Cu<sub>2</sub>O and n-type WO<sub>3</sub> photocatalysts in a suspension system is shown in Figure 29.

Trari and co-workers synthesized a series of p-n heterojunction for use under visible-light irradiation, such as p-CuMO<sub>2</sub>/n-Cu<sub>2</sub>O (M = Mn, Cr),  $^{868,869}$  p-CuAlO<sub>2</sub>/n-TiO<sub>2</sub>,  $^{870}$ p-CuFeO<sub>2</sub>/n-SnO<sub>2</sub>,<sup>871</sup> and p-ZnFe<sub>2</sub>O<sub>4</sub>/n-SrTiO<sub>3</sub>,<sup>872</sup> for photocatalytic hydrogen production from aqueous solutions containing hole scavengers. In these p-n semiconductor systems, photogenerated electrons and holes separated efficiently and were involved in photocatalytic reduction and oxidation, respectively. Jang and co-workers successfully fabricated p/n-typed photocatalytic diodes, n-CdS/p-Ag-GaS<sub>2</sub>,<sup>873</sup> p-AgGaS<sub>2</sub>/n-TiO<sub>2</sub>,<sup>874</sup> and p-CaFe<sub>2</sub>O<sub>4</sub>/n-MgFe<sub>2</sub>O<sub>4</sub>.<sup>875</sup> All of these p/n photocatalysts exhibited higher visible-light activities for hydrogen production than the single p- or ncomponent. This was mainly due to efficient charge separation, which was caused by the rectification of photogenerated electrons and holes.

Kim et al. reported that photocatalytic nanodiodes formed from p-CaFe<sub>2</sub>O<sub>4</sub>/n-PbBi<sub>2</sub>Nb<sub>0.9</sub>W<sub>0.1</sub>O<sub>9</sub> yielded greatly enhanced and stable photocatalytic activity for water oxidation under visible light compared to the single-component photocatalyst. 876 In this p—n photocatalytic nanodiode, enhanced and stable photocatalytic activity was achieved because, under the influence of the internal electric field, the photogenerated holes moved to the p-CaFe<sub>2</sub>O<sub>4</sub> side and the electrons moved to the n-PbBi<sub>2</sub>Nb<sub>1.9</sub>W<sub>0.1</sub>O<sub>9</sub> side, leading to more efficient charge-carriers separation. In a later study, the authors adopted the concept of a p-type/n-type diode structure with an Ohmic junction and fabricated the psemiconductor/metal/n-semiconductor structured composite photocatalyst n-WO<sub>3</sub>/W/p-PbBi<sub>2</sub>Nb<sub>1.9</sub>Ti<sub>0.1</sub>O<sub>9</sub> for more efficient utilization of visible-light photons.877 This nanocomposite photocatalyst showed unprecedented high activity for the photocatalytic oxidation of water under visible-light irradiation, with quantum yield estimated to be ca. 41%. Figure 30 shows the working principle of WO<sub>3</sub>/W/ PbBi<sub>2</sub>Nb<sub>1.9</sub>Ti<sub>0.1</sub>O<sub>9</sub> nanocomposite photocatalyst. It had efficient electron-hole separation and higher net photon energies available for redox reactions through creation of the metal W layer needed to separate the p-type and n-type



**Figure 30.** Working principle of n-WO<sub>3</sub>/W/p-PbBi<sub>2</sub>Nb<sub>1.9</sub>Ti<sub>0.1</sub>O<sub>9</sub> nanocomposite photocatalyst. Reprinted with permission from ref 877. Copyright 2006 American Institute of Physics.

semiconductors to yield an Ohmic contact in between. This led to more efficient photocatalytic activity than the p-type/ n-type junction structure lacking this Ohmic layer.<sup>87</sup>

## 5.3. Modification of Crystal Structure and Morphology

In general, charge separation and transfer of photogenerated electrons and holes are strongly affected by the crystal structural features of the materials, namely, crystallinity, defects, and any crystal structure distortion. Additionally, surface properties such as particle size, surface area, surface structure, and active reaction sites, which are mainly related to the morphology of the materials, are also important. Thus, a respectable research effort has been dedicated to both the modification of crystal structure and morphology of the photocatalysts with the view to improve more efficient charge separation. This has had great success in the enhancement of photocatalytic activity for water splitting.

## 5.3.1. Modification of Crystal Structure

It has been well demonstrated that the crystal structure of TiO<sub>2</sub> plays a significant role in photocatalysis.<sup>795,878–882</sup> In photocatalysis, for pure-phase TiO<sub>2</sub>, the anatase phase was considered to be more active than the rutile phase. 795,878 In anatase TiO<sub>2</sub>, the photoexcited electrons could be trapped in oxygen vacancies of the anatase structure.<sup>72</sup> This made them easier to transfer to an electron trap, such as Pt particles on the surface, to participate in the photocatalytic reaction, preventing electron—hole recombination in photocatalysis. On the other hand, the poor photoactivity of rutile TiO<sub>2</sub> was due to the fact the electrons are trapped within intrinsic defects of the rutile structure where they were deexcited by near-infrared emission.<sup>72</sup> Meanwhile, enhanced activity was observed on mixed-phase TiO<sub>2</sub> photocatalysts relative to the corresponding pure phases.<sup>73,883–887</sup> Taking an anatase/rutile mixed-phase TiO<sub>2</sub> as an example, Figure 31 reveals a junction structure formed between the anatase and rutile phases. The migration of electrons across a phase junction (as shown in Figure 32) was beneficial for charge separation, thereby enhancing the photocatalytic activity for hydrogen evolution. 73,888 Furthermore, the high crystallinity of anatase TiO<sub>2</sub>, which had fewer lattice defects, was reported to have a positive effect on the photocatalytic activity for H<sub>2</sub> production. Good crystallinity led to significant enhancement

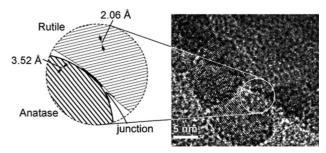
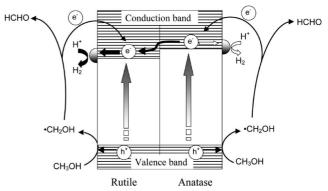


Figure 31. Junction structure formed between the anatase and rutile phase. Reprinted with permission from ref 73. Copyright 2008 Wiley-VCH.



**Figure 32.** Proposed pathway of electrons during photocatalytic  $H_2$  evolution over a mixture of anatase and rutile  $TiO_2$  nanoparticles. The amount of electron transfer proceeds predominantly in the order of solid thick arrow > thin solid arrow > unfilled arrow. Note: For simplicity, the formation of hydroxymethyl radicals (•CH<sub>2</sub>OH) by hydroxyl radicals (•OH) is represented by the hole oxidation step. Reprinted with permission from ref 888. Copyright 2010 American Chemical Society.

of the electron transport properties, whereas lattice defects behaved as recombination centers for the photoinduced electron/hole pairs with a concomitant decrease in the photocatalytic activity.889,890

Fu et al. studied the  $In_2S_3$  samples with different crystal structures.891 Their results showed that the indium vacancyordered tetragonal In<sub>2</sub>S<sub>3</sub> was inactive for hydrogen generation, whereas the vacancy-disordered cubic In<sub>2</sub>S<sub>3</sub> exhibited stable photoactivity under visible light.<sup>891</sup> Amano et al. reported that, owing to the suppression of the fast recombination of electron—hole pairs, the photocatalytic activity of visible-light-responsive Bi<sub>2</sub>WO<sub>6</sub> was greatly enhanced by crystallizaiton under a hydrothermal treatment, when compared to the negligible photocatalytic activity of amorphous Bi<sub>2</sub>WO<sub>6</sub>.892 Noda et al. demonstrated that the high photocatalytic activity of crystallized mesoporous Ta<sub>2</sub>O<sub>5</sub>, for overall water splitting, was attributable to the efficient transfer of the excited electrons and holes from inside the catalyst to the surface through the thin-walled crystalline phase.893

Domen and co-workers found that high-pressure treatment of  $\beta$ -Ge<sub>3</sub>N<sub>4</sub> and Ta<sub>3</sub>N<sub>5</sub> under ammonia was effective for reducing the defect sites and/or structural imperfections both in the bulk and on the surface, resulting in enhanced photocatalytic activities for water splitting. 278,279,894 Postcalcination at moderate temperatures was found to be effective for improving the activities of  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  and  $(Zn_{1+x}Ge)(N_2O_x)$  as visible-light-driven photocatalysts for overall water splitting. This was also attributed to a reduction in the density of lattice defects in the catalyst materials, which in return reduced the number of sites available for the

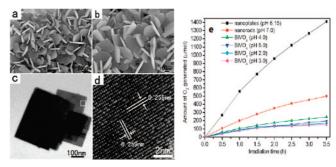
recombination of the photogenerated electrons and holes. 575,579 A H<sub>2</sub>S post-treatment was applied by Jang and co-workers for the synthesis of pure sulfide-type photocatalysts, with high crystallinities. Because of the suppressed charge recombination, the obtained AgGaS<sub>2</sub> exhibited high photocatalytic activity for hydrogen production under visible light. 496

Good crystallinity and fewer crystal defects were also discovered to benefit bulk electron-hole separation as well as the efficient photocatalytic activity of CdS.<sup>895</sup> Li et al. found that suitable photoetching could notably increase photocatalytic activity of Pt/CdS for hydrogen evolution by removing selectively grain boundary defects that were the recombination centers for photoinduced electron-hole pairs. 896 Maeda et al. found that Ta<sub>3</sub>N<sub>5</sub> nanoparticles with a lower density of defect sites exhibited enhanced photocatalytic activity for H<sub>2</sub> evolution. 472 They also reported the positive effect of a monoclinic-ZrO2 modification on the photocatalytic H<sub>2</sub> evolution activity of TaON under visiblelight irradiation. Both prior incorporation of a zirconium species into  $Zr_xTa_{1-x}O_{1+x}N_{1-x}$  solid solutions and surface modification of TaON with monoclinic-ZrO2 suppressed the formation of surface defects by inhibiting tantalum reduction during nitridation even though the ZrO<sub>2</sub> component itself did not undergo nitridation.897,898

The photocatalytic performance of semiconducting nanocrystals is also highly dictated by their crystal planes. For TiO<sub>2</sub>, different groups have reported the tailored synthesis of anatase single crystals with a high percentage of the highly reactive {001} facets. These showed superior photoreactivity compared to P25 as a benchmarking material. 899-905 Wu et al. 905 and Li and Xu906 reported that nanostructured anatase TiO<sub>2</sub> with a large percentage of other exposed high-energy facets, {010} and {100}, respectively, exhibited much higher activities than that of commercial anatase powders. Pt/TiO<sub>2</sub> nanosheets with exposed (001) facets fabricated by a hydrothermal treatment of tetrabutyl titanate and hydrofluoric acid mixture were shown to exhibit a much higher photocatalytic activity for H<sub>2</sub> evolution than Degussa P-25 TiO<sub>2</sub> and pure TiO<sub>2</sub> nanoparticles because of the synergistic effect of surface fluorination and exposed (001) facets.<sup>907</sup>

Zhao et al. prepared WO<sub>3</sub> octahedra with {111} surface facets, which were covered and stabilized with a thin tungstic acid layer. 908 The as-prepared octahedra exhibited high visible-light-driven photocatalytic activity for oxidation of H<sub>2</sub>O to O<sub>2</sub>. This was due to the high adsorption capacity of the tungstic acid sheath for Ag+ as the electron acceptor and the efficient water oxidation at the high-energy {111} facets. Nian et al. succeeded in controlling the crystalline texture of Cu<sub>2</sub>O by varying the electrodeposition conditions. <sup>909</sup> The Cu<sub>2</sub>O grown with a [111] out-of-plane texture had a better photocatalytic performance in water splitting for hydrogen production than that with a [110] texture, due to the large exposure area of {111} facets of Cu<sub>2</sub>O crystalline. 910,911 As shown in Figure 33, Xi and Ye deemed that the exposed {001} facets of the monoclinic BiVO<sub>4</sub> nanoplates led to a remarkable enhancement of the visible-light photocatalytic oxidation of water for O<sub>2</sub> generation.<sup>912</sup>

Yin et al. found evidence for the possible role of lattice vibrations in the photocatalytic  $H_2$  production activity of  $BaM_{1/3}N_{2/3}O_3$  (M = Ni, Zn; N = Nb, Ta). They found that the lower photocatalytic activity resulted from the deviation of the bond angle M-O-N from 180°. This change increased the migration energy of the charge carriers,



**Figure 33.** (a, b) Low- and high-magnification field emission scanning electron microscopy (FESEM) image of the m-BiVO<sub>4</sub> nanoplates. (c) Low-magnification transmission electron microscopy (TEM) image of the m-BiVO<sub>4</sub> nanoplates. (d) High-resolution TEM (HRTEM) image of the m-BiVO<sub>4</sub> nanoplate. (e) Photocatalytic O<sub>2</sub> evolution from an aqueous AgNO<sub>3</sub> solution (0.05 M, 270 mL) under visible-light irradiation ( $\lambda > 420$  nm) over various photocatalysts (0.1 g). Reprinted with permission from ref 912. Copyright 2010 The Royal Society of Chemistry.

thus hindering their migration of the charge carriers and leading to a higher possibility of recombination between the photogenerated electrons and holes.

Similarly, in the study of Kudo et al., <sup>21</sup> Sr<sub>2</sub>Ta<sub>2</sub>O<sub>7</sub> displayed a higher photocatalytic activity for water splitting than Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub>. As the bond angle of O-Ta-O in Sr<sub>2</sub>Ta<sub>2</sub>O<sub>7</sub> was closer to 180° than that of O-Nb-O in Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub>, the charge carriers in Sr<sub>2</sub>Ta<sub>2</sub>O<sub>7</sub> moved more easily than those in Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub>. On the other hand, Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub> was also active for the water splitting even if it was a niobate. This was due to a dipole moment along the perovskite layers resulting from the distortion of NbO<sub>6</sub> octahedra, which, it was suggested, could enhance the charge separation of the photogenerated electrons and holes in Sr<sub>2</sub>Nb<sub>2</sub>O<sub>7</sub>. In fact, a dipole moment of this type, due to the distortion of TiO<sub>6</sub> octahedra in some titanates such as BaTi<sub>4</sub>O<sub>9</sub>, was reported previously by Inoue and co-workers as being important for the charge separation in photocatalytic water splitting. 101,913,914 In their later studies, 258-260, 263, 265 the prominent effect of crystal structure distortion on photocatalytic activity was also reported for other metal oxide photocatalysts:  $Zn_2GeO_4$ ,  $MIn_2O_4$  (M = Ca, Sr),  $M_2Sb_2O_7$  (M = Ca, Sr), and AInO<sub>2</sub> (A = Li, Na). It was demonstrated that the metal oxides consisting of distorted structural units with dipole moments were all photocatalytically active in water decomposition, whereas distortion-free oxides exhibited negligible activity. Structure distortion was clearly effective for charge separation. Figure 34 compares the correlation between photocatalytic activity and dipole moment.

Recently, Shen et al. found that, in hydrogen evolution, the photocatalytic activity for  $ZnIn_2S_4$  was greatly affected by the crystal plane space along the c-axis; the increasing d (001) space greatly improved its photocatalytic activity of  $ZnIn_2S_4$ .<sup>377,915</sup> They concluded that the structural distortion of  $ZnIn_2S_4$  induced a dipole moment through increasing the d (001) space. Since the internal fields caused by the dipole moment were considered to be useful for the charge separation of the photogenerated electrons and holes, this in turn proved effective in promoting the photocatalytic activity.

#### 5.3.2. Modification of Size and Morphology

It is well-known that particle size is a crucial factor in the dynamics of electron/hole recombination (including bulk recombination and surface recombination) processes, especially in semiconductor nanomaterials, and that the move-

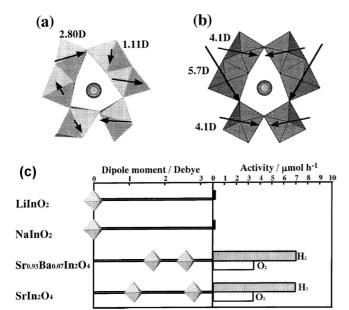


Figure 34. Dipole moments in (a) SrIn<sub>2</sub>O<sub>4</sub> and (b) BaTi<sub>4</sub>O<sub>9</sub>, and (c) correlation between photocatalytic activity and dipole moment. The octahedral structures represent InO<sub>6</sub> units. The indates with dipole moments are photocatalytically active, whereas the distortionfree indates exhibited negligible activity. Reprinted with permission from ref 260. Copyright 2003 American Chemical Society.

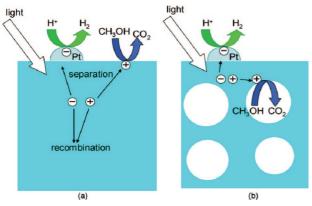
ment of electrons and holes is primarily governed by the well-known quantum confinement.<sup>300</sup> Generally, a decrease in particle size could be expected to lead to a higher efficiency in photocatalysis. 163,916-920 This was because the bulk charge recombination of photogenerated electrons and holes, dominant in the well-crystallized large semiconductor particles, was reduced by decreasing particle size. Reduction in particle size could also lead to a larger surface area and increased the available surface active sites. 209,921,922 Lee et al. reported that the smaller particle size and higher surface area of NaTaO3 led to its high photocatalytic activity in overall water splitting.<sup>209</sup> They proposed that this was due to the increased probability of surface reactions between the electrons and holes rather than recombination in the bulk. Sathish et al. 923 found that CdS nanoparticles showed a higher photocatalytic activity for hydrogen evolution compared to bulk CdS. This was correlated well with the particle size and surface area. Lunawat et al. also found that, compared to bulk CdS, its nanocrystallites dispersed into the channels of mesoporous silicate SBA-15 exhibited enhanced photocatalytic activity for water splitting under visible light. 924 This result illustrated the fact that the smaller particle size was capable of enhancing the photocatalytic activity of a semiconducting material. However, the photocatalytic activity does not monotonically increase but even decreases with the decreasing particle size. This is because surface charge recombination can become an important process. When particle size of the nanocrystalline semiconductor becomes extremely small, this in return can offset the benefits of the ultrahigh surface area of the nanocrystalline particles. 921,925

Wang et al. showed that there existed an optimal particle size for the pure nanocrystalline TiO<sub>2</sub> photocatalyst used in the photocatalytic reaction of chloroform decomposition. 926 A similar phenomenon for K<sub>2</sub>Ta<sub>2</sub>O<sub>6</sub><sup>219</sup> and CdS nanocrystals<sup>927,928</sup> was also reported. It was found that the optimal particle size was important to achieve the high photocatalytic activity for water splitting. Neither the smallest nor the largest nanocrystals showed the highest photocatalytic activity for hydrogen evolution. The smallest nanocrystals suffered more from the charge recombination on the surface, whereas, to some extent, the largest nanocrystals suffered from the same disadvantages as large-size particles. Hong et al. investigated the effects of crystal size on photoanode materials based on WO<sub>3</sub> in both a particulate suspension (PS) system and a photoelectrochemical (PEC) film system. 929 In the PS system, the large crystals showed higher photoactivity because of a well-developed space charge layer and high crystallinity, whereas in the PEC system, the smaller crystals performed better because of the reduced hole diffusion length. These different behaviors were attributed to different chargeseparation mechanisms in the two systems. Balázs and coworkers investigated the effect of particle shape on the activity of nanocrystalline TiO2 photocatalysts in phenol decomposition. 930-932 They found that the average particle diameter of the best photocatalysts was in the range of 50-70 nm, and the photocatalyst containing only polyhedral nanocrystals displayed better photocatalytic performance than those containing both spherical and polyhedral nanocrystals.

Since the first report on the ordered mesoporous silica material MCM-41, 933 inorganic materials with controlled porosity have been widely exploited in photocatalysis. 934-940 These mesoporous photocatalysts possess two major beneficial characteristics. The first is the high specific surface area, which provides more reactive sites at the surface for photocatalytic reaction to occur. The second is the nanoscale channel wall of mesopores, which can facilitate the transfer of photogenerated electrons and holes to the surface, thus avoiding their recombination in the bulk. 80,158,202 For example, different kinds of mesoporous TiO2-based photocatalysts have been previously reported as being promising candidates for efficient charge separation and transport, leading to higher efficiency in photocatalysis for water splitting.  $^{80,889,941-946}$  The mesoporous-assembled SrTiO<sub>3</sub> nanocrystal photocatalysts also exhibited much higher photocatalytic activity than the nonmesoporous-assembled commercial SrTiO<sub>3</sub> photocatalysts. 947,948 Domen and co-workers reported that, although the inorganic phases are amorphous, the photocatalytic activities for overall water decomposition over mesoporous Ta<sub>2</sub>O<sub>5</sub> and Mg-Ta oxide are much higher than over crystallized Ta<sub>2</sub>O<sub>5</sub> and MgTa<sub>2</sub>O<sub>6</sub>, respectively. <sup>202,204</sup> This high activity was due to the thin walls of the mesopores, which provided a shorter distance for the excited electrons and holes to travel to the surface.

Chen et al. found that the photocatalytic activity of mesoporous Nb<sub>2</sub>O<sub>5</sub> was 20 times higher than the bulk, which lacks porosity. 158 It was proposed that the positive effect of mesoporosity on the photocatalytic activity was abscribed to the oriented movement and efficient separation of electrons and holes, as shown in Figure 35. The mesoporous In-Nb mixed-oxide catalyst also showed photocatalytic activity for water splitting. This was attributed to the synergistic effects of a high surface area and the well-dispersed indium species intercalated into the mesoporous structure. 159

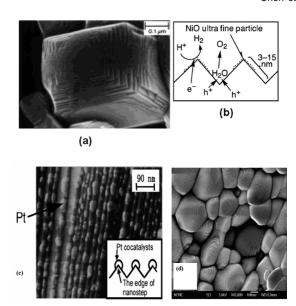
Sang and co-workers synthesized the complex oxide InVO<sub>4</sub> with a mesoporous wormhole-like structure. 949,950 It had a highly efficient photocatalytic activity due to the low recombination rate of the photoexcited electrons and holes. Ta<sub>3</sub>N<sub>5</sub> with a crystalline mesoporous framework showed the photocatalytic performance for hydrogen evolution greatly superior to that of conventional bulk Ta<sub>3</sub>N<sub>5</sub>. The thin-wall structure promoted charge transfer to active surface sites. 951 Wang and co-workers suggested that the efficiency of



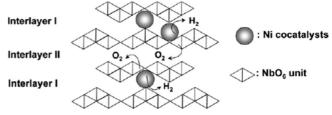
**Figure 35.** Model of photocatalytic  $H_2$  evolution on a bulk photocatalyst (a) and a mesoporous photocatalyst (b). Reprinted with permission from ref 158. Copyright 2007 Elsevier.

hydrogen production could be improved by about an order of magnitude through the introduction of mesoporosity into polymeric C<sub>3</sub>N<sub>4</sub>. 952,953 Bao et al. prepared nanoporous CdS nanostructures using a two-step aqueous route, which displayed very high photocatalytic efficiency for hydrogen production under visible-light irradiation.<sup>732</sup> The quantum yield was found to be  $\sim 60.34\%$  at 420 nm. This high activity was due the fact that the fast transport of the photogenerated carriers to the surface of the CdS by traveling only a short distance through the nanopore wall was so advantageous. It greatly decreased the bulk charge recombination, increased the quick surface charge separation and fast interfacial charge carrier transfer, and ensured the fast photochemical reaction at the CdS/electrolyte interface. Feng and co-workers developed a series of porous chalcogenides with openframework construction as efficient photocatalysts for the reduction of water:  $Na_{14}In_{17}Cu_3S_{35} \cdot xH_2O$ ,  $Cu_8Ge_5S_{16}^{4-.954-956}$ It was demonstrated that the open-framework construction helped not only to increase the number of active reaction sites owing to a high surface area but also to reduce the rate of charge recombination of the electron-hole pair.

Kudo's group was the first to report the positive effect of the ordered surface structure on the photocatalytic activity for water splitting. The photocatalytic activity of La-doped NaTaO<sub>3</sub> with an ordered surface nanostep structure was 9 times higher than that of nondoped NaTaO<sub>3</sub>. This was mainly because the ordered surface nanostep structure created by La doping (as shown in Figure 36) contributed to the separation of the photoactive sites in avoiding surface charge recombination and back-reactions. 20,216 A similar effect of the surface nanostructure was also observed on some sulfide photocatalysts, such as  $(AgIn)_{0.22}Zn_{1.56}S_2^{537}$  and  $AgGa_{0.9}$ In<sub>0.1</sub>S<sub>2</sub><sup>544</sup> solid solution. It was proposed that such nanostructured surface edges improved the photocatalytic properties because the incident photon was not shielded and the photogenerated electrons could more easily migrate to the edge of the nanostructured step. Hence, both compounds had enhanced photocatalytic activity for hydrogen evolution under visible-light irradiation. Jing and Guo prepared a highly stable and active CdS photocatalyst through thermal sulfidation of a CdO precursor. 957 The special nanostep structure was also observed at the surface of the sulfide photocatalyst, which further supported the premise that the nanostep surface was crucial for the remarkable enhancement of photocatalytic hydrogen production through active sites in suppressing the surface-charge recombination of photogenerated electrons and holes.



**Figure 36.** Ordered surface nanostep structure of NiO/NaTaO<sub>3</sub>: La (a), and mechanism of highly efficient photocatalytic water splitting over NiO/NaTaO<sub>3</sub>:La photocatalysts (b). Reprinted with permission from ref 20. Copyright 2003 American Chemical Society. Ordered surface nanostep structure of Pt/(AgIn)<sub>0.22</sub>Zn<sub>1.56</sub>S<sub>2</sub> (c) and CdS (d). (c) Reprinted with permission from ref 537. Copyright 2004 American Chemical Society. (d) Reprinted with permission from ref 957. Copyright 2006 American Chemical Society.



**Figure 37.** Schematic structure of the active  $\text{NiO}-\text{K}_4\text{Nb}_6\text{O}_{17}$  photocatalyst and the reaction mechanism of  $\text{H}_2\text{O}$  decomposition into  $\text{H}_2$  and  $\text{O}_2$ . Reprinted with permission from ref 55. Copyright 2009 The Royal Society of Chemistry.

Some ion-exchangeable layered oxides have been shown to be highly active photocatalysts with several distinct advantages compared to the so-called "bulk" type photocatalysts such as TiO<sub>2</sub> or SrTiO<sub>3</sub>. <sup>135,162,164,171,231,958–960</sup> Because of their unique structural characteristics, such layered photocatalysts utilize the intercalated water molecules as reactants to produce hydrogen and oxygen. 135,164 K<sub>4</sub>Nb<sub>6</sub>O<sub>17</sub> has an ion-exchangeable layered structure with two different kinds of alternating interlayer spaces, i.e., interlayers I and II, as shown in Figure 37, which are the active sites for the production of hydrogen and oxygen, respectively. 164 It was proposed that the electrostatic gradient in the niobate sheet caused by the different locations of the K<sup>+</sup> and H<sup>+</sup> cations between interlayer I and II could drive the electrons and holes to move to different active sites, giving rise to efficient charge separation. Domen and co-workers found that a Ni-loaded ion-exchangeable layered oxide K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub> showed high photocatalytic activity for water splitting. <sup>23,135,139</sup> K<sub>2</sub>La<sub>2</sub>Ti<sub>3</sub>O<sub>10</sub> had only one type of interlayer space, and most of the loaded Ni existed at the external surface. It was proposed that H<sub>2</sub> evolved on the Ni particles at the external surface and O2 evolved in the interlayer space. This again indicated efficient charge separation and suppression of back-reaction were achieved due to the layered structure.

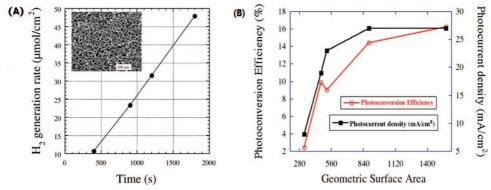


Figure 38. (A) Hydrogen photoproduction in a photoelectrochemical cell consisting of titania nanotube array photoanode and Pt mesh cathode. (inset) FESEM images of titania nanotube arrays. Reprinted with permission from ref 977. Copyright 2005 American Chemical Society. (B) Photoconversion efficiency during water photolysis and maximum photocurrent obtained using TiO<sub>2</sub> nanotube-array photoanode as a function of the geometric roughness factor of the nanotube arrays. Reprinted with permission from ref 980. Copyright 2007 IOP Publishing Ltd.

Yoshimura et al. found that different H<sub>2</sub> evolution sites in the layered structure led to quite different photocatalytic activities for hydrogen production. 959 Pt/HPb<sub>2</sub>Nb<sub>3</sub>O<sub>10</sub> with Pt in the interlayer space exhibited much higher activity than that with Pt only on the external surface. For the latter case, the electrons had to transfer a long distance through a twodimensional niobate sheet to reduce H<sup>+</sup> on the external Pt. There was a short distance for electrons to transfer to the intercalated Pt, and the possibility of charge recombination was largely reduced. They also found that the intercalation of water or hydronium ion into the interlayer space was a key factor in increasing the photocatalytic activity of layered oxides for hydrogen evolution. The intercalation resulted in a dramatic increase in reaction sites for water decomposition along with efficient charge separation. 959,961 Moreover, pillared layered photocatalyts showed a marked increase in hydrogen production activities compared with the original layered oxides. 98,189 This may be due to the increase of surface area<sup>98,962</sup> and/or interlayer spacing, <sup>189</sup> which facilitated the reaction between photogenerated charges and reactant molecules and suppressed charge recombination.

Various architectures with nanometer- and micrometersized dimensions have been found to play an important role in achieving fast charge transfer and efficient charge separation for the improvement of photocatalytic activity. 963 Onedimensional (1D) nanostructures, such as nanowires, <sup>730,964,965</sup> nanotubes, <sup>99,966–969</sup> and nanorods/nanoribbons, <sup>970–974,177</sup> have all displayed great potential in environmental/energy photocatalysis applications. Jitputti and co-workers reported that TiO<sub>2</sub> nanowires with high crystallinity and surface area could promote higher H<sub>2</sub> evolution than commercial TiO<sub>2</sub> nanoparticles. 964,975 It was proposed that the unique 1D structure might facilitate the electron transport for reacting with water molecules adsorbed at the TiO2 surface along the 1D structure. Additionally, TiO<sub>2</sub> nanofibers prepared by electrospinning techniques showed enhanced crystallinity, surface area, and photocatalytic activity for hydrogen evolution compared to nanofibers prepared by hydrothermal method. 975 Li and Lu demostrated that one apparent advantage of Na<sub>2</sub>Ti<sub>2</sub>O<sub>4</sub>(OH)<sub>2</sub> nanotubes for photocatalytic hydrogen production over TiO2 particles was their one-dimensional tubular structure, which was favorable for electron transfer and thus reduced the recombination of electron-hole pairs.<sup>649</sup>

Grimes and co-workers examined the use of titania nanotube arrays for the photocleavage of water into hydrogen and oxygen under UV irradiation. 976-982 The nanotubular architecture gave rise to superior electron lifetimes and, hence, more efficient charge separation. 977,983 On a per-watt basis, the hydrogen generation rate obtained of 24 mL/(h W) was among the highest reported for a titania-based photoelectrochemical cell, as shown in Figure 38A. 977 They also claimed that TiO2 nanotube-array photoanodes could generate H<sub>2</sub> by water photolysis with a photoconversion efficiency of 16.5% under UV light illumination (Figure 38B).<sup>980</sup> A three-step electrochemical anodization was demonstrated to contribute to the formation of highly ordered TiO<sub>2</sub> nanotube arrays and then greatly enhance the photoelectrochemical and hydrogen evolution activities.<sup>984</sup>

Cho et al. found that single-crystalline SrNb<sub>2</sub>O<sub>6</sub> nanotubes with rhombic cross sections exhibited superior photocatalytic activity for the H<sub>2</sub> evolution from pure water compared to their bulk counterparts. 985 This was attributed to the higher surface area and the smaller diffusion length of the charge carriers resulting from the morphological characteristics of the material. It was reported that the photocatalytic activity enhancement of TiO2 nanorods could be attributed to the larger surface area, 970,973 which promoted the probability of a surface reaction between the electrons and holes rather than recombination in the bulk.<sup>209</sup> Chen and Ye found that, under UV irradiation, nanorod-like SrSnO<sub>3</sub> had a photoactivity 10 times greater than the particles prepared by a solid-state reaction.<sup>268</sup> Porous nanorod-structured ZnFe<sub>2</sub>O<sub>4</sub> showed effective H<sub>2</sub> evolution from methanol aqueous solution under visible light. The nanorod shapes and their intraparticle porous structures were proven to be beneficial for more quickly transferring the photogenerated carriers onto the surfaces, which in return favored the efficient charge separation.986 The high electron mobility in 1D nanostructured nanorods also partially contributed to the high photocatalytic hydrogen yield of the (Cd<sub>0.8</sub>Zn<sub>0.2</sub>)S nanorods.<sup>987</sup> A device physics model for radial p-n junction nanorod solar cells was developed by Kayes et al. 988 Densely packed nanorods were oriented with the rod axis parallel to the incident light direction. The study indicated that high-aspectratio (length/diameter) nanorods allowed the use of a sufficient thickness of material to obtain good optical absorption while simultaneously providing short collection lengths for excited carriers in a direction normal to the light absorption. The design of the radial p-n junction nanorod device could provide large improvements in efficiency relative to a conventional planar geometry.

**Figure 39.** Model of the movements of the photoproduced electron and hole at the  $TiO_x$  nanosheet with a lepidocrosite-type structure. The electron moves in the 3d CB consisting of the  $Ti^{4+}$  network in the nanosheet and then reduces  $Ag^+$  and  $Cu^{2+}$  at the edge, whereas the hole exists at the 2p VB consisting of the  $O^{2-}$  surface and oxidizes  $Mn^{2+}$  on the surface. Reprinted with permission from ref 993. Copyright 2008 American Chemical Society.

Two-dimensional (2D) nanostructures such as nanobelts, 989-991 nanosheets, 992-994 and nanoplates 995,996 also favor the transfer of electrons and holes generated inside the crystal to the surface and promote the charge separation, which helps to enhance the photocatalytic activity to some degree. 991,997-1001 Sun et al. found that the photocatalytic properties of ZnO nanobelt arrays was better than that found for ZnO film or the rod-/comblike ZnO nanostructures under identical conditions. 989 Ultrathin nanosheets of brookite TiO2 displayed high phtocatalytic activity. The atomic scale thickness and large surface area allowed shorter migration time of carriers, suppressed bulk recombination and more electrons and holes on the surface. 1002 Jitputti et al. reported that nanosheet TiO<sub>2</sub> exhibited higher photocatalytic activity for hydrogen evolution than that of commercial TiO<sub>2</sub> anatase powder. 1003 Matsumoto et al. observed the separate photocatalytic active sites on the TiO<sub>x</sub> nanosheets by photodepositing metal and metal oxide. 993 As shown in Figure 39, the photogenerated electrons mainly reacted at the edge of the nanosheets, whereas the photogenerated holes reacted all over the surface. Thus, the recombination of electrons and holes in the  $TiO_x$ nanosheets was to some extent impeded. Osterloh and coworkers developed a modular approach to the synthesis of nanostructured catalysts for photochemical water-splitting catalysts based on exfoliated, semiconducting niobate nanosheets. 188,1004-1007 These could be integrated into twocomponent nanostructures with separate sites for water reduction and oxidation. Zhang and Zhu revealed that Bi<sub>2</sub>WO<sub>6</sub> nanoplates had a special advantage in photocatalytic reactions, because the nanoplate structure not only had a larger surface area but also promoted the efficiency of the electron-hole separation.<sup>996</sup>

Self-assembly of nanoscale building blocks into three-dimensional (3D) complex structures is another research hot spot in photocatalysis. 1008–1013 Song and Gao found that hollow NiO microspheres showed a significantly more enhanced photocatalytic activity than NiO rods. 1008 The increasing number of surface active sites and surface charge carrier transfer rate in the photocatalysis of the hollow NiO microspheres proved advantageous. Lu et al. reported that quite a good photocatalytic performance was obtained on a novel ZnO hierarchical micro/nanoarchitecture with dense nanosheet-built networks standing on hexagonal—pyramid-like microcrystals. 1009 This was because the special structural features of the micro/nanoarchitectured ZnO promoted

electrons transfer from the nanosheets with high electric-potential to the core micropyramid with low electric-potential. The probability of photogenerated electron/hole pair recombination was thus reduced. Kale et al. reported that  $CdIn_2S_4$  nanostructures with a marigold-like morphology composed of numerous nanosized petals displayed high photocatalytic activity for solar production of hydrogen from  $H_2S$  in KOH aqueous solution.  $^{1010}$ 

## 6. Photocatalytic Hydrogen Generation Systems

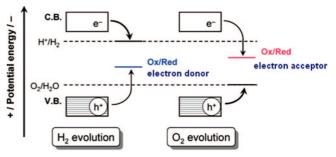
The tremendous effort put into the synthesis and modification of photocatalysts to adjust their band structures and increase the probabilty of photogenerated charge separation has resulted in the progressive development of highefficiency visible-light-driven photocatalysts. Meanwhile, it is widely believed that the construction of appropriate and efficient photocatalytic hydrogen generation systems (solution/semiconductor suspensions) based on different kinds of photocatalysts is an indispensable step to achieving high performance for photocatalytic hydrogen generation.

# 6.1. Hydrogen Generation Systems Containing Sacrificial Reagents

The basic principle of photocatalytic reactions for hydrogen/ oxygen generation using electron donors/acceptors as the sacrificial reagents is depicted schematically in Figure  $40.^{51}$  When the system is constructed in the presence of an electron donor, the photogenerated holes irreversibly oxidize the reducing electron donors instead of  $H_2O$ . If the bottom of the conduction band of the photocatalyst is located at a more negative potential than the water reduction potential, this then facilitates water reduction by the photogenerated electrons in the conduction band.  $^{51,300}$  The different kinds of typical sacrificial reagents are shown in Table 3.

#### 6.1.1. Inorganic Sacrificial Reagent Systems

**6.1.1.1.** S<sup>2-</sup>/SO<sub>3</sub><sup>2-</sup> System. Since S<sup>2-</sup> and SO<sub>3</sub><sup>2-</sup> could be oxidized by photogenerated holes to  $S_n^{2-}$  and SO<sub>4</sub><sup>2-</sup>, respectively, they can thus act independently as sacrificial reagents for photocatalytic hydrogen generation. <sup>848,849,1014–1017</sup> However, the oxidation of S<sup>2-</sup> ions to yellow polysulfides  $S_n^{2-}$  leads to a decrease in  $H_2$  formation over time. This is due to the high light absorption of the yellow polysulfide  $S_n^{2-}$  in the visible region and then to the competitive reduction of  $S_n^{2-}$  with  $H_2O$ . <sup>1014,1018–1020</sup> Fortunately,  $SO_3^{2-}$  could act as  $S_n^{2-}$ -regenerating agent and maintain the solution colorless. <sup>1021</sup> Therefore, the  $S^2$ -/SO<sub>3</sub><sup>2-</sup> mixture is most widely used as electron donors and added to the water/semiconductor suspension to improve the photocatalytic activity and stability



**Figure 40.** Basic principle of photocatalytic reactions in the presence of sacrificial reagents. Reprinted with permission from ref 51. Copyright 2007 American Chemical Society.

Table 3. Typical Sacrificial Reagent Contained Photocatalytic Hydrogen Production System

photocatalyst	mass (g)	sacrificial reagent	light Source	rate of hydrogen evolution $(\mu \text{mol} \cdot \mathbf{h}^{-1} \cdot \mathbf{g}^{-1})$	quantum yield (%)	reference
Pt-PdS/CdS	0.3	S <sup>2-</sup> /SO, <sup>2-</sup>	300-W Xe (>420 nm)	29 233	93 (420 nm)	733
Pt/WS,/TiO	0.2	$\mathbf{S}_{2}^{2-1}$	350-W Xe (>430 nm)	068		850
Pt/CdS/Ti-MCM-41	0.2	$50^{2}$	350-W Xe (>430 nm)	76	2.6 (420 nm)	849
ZnFe <sub>2</sub> O <sub>4</sub> /SrTiO <sub>3</sub>	0.25	$S_2O_3^{-2}$	M M-009	410 688	,	872
Pt-PdS/CdS	0.025	$H_2^{-}S_2^{-}$	300-W Xe (>420 nm)	$\sim 94\ 000$	30 (420 nm)	785
MoS <sub>2</sub> /CdS	0.1	lactic acid	300-W Xe (>420 nm)	$\sim$ 5300		782
$RuO_2/K_2La_2Ti_3O_{10}$ :Zn	1	_I	250-W Xe	55.5		140
Pt/TiO <sub>2</sub>	0.1	Br <sup>-</sup>	500-W Hg	$\sim$ 550		1051
$\mathrm{KBi_{3}PbTi_{5}O_{16}}$	1	$Ce^{3+}$	450-W Hg	35.21		146
$Pt/SrTiO_3$ :Rh	0.1	$\mathrm{Fe}^{2+}$	300-W Xe (>420 nm)	169		1046
NiO/TiO <sub>2</sub>	0.2	CN_	450-W Hg	1500	0.32	1052
$Pt/H_2LaNb_2O_7:In$		methanol	100-W Hg (>290 nm)	5268	1.54 (>290 nm)	183
$Pt-Ru/Y_2Ta_2O_5N_2$	0.3	ethanol	300-W Xe (>420 nm)	833		393
NiO <sub>x</sub> /CdS/KNbO <sub>3</sub>	0.2	isopropanol	500-W Hg-Xe (>400 nm)	203.5	8.8 (>400 nm)	800, 801
$Pt/TiO_2$	0.08	1-propanol	500-W Xe	$\sim 150$		1066
$Pt/TiO_2$	0.08	1-butanol	500-W Xe	$\sim$ 75		1066
Pt/Cu <sub>2</sub> O	0.01	formic acid	H (>420 nm)	155		1075
Pt/TiO <sub>2</sub>	0.1	acetic acid	300-W Hg	28 380		1064
Ni-La,0,CO,	0.1	formaldehyde	125-W Xe (>400 nm)	36		1065
Pt/TiO <sub>2</sub>	0.08	acetaldehyde	500-W Xe	45.3		1066
DPf/TiO <sub>3</sub>	0.02	oxalic acid	250-W Hg	2800		1056
Pt/TiO.:Gd	0.1	glycol	1200-W Ulfra-Vitalux	13.775		82
Wells—Dawson heteronoly blue/Pt/TiO.	0.05	olycerol	300-W H (>420 nm)	5 C9~	20.0 (670 nm)	637 638
$H_1Nh_2O_{12}/Pt/R_1(hnv)h_1/4/(PO_3H_2)hnv)^{2+}$	0.005	FDTA	300-W Xe (>420 nm)	4320	22 (450 nm)	623, 624
114110601/11 4114(0PJ)2(11,1 (1 03112)20PJ) Pd/TiO	0.05	alucose	125-W Ha	4580	(120 000)	1085
Pt/TiO.: R N	0.05	SUCTOSE	300-W XP	~2500 ~2500		455
TiO./Rh/Fosin-Y	0.02	diethanolamine	200-W H (>420 nm)	1185 5	10 27 (>420 nm)	647
TiO_/VF/Fe3+_Fosin_V	0.12	TFA	400-W H (>420 nm)	2750	19 1 (>420 mm)	654
Mos/Cds	0.1	lactic acid	300-W Xe (>420 mm)	5400	17.1 (~450 mm)	182
Pt/TiO.	0.1	nacio acid	500-W XC (~420 mm)	28		1070
Pt/T:O.	0.0	n-bentane	500-W Xe	36.7		1070
Pt/TiO,	0.3	isooctane	500-W Xe	313		1070
Pt/TiO,	0.3	n-naraffin	500-W Xe	7.7		1070
Pt/TiO,	0.3	nolvethylene	500-W Xe	÷ ; ;		1070
Pt/TiO,	0.3	per zene hen zene	500-W Xe	130		1070
Pt/TiO,	0.3	phenol	500-W Xe	011		1070
Pt/TiO <sub>2</sub>	0.3	pyridine	500-W Xe	103.3		1070
Pt/TiO <sub>2</sub>	0.3	coal	500-W Xe	30		1070
Pt/TiO2	0.3	tar sand	500-W Xe	33.3		1070
Pt/TiO,	0.3	pitch	500-W Xe	33.3		1070
Pt/TiO <sub>2</sub>	0.08	Acid Orange 7	500-W Xe	$\sim \! 502.5$		1071
Pt/TiO <sub>2</sub>	0.08	Basic Red 46	500-W Xe	$\sim 187.5$		1071
Pt/TiO <sub>2</sub>	0.08	Basic Blue 41	500-W Xe	$\sim$ 112.5		1071
$Pt/TiO_2$	0.04	monochloroacetic acid	250-W Hg	375		1073
$Pt/TiO_2$	0.04	dichloroacetic acid	250-W Hg	197		1073
$Pt/TiO_2$	0.04	trichloroacetic acid	250-W Hg	trace		1073
$Pt/TiO_2$	0.02	monoethanolamine	200-W Hg	191.7		1074
$Pt/TiO_2$	0.01	hydrazine dihydrochloride	500-W Hg	$\sim$ 8370		1076
WO <sub>2</sub> /TiO <sub>2</sub>	0.16	2,4-dichlorophenoxyacetic acid	250-W Na <sup>a</sup>	$\sim\!0.036$		1082
WO <sub>x</sub> /11O <sub>2</sub> Pt/TiO.	0.I6 0.3	4-chlorophenol Polyvinyl alcohol	250-W Na 500-W Xe	$\sim$ 0.005	~0 66 (380 nm)	1082
FU 1102	J.,	FOLY VILLY LABORING	300-14 AG	7.0.7	νος) ου.υ	0001

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photocatalyst	mass (g)	sacrificial reagent	light Source	rate of hydrogen evolution $(\mu \text{mol} \cdot \text{h}^{-1} \cdot \text{g}^{-1})$	quantum yield (%)	reference
Pt/TiO,	0.3	polyvinylchloride	500-W Xe	15	~0.35 (380 nm)	1083
Pt/TiO,	0.3	Teflon	500-W Xe	1.17	$\sim 0.027 (380 \text{ nm})$	1083
$Pt/TiO_2^2$	0.3	trichlorobenzene	500-W Xe	16	$\sim 0.37 (380 \text{ nm})$	1083
Pt/TiO,	0.3	trichloroethylene	500-W Xe	18.3	$\sim 0.42 (380 \text{ nm})$	1083
Pt/TiO,	0.3	sugar	500-W Xe	306.7	$\sim$ 7 (380 nm)	1083
Pt/TiO2	0.3	starch	500-W Xe	80	$\sim 1.8 (380 \text{ nm})$	1083
$\text{Pt/TiO}_2^2$	0.05	microwave treated soluble starch	125-W Hg	$\sim$ 1857	,	1085
$Pt/TiO_2$	0.3	cellulose (filter paper)	500-W Xe	13.3	~0.3 (380 nm)	1083
Pt/TiO <sub>2</sub>	0.3	glycine	500-W Xe	73.3	$\sim 1.7 \ (380 \ \text{nm})$	1083
Pt/TiO <sub>2</sub>	0.3	glutamic acid	500-W Xe	42	~0.97 (380 nm)	1083
$Pt/TiO_2$	0.3	proline	500-W Xe	43.3	$\sim$ 1 (380 nm)	1083
Pt/TiO <sub>2</sub>	0.3	gelatin	500-W Xe	23.7	$\sim 0.55 (380 \text{ nm})$	1083
$Pt/TiO_2$	0.3	stearic acid	500-W Xe	29.3	~0.68 (380 nm)	1083
$Pt/TiO_2$	0.3	olive oil	500-W Xe	10.7	~0.25 (380 nm)	1083
Pt/TiO <sub>2</sub>	0.3	Chlorella	500-W Xe	06	$\sim 2.1 (380 \text{ nm})$	1083
Pt/TiO <sub>2</sub>	0.3	Laver	500-W Xe	110.7	$\sim 2.6 (380 \text{ nm})$	1083
$Pt/TiO_2$	0.3	rice plant	500-W Xe	58.3	$\sim 1.35 (380 \text{ nm})$	1083
$Pt/TiO_2$	0.3	turf	500-W Xe	32.7	~0.75 (380 nm)	1083
Pt/TiO <sub>2</sub>	0.3	cockroach	500-W Xe	28.7	~0.66 (380 nm)	1083
$Pt/TiO_2$	0.3	human (urine, feces)	500-W Xe	58.7	$\sim 1.43 (380 \text{ nm})$	1083
$Pt/TiO_2$	0.3	cow dung	500-W Xe	99	$\sim$ 1.52 (380 nm)	1083
<sup>a</sup> Sodium lamp.						
•						

for hydrogen evolution from water.  $^{732,733,1022-1027}$  For example, a quamtum yield of  $\sim 93\%$  at 420 nm has recently been reported for the Pt–PdS/CdS photocatalyst, the highest photocatalytic activity so far achieved for hydrogen production in the presence of sacrificial reagents of  $S^{2-}/SO_3^{2-}$  under visible-light irradiation.  $^{733}$  The reaction mechanism in the presence of  $S^{2-}/SO_3^{2-}$  as sacrificial reagent is described by eqs 6-11.  $^{732}$ 

photocatalyst 
$$+ hv \rightarrow h^+ + e^-$$
 (6)

$$2e^{-} + 2H_{2}O \rightarrow H_{2} + 2OH^{-}$$
 (7)

$$SO_3^{2-} + 2OH^- + 2h^+ \rightarrow SO_4^{2-} + 2H^+$$
 (8)

$$2S^{2-} + 2h^+ \to S_2^{2-} \tag{9}$$

$$S_2^{2-} + SO_3^{2-} \rightarrow S_2O_3^{2-} + S^{2-}$$
 (10)

$$SO_3^{2-} + S^{2-} + 2h^+ \rightarrow S_2O_3^{2-}$$
 (11)

Since  $S_2O_3^{2-}$  could be oxidized to  $SO_3^{2-}$  and subsequently to  $SO_4^{2-}$  by photogenerated holes, it has also been occasionally used as a sacrificial reagent for photocatalytic hydrogen generation.  $^{870,871,1028-1030}$ 

**6.1.1.2.** H<sub>2</sub>S-Splitting System. A particularly meaningful system from the point of view of energy conservation and environmental protection is the photocatalytic hydrogen generation from H<sub>2</sub>S dissolved in water or alkali solution. <sup>398,785,806,1010,1031–1041</sup> This approach via H<sub>2</sub>S could have practical applications using waste gas H<sub>2</sub>S from chemical industries such as natural gas utilization and desulfurization process in petrochemical plants. <sup>56,1042</sup> When H<sub>2</sub>S is dissolved in alkaline aqueous solution, sulfide ions are formed and henceforth can play the role of sacrificial reagent. As described by eqs 6 and 12–14, the overall process corresponds to H<sub>2</sub>S splitting by two photons of visible light. This requires 39.3 kJ/mol. <sup>1043,1044</sup>

$$H_2S + OH^- \rightarrow HS^- + H_2O \tag{12}$$

$$2HS^{-} + 2h^{+} \rightarrow S_{2}^{2-} + 2H^{+}$$
 (13)

$$2H^{+} + 2e^{-} \rightarrow H_{2}$$
 (14)

Recently, Li and co-workers developed a nonaqueous ethanolamine solution as the solvent and reaction medium for the direct splitting of  $H_2S$  into  $H_2$  and S on a CdS-based photocatalyst under visible-light irradiation. The potential of  $H_2S$  splitting in ethanolamine was greatly lowered, and the photogenerated electrons could be fully used to reduce protons for hydrogen production. The quantum efficiency was as high as 30% under visible-light irradiation. They also investigated hydrogen generation using gas—solid-phase photocatalytic  $H_2S$  splitting on semiconductor photocatalysts. However, performance under gas—solid-phase conditions was substantially lower than that under liquid—solid-phase conditions.

**6.1.1.3.** Other Inorganic Sacrificial Reagent Systems. Other inorganic ions, such as Fe<sup>2+</sup>, <sup>274,752,1046,1047</sup> Ce<sup>3+</sup>, <sup>146,1048,1049</sup> I<sup>-</sup>, <sup>140,351,613,625,626,642–644,1050</sup> Br<sup>-</sup>, <sup>1051</sup> and CN<sup>-</sup>, <sup>1052</sup> have also been used as sacrificial reagents for hydrogen generation. These inorganic ions were easily oxidized by the photoge-

nerated holes to Fe<sup>3+</sup>, Ce<sup>4+</sup>, I<sub>3</sub><sup>-</sup> (or IO<sub>3</sub><sup>-</sup>), Br<sub>2</sub>, and OCN<sup>-</sup>, respectively. The result is hydrogen generation via water reduction by photogenerated electrons. Taking I<sup>-</sup> as an example of an inorganic sacrificial reagent, the photocatalytic mechanism for hydrogen generation is described as eq 6 and eqs 15 and 16. <sup>1050</sup>

$$2H^{+} + 2e^{-} \rightarrow H_{2}$$
 (15)

$$3I^- + 2h^+ \rightarrow I_3^-$$
 (in acidic solution)  
 $I^- + 6OH^- + 6h^+ \rightarrow IO_3^- + 3H_2O$  (in basic solution) (16)

Some photooxided species, such as  $Fe^{3+}$ ,  $Ce^{4+}$ , and  $IO_3^-$ , could be easily reduced by photogenerated electrons back to  $Fe^{2+}$ ,  ${}^{269,336,764,1053}$   $Ce^{3+}$ ,  ${}^{269,1049}$  and  $I^-$ ,  ${}^{1054}$  respectively. Thus, they could act as electron acceptors for photocatalytic oxygen generation from aqueous solution.  $Fe^{2+}/Fe^{3+}$ ,  $Ce^{3+}/Ce^{4+}$ , and  $I^-/IO_3^-$  have therefore been used as redox mediators to construct a suitable system for overall water splitting using the Z-scheme system shown in section 6.2.2.

## 6.1.2. Organic Sacrificial Reagent System

Organic compounds, such as alcohols (methanol, ethanol, isopropanol, etc.),  $^{82,91,1054-1061}$  organic acids (formic acid, acetic acid, etc.),  $^{91,938,1062-1064}$  and aldehydes (formaldehyde, acetaldehyde, etc.) $^{91,1056,1065,1066}$  have all been used as electron donors for photocatalytic hydrogen generation. Among them, methanol was most widely used, and the hydrogen generation process is described as eq 6 and eqs 17-21: $^{183,184,1067}$ 

$$h^{+} + H_{2}O \rightarrow \cdot OH + H^{+} \tag{17}$$

$$CH_3OH + \bullet OH \rightarrow \bullet CH_2OH + H_2O$$
 (18)

$$\cdot \text{CH}_2 \text{OH} \rightarrow \text{HCHO} + \text{H}^+ + \text{e}^-$$
 (19)

$$2H_2O + 2e^- \rightarrow H_2 + 2OH^-$$
 (20)

overall reaction: 
$$CH_3OH \xrightarrow{hv,catalyst} HCHO + H_2$$

(21)

The product, formaldehyde (HCHO), could be further oxidized to methanoic acid HCOOH and subsequently to  $CO_2$  together with hydrogen generation via eqs 22 and 23: $^{1054,1068,1069}$ 

$$\text{HCHO} + \text{H}_2\text{O} \xrightarrow{hv,\text{catalyst}} \text{HCOOH} + \text{H}_2$$
 (22)

$$HCOOH \xrightarrow{hv, catalyst} CO_2 + H_2$$
 (23)

In these photocatalytic systems, organic compounds are oxidized and decomposed by the photogenerated holes. Meanwhile, the remaining photogenerated electrons reduce water to hydrogen. Thus, it can be envisaged that a bifunctional photocatalytic system could be constructed, in which organic pollutants will be used to act as electron donors that achieve photocatalytic production of hydrogen from polluted water and simultaneous degradation of organic pollutants. <sup>91,1066</sup> Hashimoto et al. pioneered the work on

photocatalytic  $H_2$  generation from fossil fuels and hydrocarbons with water with powdered Pt/TiO<sub>2</sub> catalyst suspended in solution. The mechanism for photocatalytic  $H_2$  generation from aqueous solutions of hydrocarbons could be explained by eq 6 and eqs 24-30.

$$H_2O + h^+ \rightarrow \cdot OH + H^+$$
 (24)

$$H^+ + e^- \rightarrow 1/2H_2$$
 (25)

$$RCH_2CH_3 + 2 \cdot OH \rightarrow RCH_2CH_2OH + H_2O$$
 (26)

$$RCH_2CH_2OH \rightarrow RCH_2CHO + H_2$$
 (27)

$$RCH_2CHO + H_2O \rightarrow RCH_2COOH + H_2$$
 (28)

photo-Kolbe reaction: 
$$RCH_2COOH \rightarrow RCH_3 + CO_2$$
 (29)

overall reaction: 
$$RCH_2CH_3 + 2H_2O \xrightarrow{hv, catalyst}$$

$$RCH_3 + 3H_2 + CO_2 \quad (30)$$

To date, different kinds of model pollutants (azo-dyes, oxalic acid, formic acid, formaldehyde, chloroacetic acids, acetic acid, EDTA, dithiothreitol, TEA, hydrazine, dichlorophenoxyacetic acid, chlorophenol, etc.) were used as electron donors to construct the bifunctional photocatalytic systems. The photocatalytic decomposition of organic pollutants happened synchronously with efficient hydrogen generation. 649,1056,1064,1071–1082 Furthermore, it has been suggested that the photocatalytic reforming of biomass, which serves as the energy resource in plants and animals, may be anthor promising way of producing hydrogen from water. Kawai and Sakata constructed a water splitting system based on TiO2 catalyst for photocatalytic hydrogen production involving decomposition of protein, algae, dead insects, and excrement with CO2 as the only byproduct. 1083,1084 Carbohydrates, such as starch, cellulose, glycerol, glucose, and sucrose, were used as the sacrificial electron donors to construct water-splitting system for hydrogen generation by photocatalytic reforming of biomass over metal-loaded TiO<sub>2</sub> catalysts as well. 455,755,1084–1089 A probable mechanism for the photocatalytic reforming of glucose for hydrogen generation over Pt/TiO<sub>2</sub> was proposed by Fu et al. <sup>1085</sup> The reactions involved in the mechanism are summarized by eqs 31-37 and Figure 41.

$$Pt/TiO_2 + hv \rightarrow e^- + h^+ \tag{31}$$

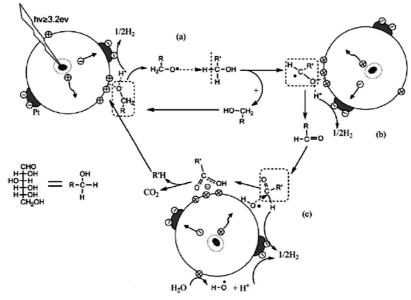


Figure 41. Proposed mechanism of the photocatalytic reforming of glucose on Pt/TiO<sub>2</sub>. Reprinted with permission from ref 1085. Copyright 2008 Elsevier.

$$h^+ + H_2O \rightarrow H^+ + \bullet OH \tag{32}$$

$$H^+ + e^- \rightarrow 1/2H_2$$
 (33)

$$R'\dot{C}H_2OH + h^+ \rightarrow H^+ + R'\dot{C}HO$$
 →  $R'CHO$   
 $H^+ + e^- \rightarrow 1/2H_2$  (35)

R'CHO + ·OH 
$$\rightarrow$$
 [R'COOH]<sup>-</sup> + H<sup>+</sup>
H<sup>+</sup> + e<sup>-</sup>  $\rightarrow$  1/2H<sub>2</sub>

[R'COOH]<sup>-</sup> + h<sup>+</sup>  $\rightarrow$  R'H + CO<sub>2</sub>

overall reaction: 
$$C_6H_{12}O_6 + 6H_2O \xrightarrow{hv,catalyst} 6CO_2 +$$

 $12H_2$  (37)

In spite of the limited work in this special area, the results are encouraging and promising for the construction of photocatalytic systems involving efficient hydrogen generation as well as pollutant decomposition or biomass reforming.

## 6.2. Overall Water-Splitting Systems

#### 6.2.1. Pure Water-Splitting System

As illustrated in Figure 3, the photocatalyst acts as two roles, H<sub>2</sub> generators and O<sub>2</sub> generators in the overall pure water-splitting system. Over the past several decades, a number of photocatalysts have been successfully developed to construct an overall water-splitting system for simultaneous hydrogen and oxygen generation in the absence of sacrificial reagents. 20,22,24,51,353,1090 Among them,  $(Ga_{1-x}Zn_x)(N_{1-x}O_x)$  has been proven as the most promising photocatalyst for overall water splitting.<sup>575</sup> Steady and stoichiometric H<sub>2</sub> and O<sub>2</sub> evolutions were found to evolve with a quantum yield of 5.9% in the range of 420–440 nm. 575

## 6.2.2. Biomimetic Z-Scheme Water-Splitting System

The biomimetic Z-schem system mechanism using reversible redox mediators has been investigated with a view to constructing a photocatalytic system for overall water splitting into H<sub>2</sub> and O<sub>2</sub>. For example, Arakawa and co-workers constructed an artificial Z-scheme system for overall water splitting using WO<sub>3</sub> photocatalyst and a Fe<sup>2+</sup>/Fe<sup>3+</sup> redox mediator. 1091, 1092 Kozlova et al. investigated the overall water splitting over a Pt/TiO<sub>2</sub> catalyst with a Ce<sup>3+</sup>/Ce<sup>4+</sup> shuttle charge transfer system. However, in these studies, the photocatalytic activities for O<sub>2</sub> and H<sub>2</sub> production were low, due to the back-reaction and the interactional reaction characteristics. Fujihara et al. constructed a Z-scheme watersplitting system using a TiO<sub>2</sub>-rutile photocatalyst and two redox mediators (Br<sub>2</sub>/Br<sup>-</sup> and Fe<sup>3+</sup>/Fe<sup>2+</sup>) in a two-compartment cell. 1051 As the production of H<sub>2</sub> and O<sub>2</sub> were carried out in separated compartments, the reversible reactions on photocatalysts, which often suffered from the effects of backreactions, were largely prevented.

Abe et al. used an innovatively designed Z-scheme system for overall water splitting into H<sub>2</sub> and O<sub>2</sub> using a two-step photoexcitation. 1093 It was composed of an IO<sub>3</sub>-/I- shuttle redox mediator and two different photocatalysts: Pt-loaded anatase TiO<sub>2</sub> for H<sub>2</sub> evolution and rutile TiO<sub>2</sub> for O<sub>2</sub> evolution. The proposed photocatalytic reaction mechanism for the Z-scheme water-splitting system is depicted schematically in Figure 42. Under UV irradiation, simultaneous gas evolution of  $H_2$  (180  $\mu$ mol/h) and  $O_2$  (90  $\mu$ mol/h) was observed from a basic (pH = 11) NaI aqueous suspension of these two different TiO2 photocatalysts. The overall water splitting proceeded by the redox cycle between IO<sub>3</sub><sup>-</sup> and I<sup>-</sup> under basic conditions as follows: (a) water reduction to H<sub>2</sub> and I<sup>-</sup> oxidation to IO<sub>3</sub><sup>-</sup> over Pt-TiO<sub>2</sub>-anatase and (b) IO<sub>3</sub><sup>-</sup> reduction to I<sup>-</sup> and water oxidation to O<sub>2</sub> over TiO<sub>2</sub>-rutile. IO<sub>3</sub><sup>-</sup> reduction to I<sup>-</sup> over Pt-TiO<sub>2</sub>-anatase was an undesirable side reaction. If this reaction was suppressed, the total water-splitting reaction took place more efficiently. The

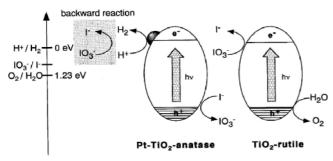


Figure 42. Proposed photocatalytic reaction mechanism for Z-scheme water-splitting system using an IO<sub>3</sub><sup>-</sup>/I<sup>-</sup> redox mediator and a mixture of Pt-TiO<sub>2</sub>-antase and TiO<sub>2</sub>-rutile photocatalysts. Reprinted with permission from ref 1093. Copyright 2001 Elsevier.

advantage of this system was that H<sub>2</sub> gas was evolved only over the Pt-TiO<sub>2</sub>-anatase photocatalyst and that O<sub>2</sub> gas was evolved over the TiO2-rutile photocatalyst only, even from a mixture of IO<sub>3</sub><sup>-</sup> and I<sup>-</sup> in a basic aqueous solution. Therefore, another undesirable backward reaction, H<sub>2</sub>O formation from H<sub>2</sub> and O<sub>2</sub> on Pt particles, was suppressed. 1093

In the later studies, a series of Z-scheme overall watersplitting systems active under visible-light irradiation were constructed, such as (Pt/ATaO<sub>2</sub>N (A = Ca, Sr, Ba))-(Pt/  $WO_3$ )- $(IO_3^-/I^-)$ ,  $^{395,1094}$  (Pt/TaON)- $(Pt/WO_3)$ - $(IO_3^-/I^-)$ ,  $^{471}$ (Pt/SrTiO<sub>3</sub>:Cr/Ta)-(Pt/WO<sub>3</sub>)-(IO<sub>3</sub>-/I<sup>-</sup>), <sup>345,1095,1096</sup> (Pt/ZrO<sub>2</sub>- $TaON)-(Pt/WO_3)-(IO_3^-/I^-)$ , 897,1097 (Pt/TaON)-(RuO<sub>2</sub>/  $TaON) - (IO_3^-/I^-),^{1098} (Ru/SrTiO_3:Rh) - (BiVO_4) - (Fe^{3+}/I^-),^{1098} (Ru/SrTiO_3:Rh) - (Fe^{3+}/I^-),^{1098} (Ru/SrTiO_3$  $Fe^{2+}$ ), 752 (Pt/SrTiO<sub>3</sub>:Rh)-(Bi<sub>2</sub>MoO<sub>6</sub>)-(Fe<sup>3+</sup>/Fe<sup>2+</sup>), 1046 (Pt/  $SrTiO_3:Rh)-(WO_3)-(Fe^{3+}/Fe^{2+}),^{1046,1047}$   $Rh)-(BiVO_4)-(Fe^{3+}/Fe^{2+}),^{1046}$  (Pt/ZrO (Pt/SrTiO<sub>3</sub>:  $(Pt/ZrO_2-TaON)-(Ir/$  $Ta_3N_5-TiO_2$ )- $(IO_3^-/I^-)$ ,  $^{1099}$  and (coumarin-dye-adsorbed Pt/  $H_4Nb_6O_{17}$ )- $(IrO_2-Pt/WO_3)-(IO_3^-/I^-)$ . These are summarized in Table 4. Of these, the (Pt/ZrO<sub>2</sub>-TaON)-(Pt/ WO<sub>3</sub>)-(IO<sub>3</sub><sup>-</sup>/I<sup>-</sup>) system showed the highest photocatalytic activity for overall water splitting with a quantum efficiency of ca. 6.3% at 420.5 nm. In this Z-scheme overall watersplitting system, the H<sub>2</sub> and IO<sub>3</sub><sup>-</sup> production proceeded over the Pt/ZrO<sub>2</sub>-TaON photocatalyst; meanwhile, the IO<sub>3</sub>reduction and water oxidation to O<sub>2</sub> took place over Pt-WO<sub>3</sub> under visible-light irradiation. 1097 In general, a redox mediator (such as  $IO_3^-/I^-$  and  $Fe^{3+}/Fe^{2+}$ ) is required for the construction of a Z-scheme photocatalytic system. Undesirable side effects such as backward reactions to form water from the H<sub>2</sub> and O<sub>2</sub> evolved can occur.

Recently, Kudo et al. had a surprising success in fabricating an overall water-splitting system driven by a Z-scheme interparticle electron transfer between H<sub>2</sub>- and O<sub>2</sub>-photocatalysts (Ru/SrTiO3:Rh-BiVO4) without a redox media-

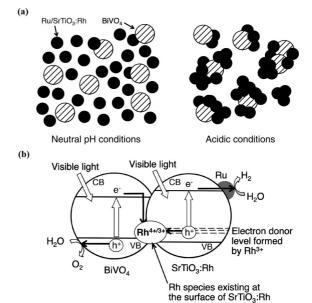


Figure 43. Mechanism of water splitting using the Z-scheme photocatalysis system driven by electron transfer between H<sub>2</sub>- and O<sub>2</sub>-photocatalysts. (a) Suspension of Ru/SrTiO<sub>3</sub>:Rh and BiVO<sub>4</sub> at neutral and acidic conditions. (b) Scheme of photocatalytic water splitting. Reprinted with permission from ref 1101. Copyright 2009 American Chemical Society.

tor. 1101 This is shown in Figure 43, in which the undesirable reactions and negative effects by an electron mediator were excluded. Although the activity of a Z-scheme overall watersplitting system is quite low and only a little higher than that achieved by conventional pure water-splitting systems using one single photocatalyst, these results provided a promising approach to constructing efficient overall watersplitting systems.

## 7. Summary and Prospects

So far, the sacrificial reagent-containing water-splitting systems constructed based on the Pt/CdS, 732 Pt-PdS/CdS, 733 and Zn/Cr layered double hydroxide<sup>667</sup> photocatalysts demonstrated the best performance for hydrogen production and oxygen production, with the highest quantum yields of ca. 60.35%, 93%, and 60.1%, respectively, at 420 nm. However, the efficiency of an overall water-splitting system based on visible-light-driven photocatalysts has been still quite low, with the maximum quantum efficiency of ca. 5.9% over  $Rh_{2-y}Cr_yO_3/GaN-ZnO$  in the range of 420-440 nm.<sup>575</sup> This is still far from the quantum efficiency (ca. 30% at 600 nm) designated the initial starting point for practical applications<sup>51,56</sup>

Table 4. Z-Schematic Photocatalytic System for Water Splitting to Hydrogen and Oxygen under Visible-Light Irradiation

					activity (µmo	$ol \cdot h^{-1} \cdot g^{-1}$		
H <sub>2</sub> photocatalyst	O2 photocatalyst	mediator	light source	incident light	$H_2$	$O_2$	QY	ref
Pt/TaON (0.2 g)	Pt/WO <sub>3</sub> (0.2 g)	IO <sub>3</sub> -/I-	300-W Xe	>420 nm	120	60	0.4 (420 nm)	471
Pt/CaTaO <sub>2</sub> N (0.1 g)	$Pt/WO_3 (0.1 g)$	$IO_3^-/I^-$	300-W Xe	>420 nm	43	21		395
Pt/BaTaO <sub>2</sub> N (0.1 g)	Pt/WO <sub>3</sub> (0.1 g)	$IO_3^-/I^-$	300-W Xe	>420 nm	67	33	0.1 (420-440 nm)	395
Pt/SrTiO <sub>3</sub> :Cr/Ta (0.2 g)	Pt/WO <sub>3</sub> (0.325 g)	IO <sub>3</sub> <sup>-</sup> /I <sup>-</sup>	300-W Xe	>420 nm	80	24.6	1 (420 nm)	345, 1095, 1096
Pt/ZrO <sub>2</sub> -TaON (0.025 g)	Pt/WO <sub>3</sub> (0.05 g)	$IO_3^-/I^-$	300-W Xe	>420 nm	1040	266	6.3 (420.5 nm)	897, 1097
Pt/TaON (0.05 g)	RuO <sub>2</sub> /TaON (0.05 g)	$IO_3^-/I^-$	300-W Xe	>420 nm	60	30	0.1 - 0.2	1098
Ru/SrTiO <sub>3</sub> :Rh (0.05 g)	BiVO <sub>4</sub> (0.05 g)	Fe3+/Fe2+	300-W Xe	>420 nm	378	179	0.3 (420 nm)	752
Pt/SrTiO <sub>3</sub> :Rh (0.02 g)	$Bi_2MoO_6 (0.02 g)$	Fe3+/Fe2+	300-W Xe	>420 nm	950	445	0.2 (440 nm)	1046
Pt/SrTiO <sub>3</sub> :Rh (0.02 g)	$WO_3 (0.02 g)$	Fe3+/Fe2+	300-W Xe	>420 nm	390	200	0.2 (440 nm)	1046
Pt/SrTiO <sub>3</sub> :Rh (0.02 g)	$BiVO_4 (0.02 g)$	Fe3+/Fe2+	300-W Xe	>420 nm	750	360	0.3 (440 nm)	1046
Pt/ZrO <sub>2</sub> -TaON (0.05 g)	$Ir/TiO_2/Ta_3N_5$ (0.05 g)	$IO_3^-/I^-$	300-W Xe	>400 nm	~45	$\sim$ 22		1099
coumarin/Pt/H <sub>4</sub> Nb <sub>6</sub> O <sub>17</sub> (0.05 g)	$IrO_2$ -Pt/WO <sub>3</sub> (0.1 g)	$IO_3^-/I^-$	300-W Xe	>400 nm	48.3	12.1	<0.1 (500 nm)	1100
Ru/SrTiO <sub>3</sub> :Rh	BiVO <sub>4</sub>	-	300-W Xe	>400 nm	400	190	1.7 (420 nm)	1101

or the critical conversion efficiency of light energy to hydrogen from photocatalytic water decomposition (15%). Therefore, more efficient visible-light-driven photocatalysts need to be developed to construct the necessay higherficiency and cost-effective water-splitting systems.

To develop more efficient visible-light-driven photocatalysts, it is necessary to narrow the band gaps to harvest visible light in the longer-wavelength regions and enhance the photogenerated charge separation in photocatalysis. Thus, factors such as electronic properties, chemical composition, structure and crystallinity, surface states, and morphology, all of which determine the photocatalytic activity of such materials, need to be further elucidated in great detail. Band gap engineering for the modification of the band structure of semiconductor photocatalysts using ion doping, semiconductor sensitization, or solid solution all present significant opportunities to render such materials active in the visiblelight region. Computational prediction based on first-principle calculations could not only provide an efficient way to identify a candidate but also suggest useful processing and production conditions. The crystallinity and crystal structure, surface structure, and morphology of photocatalysts on a nanometric scale need to be investigated systematically and modified to optimize the preparation methods and refine the materials for maximizing efficiency. These are the properties that have a major impact on charge separation, mobility, and the lifetime of the photogenerated electrons and holes. The loading of cocatalysts on the surface of the host photocatalysts has been shown to be rather effective in inhibiting charge recombination. Alternative more economical cocatalysts, such as the non-noble metals and other derived metalbased compounds, also need to be tested as possible substitutes for the most frequently used noble metals such as Pt, which although very efficient is expensive. Additionally, new insights are needed into the water-splitting mechanism, particularly with regards to identification of any thermodynamic and kinetic bottlenecks. This would facilitate the design of the most effective photocatalytic water-splitting systems.

Presently, the available efficiency for overall water (or natural water<sup>1103–1105</sup>) splitting systems for simultaneous hydrogen and oxygen production under visible-light irradiation is still quite low due to fast charge recombination and backward reactions. To achieve enhanced and sustainable hydrogen production, the continual addition of electron donors is required to make up half of the water-splitting reaction to reduce H<sub>2</sub>O to H<sub>2</sub>. These sacrificial electron donors can irreversibly consume photogenerated holes, thus prohibiting undesirable charge recombination. Taking into account the lowering cost for solar-to-H<sub>2</sub> energy conversion, polluting byproduct from industries and low-cost renewable biomass from animals or plants are preferential sacrificial electron donors in water-splitting systems. At little or no cost, they could be exploited to accomplish both the tasks of hydrogen production and waste treatment and biomass reforming simultaneously. The molecular mechanisms and reaction kinetics need to be considered carefully when designing such photocatalytic hydrogen production systems. Nevertheless, such an admirable goal for the practical application of water-splitting systems is especially interesting in light of worldwide energy and environmental concerns.

The current lack of industrial applications of the semiconductor-based photocatalytic hydrogen generation is largely due to two reasons: the low photocatalytic efficiency and the lack of extensive studies for a successful scale-up of the laboratory setup into an industrially relevant scale. To directly utilize solar light in the open air, a compound parabolic concentrator (CPC)-based photocatalytic hydrogen production reactor was recently designed by Jing et al. <sup>1106</sup> Efficient photocatalytic hydrogen production under direct solar light was accomplished by coupling a tubular reactor with the CPC concentrator. This demonstration drew attention for further studies in this promising direction. Nevertheless, both for material and reactor design, reduction of cost will have to be given special priority, before the final utilization of semiconductor-based photocatalytic hydrogen generation. <sup>1107</sup>

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## 9. References

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